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2 **Cubic zirconia in >2370 °C impact melt records Earth's hottest crust**

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## Cubic zirconia in >2370 °C impact melt records the hottest crust on Earth

### Highlights

- Zircon has partially dissociated in impact melt rock from a Canadian impact crater
- Former presence of cubic ZrO<sub>2</sub> is crystallographically encoded in reaction rims
- Cubic zirconia required >2370 °C melt, which is hottest recorded on Earth's surface
- Such superheated melt susceptible to devolatilization resulting in dry rigid crust
- Potential global effects for crustal evolution during bombardment of early Earth

23 **ABSTRACT**

24 Bolide impacts influence primordial evolution of planetary bodies because they can  
25 cause instantaneous melting and vaporization of both crust and impactors.  
26 Temperatures reached by impact-generated silicate melts are unknown because  
27 meteorite impacts are ephemeral, and established mineral and rock thermometers have  
28 limited temperature ranges. Consequently, impact melt temperatures in global  
29 bombardment models of the early Earth and Moon are poorly constrained, and may  
30 not accurately predict the survival, stabilization, geochemical evolution and cooling  
31 of early crustal materials. Here we show geological evidence for the transformation of  
32 zircon to cubic zirconia plus silica in impact melt from the 28 km diameter Mistastin  
33 Lake crater, Canada, which requires super-heating in excess of 2370 °C. This new  
34 temperature determination is the highest recorded from any crustal rock. Our phase  
35 heritage approach extends the thermometry range for impact melts by several hundred  
36 degrees, more closely bridging the gap between nature and theory. Profusion of >  
37 2370 °C superheated impact melt during high intensity bombardment of Hadean Earth  
38 likely facilitated consumption of early-formed crustal rocks and minerals, widespread  
39 volatilization of various species, including hydrates, and formation of dry, rigid,  
40 refractory crust.

41

42 **KEYWORDS:** cubic zirconia, zircon, phase transformation, impact melt, planetary  
43 evolution, early Earth

44

45

46 **1. INTRODUCTION**

47 Shock wave propagation during hypervelocity impact can melt and vaporize  
48 both the impactor and target rocks (Melosh, 1989). The immediate, post-impact,  
49 thermal pulse far exceeds both ultra-high-temperature metamorphism in tectonic  
50 settings (Heisinger and Head, 2006; Korhonen et al., 2014) and the liquidus  
51 temperatures of target rocks, and is therefore important for the evolution of planet and  
52 asteroid surfaces. The heat associated with impacts had a profound significance on the  
53 early Earth, and has been shown to have affected processes in Earth's core (Arkani-  
54 Hamed and Ghods, 2011; Monteux et al., 2015; Sleep, 2016), mantle (Watters et al.,  
55 2009), crust and atmosphere (Marchi et al., 2016; Marchi et al., 2014; Marchi et al.,  
56 2013; Melosh, 2008).

57 Volatile depletion and isotope fractionation of Earth, Moon and other planetary  
58 bodies are attributed to giant basin-forming impacts (Albarede et al., 2013; Day and  
59 Moynier, 2014; Moynier et al., 2010; Pringle et al., 2014). Vaporization of target and  
60 impactor material during impacts is temperature dependant. Preferential mass-  
61 dependant volatile loss and isotope fractionation from impact melts can therefore  
62 locally influence subsequent remelting and rheological behaviour, and so giant  
63 impacts and/or the cumulative effects of intense periods of impact bombardment have  
64 important geodynamic consequences (Albarede, 2009). These effects, in turn, have  
65 implications for the evolution and habitability of Earth's surface environment  
66 (Abramov and Mojzsis, 2009; Ryder, 2002), and, by extension, the habitability of  
67 extraterrestrial planetary bodies and exoplanets.

68 Initial thermal properties of impact melt also control their ability to digest pre-  
69 existing solid material and entrained debris (Onorato et al., 1978), including metals  
70 and sulphides essential for forming impact-generated economic ore deposits (Keays

71 and Lightfoot, 2004), their cooling rate (Onorato et al., 1978), and consequently the  
72 preservation of all pre-existing material and high-P phases formed during the passage  
73 of the shock wave (Tschauner et al., 2014). The highest temperature superheated  
74 melts (i.e., above the liquidus - the equilibrium temperature for complete melting) and  
75 vapor are produced during shock-release melting and frictional melting, whereas  
76 exhumation-related decompression melting is limited to sub-liquidus temperatures  
77 (Ahrens and O'Keefe, 1971; French and Koeberl, 2010; Melosh, 1989; Melosh, 2005;  
78 Riller et al., 2010; Spray, 1998).

79 The critical melting temperature of target rocks depends on their composition,  
80 proximity to the impact site, initial conditions (Gibbons et al., 1975; Hörz et al., 2005;  
81 Stöffler, 1971) and porosity (Kieffer et al., 1976; Wünnemann et al., 2008).

82 Vaporization temperature is taken as the highest vaporization temperature of oxides  
83 present in the system; in real systems, incongruent vaporization may occur over a  
84 range of temperatures (Ahrens and O'Keefe, 1971; Grieve et al., 1977; Lamoreaux et  
85 al., 1987; Lou et al., 1985). However, the complex behaviour of silicates has made  
86 shock-related heating and, thus, the magnitudes of impact melt superheating and  
87 vaporization difficult to predict with any degree of certainty (Ahrens and O'Keefe,  
88 1971). Accordingly, our understanding of temperatures achieved by impact melts  
89 necessarily relies on empirical constraints from the rock record via established  
90 geothermometers, which have inherently limited temperature ranges (Asimow and  
91 Ghiorso, 1998; Cherniak et al., 2007; Hart and Davis, 1978; Kubaschewski, 1982;  
92 Lindsley and Andersen, 1983; Sack and Ghiorso, 1991; Taylor et al., 2015). As there  
93 are large ranges between liquidus and vaporization temperatures for most rock types,  
94 maximum impact melt temperatures are largely unaccounted for, thus the effects of  
95 superheated melt may be underappreciated in planetary surface studies.

96 In the rock record, glass spherules and some types of microtektites contained  
97 within impact ejecta horizons provide *a priori* evidence for atmospheric condensation  
98 of silicate vapor (Glass and Simonson, 2012). In rare cases, nickel-rich  
99 magnesiowüstite inclusions in spherules indicate condensation temperatures >2300 °C  
100 (Kyte and Bohor, 1995). However, no direct geological evidence has been reported  
101 for high temperatures in contiguous impact melt rock, quenched ejected melt droplets  
102 (tektites), or impact melt-bearing breccia (suevite) because no geothermometers  
103 applicable to impact melt are calibrated beyond ~2000 °C (Asimow and Ghiorso,  
104 1998; Cherniak et al., 2007; Hart and Davis, 1978; Kubaschewski, 1982; Lindsley and  
105 Andersen, 1983; Sack and Ghiorso, 1991; Taylor et al., 2015). Furthermore, models  
106 of impact melt formation often assume initial melt temperatures of ~1700 °C  
107 (Abramov et al., 2013; Onorato et al., 1978; Simonds, 1975), and in some cases allude  
108 to temperatures >2000 °C (Wünnemann et al., 2008).

109 In this study, we investigate the microstructure of zirconia ( $\text{ZrO}_2$ ) produced during  
110 the dissociation of zircon ( $\text{ZrSiO}_4$ ) in impact glass (quenched impact melt) sampled  
111 from the 28-km-diameter,  $37.83 \pm 0.05$  Ma Mistastin Lake impact structure in  
112 northern Labrador, Canada ( $55^{\circ}53'N$ ;  $63^{\circ}18'W$ ) (Marion and Sylvester, 2010;  
113 Sylvester et al., 2013). We present new thermometry based on reconstruction of the  
114 polymorphic transformation history of zirconia in a reaction rim around a zircon  
115 grain, an approach we refer to as phase heritage. Crystallographic relationships reveal  
116 the former presence of cubic zirconia, which definitively quantifies the minimum  
117 impact melt temperature to > 2370 °C, far exceeding commonly assumed impact melt  
118 temperatures. These results thus present fundamental new constraints on the evolution  
119 of impact melts, and influence our understanding of the evolution of early planetary  
120 crust, when impact rates were orders of magnitude higher than now, substantial

121 resurfacing by impact structures comparable to or larger in size than Mistastin Lake  
122 occurred (Marchi et al., 2014; Morbidelli et al., 2012), and all physical evidence of  
123 cratering was subsequently destroyed.

124

125 **2. METHODS AND APPROACH**

126 We characterised the zircon grain and surrounding area *in situ* on a polished thin  
127 section with cathodoluminescence (CL), backscattered electron (BSE) imaging and  
128 wavelength dispersive spectroscopy (WDS). We used electron backscatter diffraction  
129 (EBSD) to verify the mineral assemblage and characterize and quantify the  
130 crystallographic orientation and microstructure of zirconia ( $\text{ZrO}_2$ ) produced by the  
131 dissociation of zircon ( $\text{ZrSiO}_4$ ). See Supplementary File for further information about  
132 data acquisition and processing. Crystallographic relationships among zirconia grains  
133 preserve the phase transformation history, or phase heritage, of zirconia polymorphs  
134 (Cayron et al., 2010; Kerschhofer et al., 2000). Transformations between mineral  
135 phases commonly occur with systematic crystallographic orientation relationships  
136 (e.g., Pearce et al., 2013). New phases tend to nucleate at multiple sites in one of  
137 several symmetrically equivalent orientations. The new grain orientations in the  
138 resulting polycrystalline microstructure are related by systematic misorientations,  
139 described by specific angular rotations around specific crystallographic axes that are  
140 strictly controlled by the symmetry relationships between the old and new phases.  
141 Therefore, the original (parent) crystal orientation can be inferred by combining the  
142 misorientations between the new (daughter) phase grains and the transformation  
143 relationships, even when the old crystal has been completely transformed. This  
144 concept of orientation-based phase heritage has been used to identify the former  
145 presence of high-temperature zirconia polymorphs in manufactured ceramics (Cayron

146 et al., 2010; Chevalier et al., 2009) and kimberlites (Kerschhofer et al., 2000), and the  
147 high-pressure ZrSiO<sub>4</sub> polymorph reidite in impactites (Cavosie et al., 2016; Timms et  
148 al., 2017), and we use it here to investigate the thermal evolution of zircon entrained  
149 into impact melt.

150 The orientation relationships (OR) for the transformations from cubic-  
151 tetragonal-monoclinic zirconia due to cooling are well known from the literature on  
152 refractory ceramics, and measured orientations of low-temperature monoclinic  
153 zirconia (baddeleyite) can be used to reconstruct orientations of parent cubic grains,  
154 the highest temperature zirconia polymorph (Cayron, 2007; Cayron et al., 2010). This  
155 approach relies on  $\langle a \rangle_{\text{cubic}} \rightarrow \langle a \rangle_{\text{tetragonal}}$  or  $\langle c \rangle_{\text{tetragonal}}$ , generating up to three  
156 possible unique tetragonal orientations from a single cubic grain. During subsequent  
157 tetragonal-monoclinic transformation upon further cooling, the following  
158 transformation rules apply:

159  $\langle a \rangle_{\text{tetragonal}}$  or  $\langle c \rangle_{\text{tetragonal}} \rightarrow \langle b \rangle_{\text{monoclinic}}$

160 plus either

161  $\langle a \rangle_{\text{tetragonal}}$  or  $\langle c \rangle_{\text{tetragonal}} \rightarrow \langle a \rangle_{\text{monoclinic}}$  (type 1 OR after Cayron et al., 2010)

162 or

163  $\langle a \rangle_{\text{tetragonal}}$  or  $\langle c \rangle_{\text{tetragonal}} \rightarrow \langle c \rangle_{\text{monoclinic}}$  (type 2 OR after Cayron et al., 2010)

164 allowing for up to four unique orientation variants from each tetragonal identity  
165 (Chevalier et al., 2009), totalling twelve possible monoclinic variants from a single  
166 cubic identity (Cayron et al., 2010).

167 The Python-based software combination of ARPGE and GenOVA were used to  
168 perform a crystallographic orientation analysis of baddeleyite (monoclinic zirconia) to  
169 reconstruct evidence for precursor zirconia polymorphs following the approach  
170 applied by Cayron (2007) and Cayron et al. (2010) for ceramic applications. This

171 method automatically and objectively reconstructs ‘parent’ grains using known sets of  
172 orientation relationships among ‘daughter’ grains that have been established to result  
173 from phase transformations between specific zirconia polymorphs.

174

175 **3. RESULTS**

176 The zircon grain analyzed, MZRN-1, is in a glassy impact melt rock, exposed on  
177 top of the 80 m thick columnar jointed impact melt body at Discovery Hill near the  
178 crater wall (Marion and Sylvester, 2010). Mistastin impact melt rock composition  
179 ranges from ~53 to ~59 wt % SiO<sub>2</sub> and is interpreted to reflect mixing of different  
180 proportions of various crystalline igneous target rocks (Marion and Sylvester, 2010).

181 Field relations indicate the melt ponded at Earth’s surface at ambient atmospheric  
182 pressure (Marion and Sylvester, 2010). The zircon records a pre-impact U-Pb  
183 crystallization age of 1403 +/- 10 Ma (2 $\sigma$  error, n = 3, MSWD = 0.85), consistent with  
184 derivation from Mesoproterozoic bedrock in the region (Zanetti, 2015). Other phases  
185 in the matrix include ballen-textured silica, which is also characteristic of high-  
186 temperature (>1200 °C) impact melt (Ferrière et al., 2009).

187 Zircon grain MZRN-1 has an oscillatory zoned core in cathodoluminescence (CL)  
188 that is truncated by a ~2 µm wide, bright CL rim that contains zirconia particles with  
189 interspersed sub-µm silicate melt inclusions (Fig. 1). The grain is cut by irregular  
190 fractures and contains two voids, all filled with glassy silicate impact melt (Fig. 1B).  
191 The zircon does not preserve any diagnostic shock-deformation microstructures, such  
192 as twins or the high-pressure ZrSiO<sub>4</sub> polymorph reidite (Timms et al., 2017)  
193 (Supplementary File item 2). Minor lattice misorientations (<5°) are associated with  
194 rigid block rotation across brittle fractures (Supplementary File item 2), but in general  
195 the core of the zircon appears undeformed.

196       The core is surrounded by a ~40 µm thick corona of vermicular ZrO<sub>2</sub> crystals  
197       ranging from ~0.5 to ~5 µm across and up to ~20 µm long (Fig. 1). Elongate crystals  
198       tend to be aligned at high angles to the zircon core, and form domains of  
199       morphologically similar clusters up to ~50 µm wide. Individual crystals generally do  
200       not impinge on each other, and are separated by glassy silicate impact melt similar in  
201       composition to the surrounding crystal-free impact melt (Fig. 1). ZrO<sub>2</sub> contains lower  
202       trace elements abundances (e.g., ~350 ppm Ti, ~275 ppm Th, ~90 ppm Y) than zircon  
203       (e.g., ~900 ppm Ti, ~600 ppm Th, ~1900 ppm Y) (Supplementary File item 4). Most  
204       (>99%) of the ZrO<sub>2</sub> grains index as baddeleyite (monoclinic ZrO<sub>2</sub>). No crystalline  
205       SiO<sub>2</sub> phases were detected (Figs 1, 2).

206           Baddeleyite grains are commonly twinned and have a wide range of  
207       crystallographic orientations (Fig. 2A, C). Morphologically distinct clusters of  
208       baddeleyite grains preserve up to twelve unique crystallographic orientation variants  
209       with a systematic relationships among them (Fig. 2D, Supplementary File item 5).  
210       Groups of grains are crystallographically orientated approximately orthogonal to one  
211       another, and systematic deviations from orthogonality of ~20° form cross-shaped  
212       patterns on pole figures of <100> (Fig. 2D).

213           The occurrence of spatially-clustered baddeleyite grains with these distinctive  
214       patterns of twelve orientations uniquely identifies the former presence and orientation  
215       of a precursor cubic zirconia polymorph from the two-stage transformation from  
216       cubic to tetragonal to monoclinic zirconia (Fig. 2B, E) (Cayron et al., 2010). Initial  
217       processing of the EBSD maps of MZRN-1 involved dilation of daughter monoclinic  
218       zirconia grains by an iterative nearest neighbour extrapolation routine so that they  
219       impinge on one another (Supplementary File Fig. S3). This was required in order to  
220       perform neighbour-pair disorientation analysis within the ARPGE software.

221 Observed peaks we identified in disorientation analysis of MZRN-1 at 90°,  
222 115°, and 180° are consistent with cubic-monoclinic transformation twinning (Fig. 3).  
223 We identified symmetry operators for disorientations between daughter monoclinic  
224 zirconia grains at these specific angles in crystal reference frame (Fig. 3).

225 We used the software GenOVA to develop a list of theoretical symmetry  
226 operators for the cubic-monoclinic transformation according to OR type 1 or type 2  
227 OR (Cayron et al., 2010) to reconstruct the parent cubic zirconia grains (Figs 2, 3).  
228 Parent grain reconstruction involved a quadruplet search method with a minimum of  
229 five daughter grains per parent grain (Cayron et al., 2010). The disorientations we  
230 observed (angles +axes) fit very well with rotations by the operators expected from  
231 OR type 2, within an angular tolerance range of <5 ° (Fig. 3). We conclude that OR  
232 type 2 is the OR in the samples: there is no evidence that OR type 1 is present. The  
233 observed spread of the orientations (Fig. 3) is not due to a mixture of OR types 1 and  
234 2, and may be due to limited crystal-plasticity in the dynamic impact melt  
235 environment. We used 15-20° as the tolerance angle for the parent cubic grain  
236 reconstruction (Fig. 2B, D, Supplementary File Fig. S5).

237 Reconstruction of parent cubic zirconia grains from baddeleyite electron  
238 backscatter diffraction (EBSD) data shows that the dissociation corona was once  
239 comprised of large (~5 to ~50 µm across) domains of vermicular single crystals of  
240 cubic zirconia (Fig. 2B). The morphology of the cubic parent grains indicates that the  
241 vermicular ZrO<sub>2</sub> grains formed a connected 3D network that broadly pseudomorphed  
242 the original rim of the zircon grain. A second zircon from the same sample (MZRN-2)  
243 displays the same partial dissociation texture and also successfully yields  
244 reconstructed parent cubic zirconia grains (Supplementary File items 3 and 5),  
245 indicating the same cubic zirconia phase heritage as MZRN-1.

246

247 **4. DISCUSSION**

248 The microstructural and geochronological data, field relations, and thermodynamic  
249 constraints all support the interpretation that the zircon grains described here record  
250 an extreme high-temperature history at low-pressure conditions. The Mesoproterozoic  
251 U-Pb age and preserved igneous zoning of MZRN-1 indicate that the grain is a  
252 xenocryst derived from one of the plutonic igneous target rocks. The complete lack of  
253 crystal-plastic deformation, shock-twins and high-pressure polymorphs are consistent  
254 with the grains not having been affected by high-pressure shock conditions. The  
255 grains were, however, entrained into impact melt in the dynamic crater environment  
256 after passage of the shock wave. The ZrO<sub>2</sub> coronae are therefore interpreted to have  
257 developed solely due to thermal effects from immersion in super-heated impact melt  
258 at ambient pressure. The inference of ambient pressure is also consistent with  
259 available thermodynamic constraints that indicate that the zircon dissociation reaction  
260 line has a steep Clapeyron slope (29 °C/bar) and therefore is inevitably a low-pressure  
261 process (Timms et al., 2017).

262 A minimum temperature achieved by the impact melt can be constrained via  
263 analysis of phases in the zirconia corona, and we use the ZrO<sub>2</sub>-SiO<sub>2</sub> binary phase  
264 diagram at 1 atm to establish the temperature of zircon dissociation (Fig. 4) (Kaiser et  
265 al., 2008). The presence of a thick zirconia corona clearly records the melt achieving  
266 high enough temperature for zircon dissociation to proceed (Fig. 4) (Kaiser et al.,  
267 2008). Phase equilibria predict that zircon dissociates to tetragonal zirconia and  
268 cristobalite at 1673 °C (Fig. 4). The presence of silicate melt and absence of  
269 cristobalite indicates the sample exceeded 1687 °C; liquid silica released by  
270 dissociation would have dissolved into surrounding melt.

271 Clusters of morphologically distinct zirconia crystals that contain up to twelve  
272 unique orientations with systematic misorientations are consistent with a two-stage  
273 solid-state transformation from cubic zirconia via tetragonal zirconia. The former  
274 presence of cubic zirconia requires that the zircon, and therefore the melt that  
275 surrounds it, reached a minimum temperature of 2370 °C, but did not exceed 2700 °C  
276 above which cubic zirconia would have melted (Fig. 4), and thus prevented  
277 crystallographic phase heritage relations from being preserved. Upon cooling below  
278 2370 °C, the displacive transformation of cubic to tetragonal zirconia results in up to  
279 three distinct orientations of tetragonal zirconia whereby  $(001)_{\text{tetragonal}}$  is parallel to a  
280  $\{100\}_{\text{cubic}}$  (Heuer, 1987). Further cooling below ~1180 °C results in a stable  
281 assemblage of baddeleyite + zircon (Fig. 4). The tetragonal –monoclinic zirconia  
282 transformation is martensitic, and resulted in four orientation variants of baddeleyite  
283 from each tetragonal parent (Cayron et al., 2010). Thus, the transformation of higher-  
284 temperature cubic ZrO<sub>2</sub> polymorphs in the corona to the low temperature ZrO<sub>2</sub>  
285 polymorph baddeleyite produced systematic orientation relationships among  
286 baddeleyite grains. These are interpreted as transformation (reversion) twins formed  
287 upon cooling (Supplementary File item 6).

288 Even though cubic zirconia is the highest temperature ZrO<sub>2</sub> polymorph, its  
289 stability is pressure-dependant, such that it is stabilized at lower temperature with  
290 increasing pressure (Bouvier et al., 2000; Kerschhofer et al., 2000) (Supplementary  
291 File item 7). However, zirconia in the Mistastin sample is a product of zircon  
292 dissociation, which ostensibly is a low-pressure phenomenon that occurs at post-  
293 shock conditions (see Timms et al., 2017 for further details and thermodynamic  
294 calculations). All available field evidence, theoretical constraints, microstructural and  
295 geochemical data supports zircon dissociation in an unconfined impact melt at

296 ambient, post-shock pressure. Therefore, the transformations among zirconia  
297 polymorphs that took place after zircon dissociation necessarily also occurred in the  
298 melt sheet at post-shock low-pressure conditions, which constrains the stability of  
299 cubic zirconia to extremely high temperatures. Impurities, such as REE and Y, can  
300 also stabilize cubic zirconia at lower temperatures (Swab, 2001) (Supplementary File  
301 item 7). However, the levels of trace elements present in the ZrO<sub>2</sub> are much less than  
302 0.1 wt% (Supplementary File item 4), and so the effects of impurities on the cubic to  
303 tetragonal zirconia transformation temperature were negligible (Swab, 2001)  
304 (Supplementary File item 7). Therefore, the former presence of cubic zirconia in the  
305 Mistastin Lake zircon indicates that the impact melt temperature was in excess of  
306 2370 °C (Fig. 3) (Kaiser et al., 2008).

307 Our study documents forensic geological evidence for a minimum temperature  
308 of 2370 °C in a sample of impact melt. This temperature constraint is the highest  
309 recorded by any rock on Earth's surface, and is several hundred degrees higher than  
310 previous estimates for average temperatures of superheated impact melts (Fig. 5)  
311 (Onorato et al., 1978; Simonds, 1975; Wünnemann et al., 2008).

312 The ability to assess the phase heritage of ZrO<sub>2</sub> polymorphs by  
313 crystallographic orientation relationships is a new methodology for establishing the  
314 thermal evolution of superheated melt. Textural evidence of zircon dissociation and  
315 associated zirconia has been reported in numerous examples, including impact melt  
316 rocks, tektites, lunar melt breccias, kimberlites, as well as anthropogenic settings,  
317 such as slag from metal smelting and nuclear test sites (Timms et al., 2017). Given  
318 that zircon is a common accessory mineral in many rocks, this approach can be used  
319 to assess minimum temperatures of zircon-bearing impact melt rocks from Earth,

320 Moon, and meteorites. The broader use of determining phase heritage can potentially  
321 be applied to a wide range of polymorphic transformations in geological systems.

322 Our results begin to constrain the conditions under which zircon and its U-Pb  
323 isotope systematics may survive during impact cratering. During high temperature  
324 and low shock pressure events, zircon is capable of surviving extreme conditions  
325 without necessarily recording microstructural evidence typical of shock deformation  
326 (Erickson et al., 2017; Timms et al., 2017; Timms et al., 2012). The Mesoproterozoic  
327 U-Pb protolith zircon age preserved in the ~38 Ma impact melt investigated here  
328 indicates that, despite being entrained in impact melt in excess of 2370 °C, the  
329 residence time at high temperature was not sufficient to cause significant Pb-loss from  
330 the core of the grain by volume diffusion (Cherniak and Watson, 2001). The glassy  
331 matrix is further evidence that duration at extremely high temperatures was short and  
332 the sample quenched rapidly. However, increased melt volumes in larger impact  
333 structures with slower cooling rates may lead to complete digestion of zircon  
334 xenocrysts, erasing mineralogical evidence of superheated temperatures. Early during  
335 Earth's history when impact rates were high, digestion of zircon by superheated  
336 impact melt may have contributed to the paucity of Hadean zircon preserved in the  
337 geological record (Marchi et al., 2014).

338 The formation of >2370 °C impact melt during moderate sized impacts has  
339 several significant, global-scale consequences during the early evolution of the Earth  
340 and Moon when the frequency and magnitude of meteorite flux was much higher  
341 (Gomes et al., 2005; Kring and Cohen, 2002). The extent of compositional  
342 modification by differential volatile loss of major and trace elements from impact  
343 melts by selective vaporization at temperatures >2370 °C has not been fully  
344 investigated, yet are sufficiently high to cause significant volatilization of many key

345 oxide constituents of silicate melts (Lamoreaux et al., 1987). Preferential  
346 volatilization of SiO<sub>2</sub>, Na<sub>2</sub>O and K<sub>2</sub>O components from hot impact melt bodies could  
347 provide a mechanism for localised densification of residual crust, potentially with  
348 geodynamic consequences. Molecular species essential for terrestrial life, such as  
349 H<sub>2</sub>O and CO<sub>2</sub>, are potentially lost due to volatilization of superheated impact melt.  
350 High-temperature impact-melting could locally form dry, rigid, refractory residual  
351 crust, resistant to subsequent melting. Moderate sized impacts, comparable to  
352 Mistastin, are predicted to have completely resurfaced the Earth within the first tens  
353 of millions of years after the Moon-forming event, irrespective of the choice of the  
354 suggested bombardment models (Marchi et al., 2014). Therefore, the resultant net  
355 effect of hypothesised high impact flux early during Earth's history, combined with  
356 commonplace high temperatures of impact melts, potentially led to widespread  
357 depletion of hydrous minerals from the lithosphere and therefore governed  
358 rheological properties of early crust. Further investigation of the geodynamic  
359 consequences of this phenomenon for early Earth would require accurate predictions  
360 of impact melt temperatures, flux, size, impact angles, and velocity distributions of  
361 impactors, and appropriate scaling relationships for melt volumes linked to these  
362 variables. Furthermore, long term effects would need to account for rehydration  
363 effects by interaction with Earth's hydrosphere. Our results on high temperature melt  
364 and attendant volatilization processes have implications for the early evolution of the  
365 geochemical reservoirs and multi-element isotope systematics of planetary bodies  
366 (Drake and Righter, 2002) and that extremely high melt temperatures can be achieved  
367 even in moderate-sized impact events, and are not limited to giant, basin-forming  
368 impacts.

369

370

371 **FIGURE CAPTIONS**

372 Figure 1. A. Panchromatic cathodoluminescence (CL) image of zircon (Zrn)  
373 xenocryst with a corona of baddeleyite (Bdy) and silicate glass. B. Backscattered  
374 electron image of area shown by white box in A showing the zircon-baddeleyite-glass  
375 interface.

376

377 Figure 2. A. Electron backscatter diffraction map coloured for measured  
378 crystallographic orientation of baddeleyite. Zircon is shown in grayscale based on  
379 EBSD pattern quality (band contrast). B. Map showing crystallographic orientations  
380 of reconstructed parent cubic zirconia grains. Inverse pole figure (IPF) orientation  
381 colour scheme. C-F. Pole figures showing measured ‘daughter’ baddeleyite  
382 orientations and reconstructed cubic zirconia ‘parent’ orientations. Lower hemisphere  
383 plots in sample x-y-z reference frame.

384

385 Figure 3. Results from orientation analysis of daughter monoclinic zirconia grains of  
386 MZRN-1 using the software ARPGE (Cayron, 2007; Cayron et al., 2010). A.  
387 Histogram of disorientation angles between daughter monoclinic grains. B.  
388 Disorientation symmetry operator statistics for adjacent daughter grains for operators  
389 for the type 2 orientation relationship (OR) of Cayron et al. (2010). C. Pole figures  
390 showing the distribution of 90°, 115°, and 180° disorientation axes between daughter  
391 grains in the crystal reference frame.

392

393 Figure 4. T-X phase diagram for  $\text{ZrO}_2$ - $\text{ZrSiO}_4$ - $\text{SiO}_2$  system modified after Kaiser et  
394 al. (2008). Tridymite stability after Swamy et al. (1994). Crosshatch shows zircon-  
395 bearing fields, stipple shows melt-bearing fields.

396

397 Figure 5. Ranges for various types of high-temperature thermometers available in  
398 geosciences. Cubic zirconia (blue bar) related to zircon dissociation (blue arrow) is  
399 the thermometer for highest known temperatures (numbers on left indicate  
400 references). Vaporization field is approximate and based on rates for common species  
401 present in silicate melts (Lamoreaux et al., 1987).

402

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412

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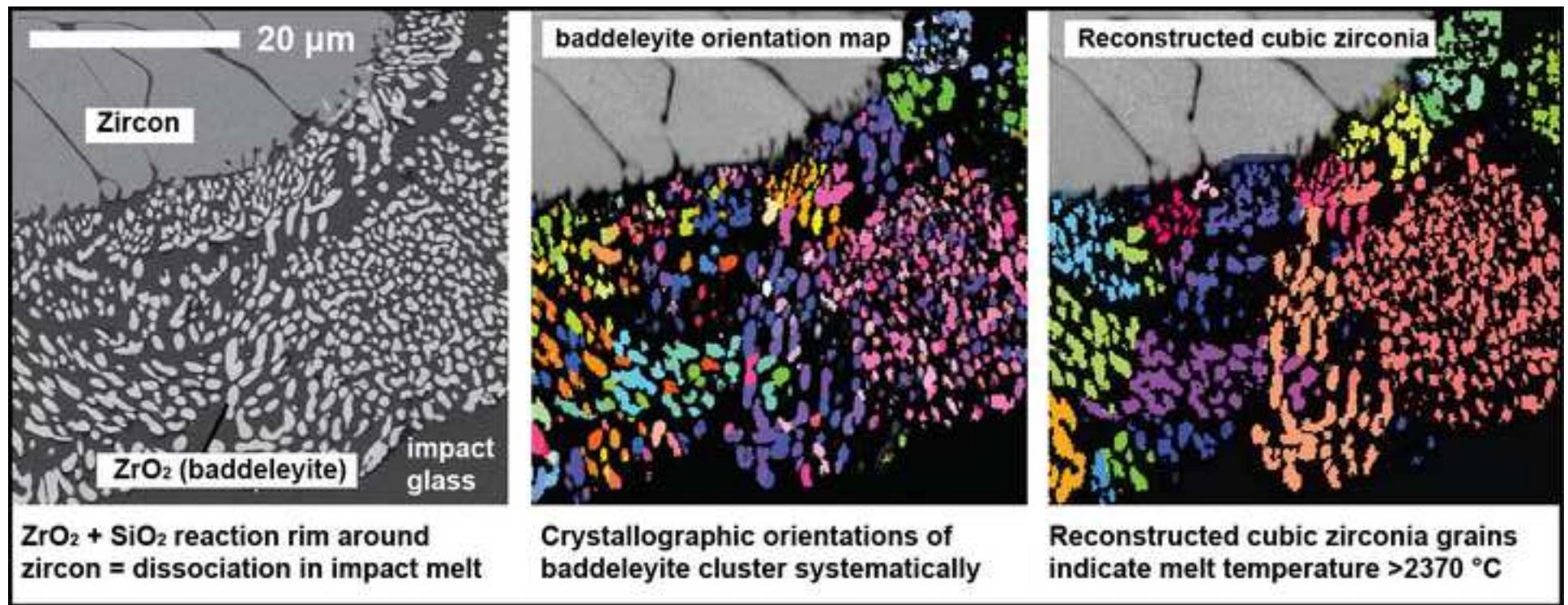
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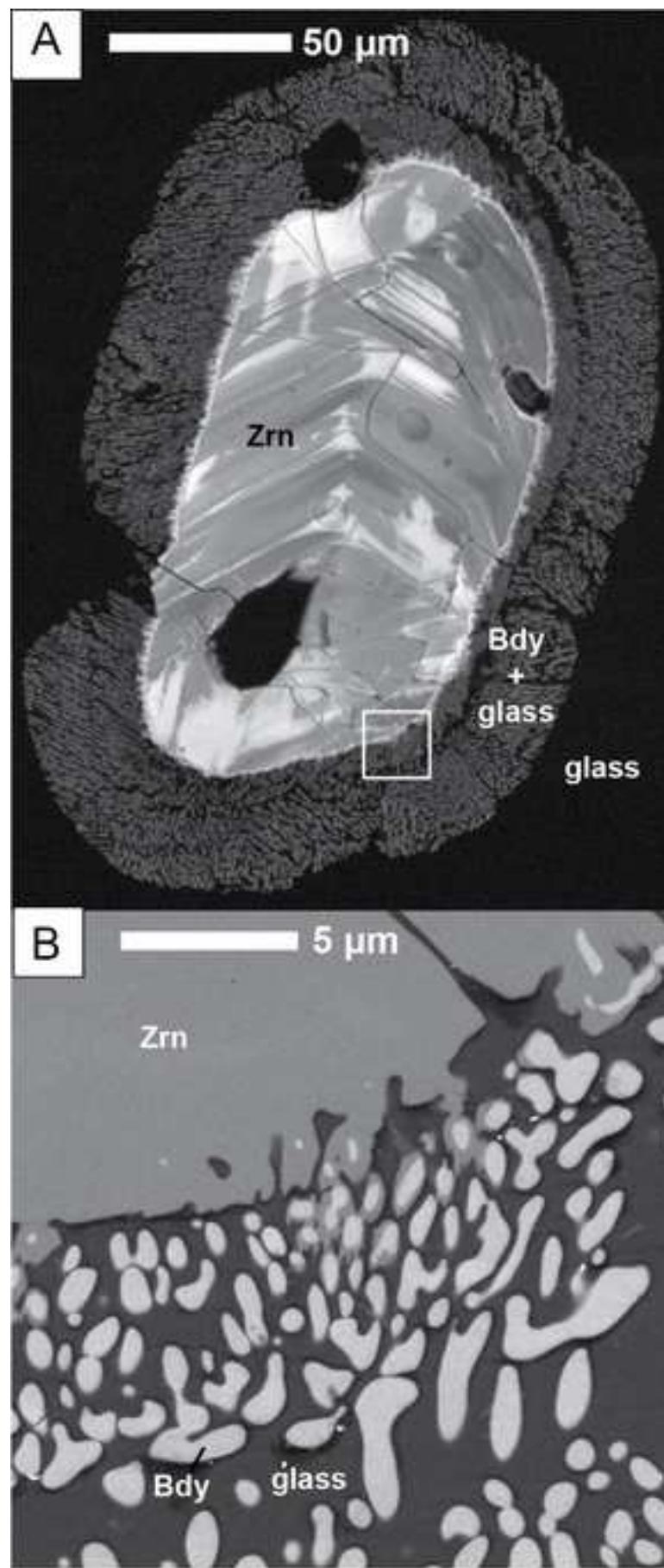
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**Figure 1**

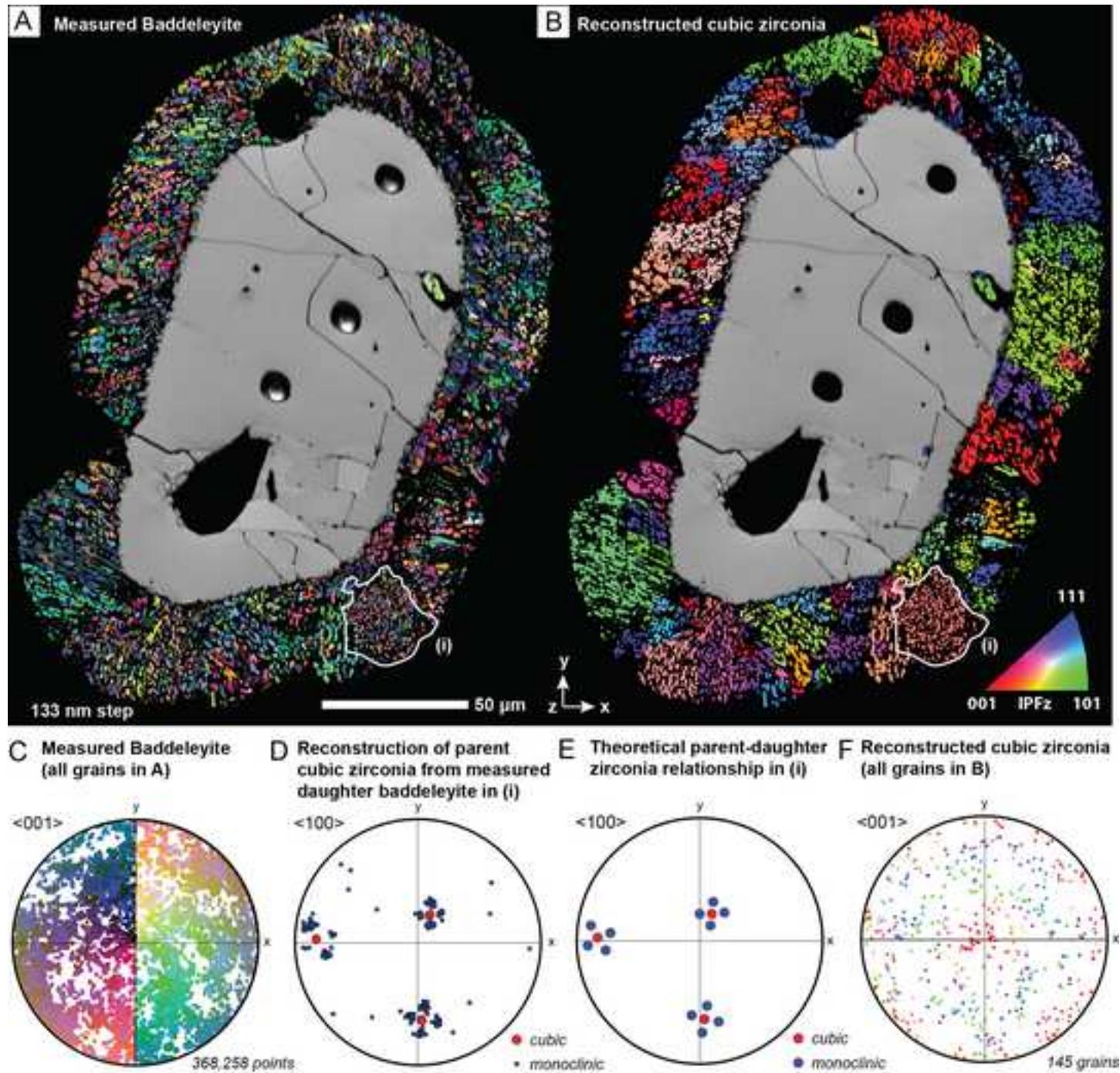
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*Timms et al. Figure 1*

**Figure 2**

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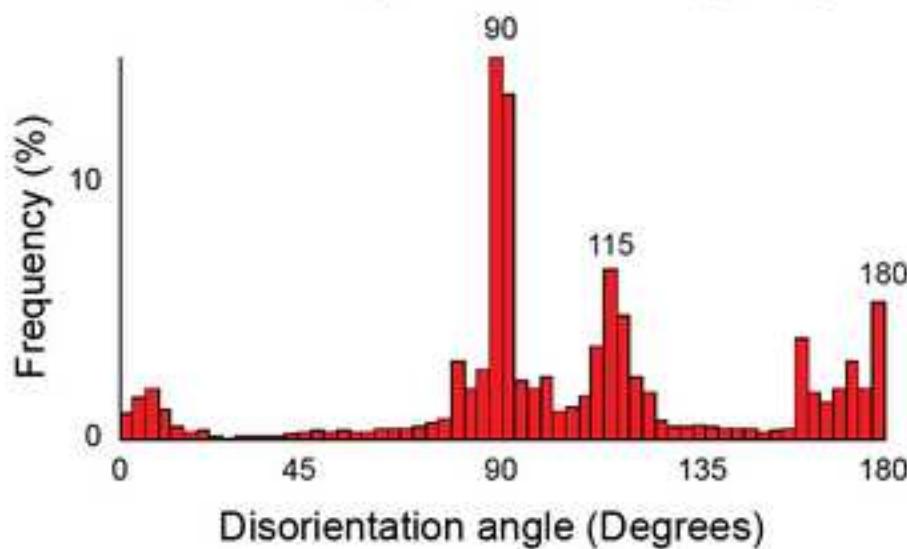


Timms et al. Figure 2

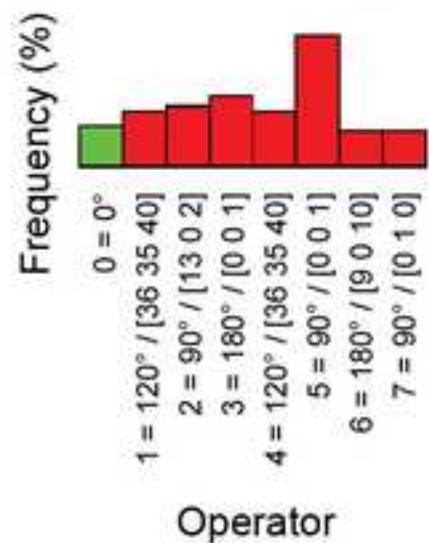
**Figure 3**

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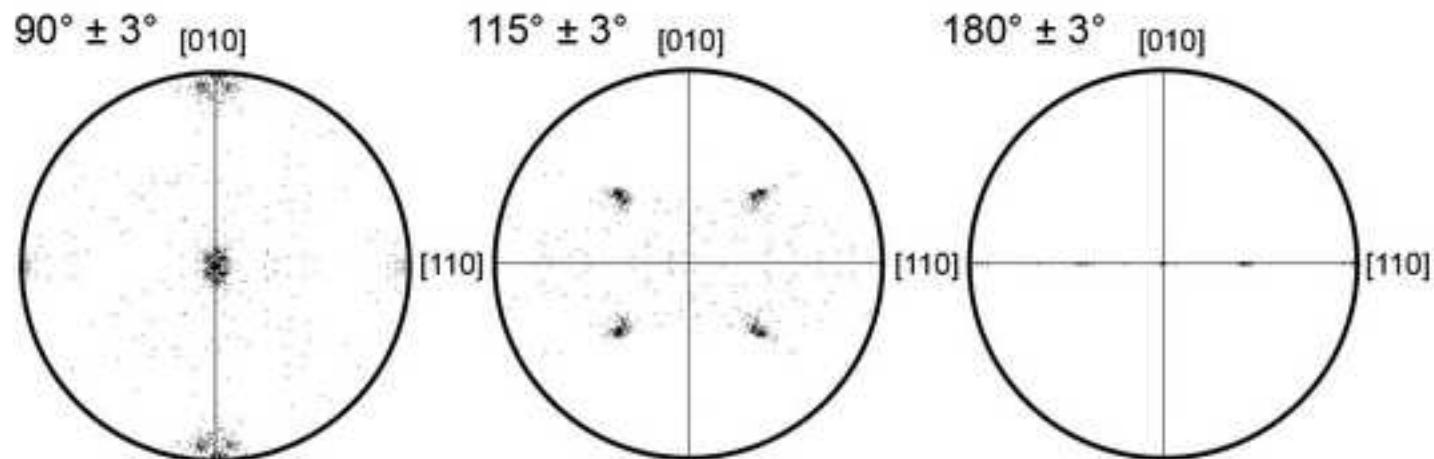
**A. Disorientation angles between daughter grains**



**B. Operator statistics  
(minimum rotation for OR type 2)**



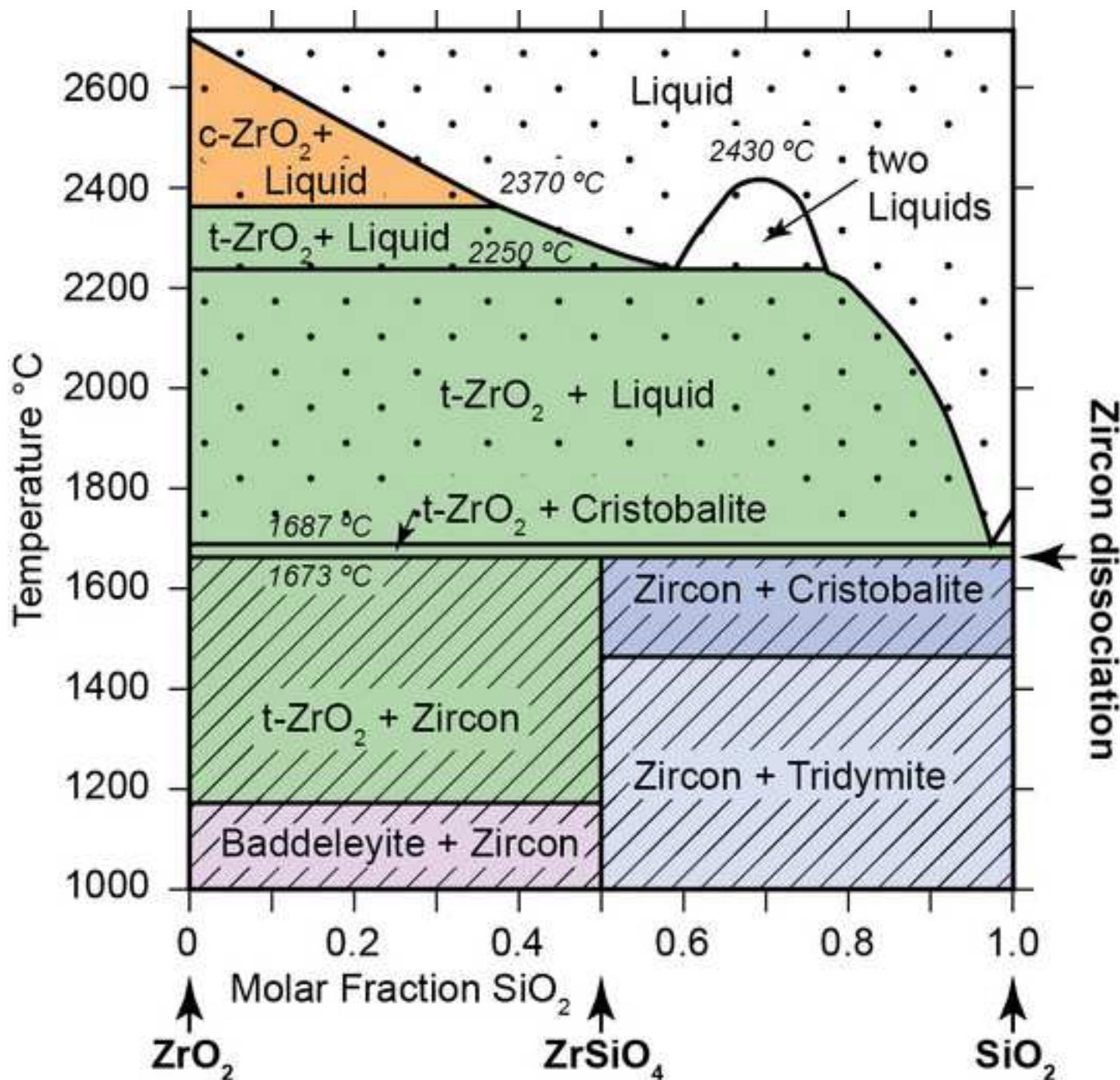
**C. Distribution of disorientation axes between daughter grains  
(crystal reference frame)**



*Timms et al. Figure 3*

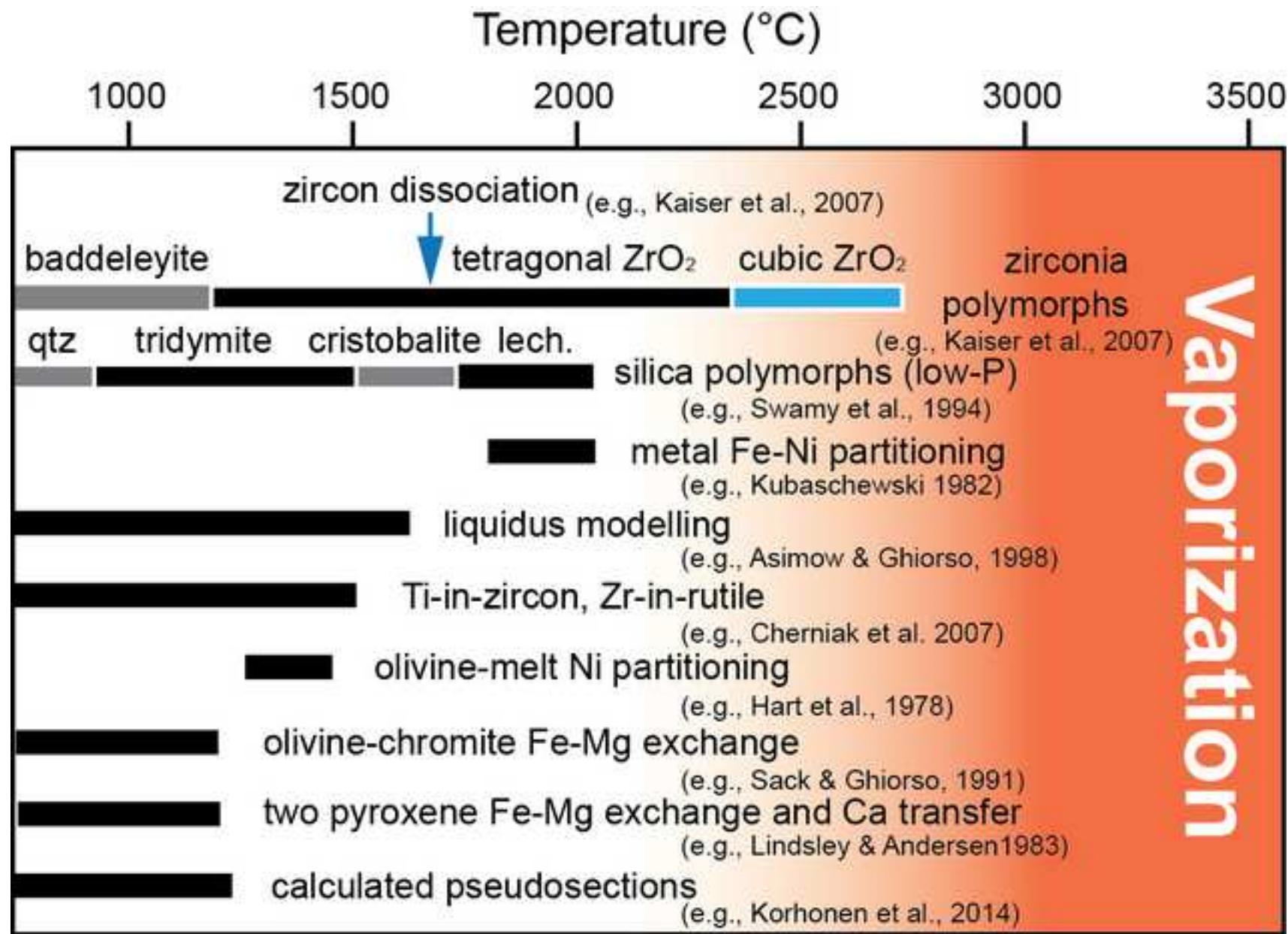
Figure 4

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Timms et al. Figure 4

Figure 5

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# Supplementary File

## Cubic zirconia in >2370 °C impact melt records Earth's hottest crust

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## **Item 1. Sample preparation and scanning electron microscopy (SEM) data collection methods and settings**

### **Sample preparation**

We polished the petrographic sections of the sample with 1 µm diamond paste prior to mounting regions of interest that include the zircon grains into an epoxy disk. The sample was given a further chemical-mechanical polish for EBSD analysis using colloidal silica in NaOH for three hours on a Buehler Vibromet II polisher. A thin coat of carbon was applied to mitigate charging during electron microscopy yet permit EBSD patterns to be acquired.

### **Cathodoluminescence and backscatter electron imaging**

At Curtin University, we performed scanning electron microscopy (SEM) using a Tescan MIRA3 field emission (FE-)SEM and atomic number contrast imaging with a pole piece backscatter detector, an accelerating voltage of 5 kV, a beam current of 22 nA, and a working distance of 15 mm. Moreover, we did cathodoluminescence imaging at 15 kV acceleration voltage and 15 mm working distance.

### **Electron backscatter diffraction (EBSD) mapping**

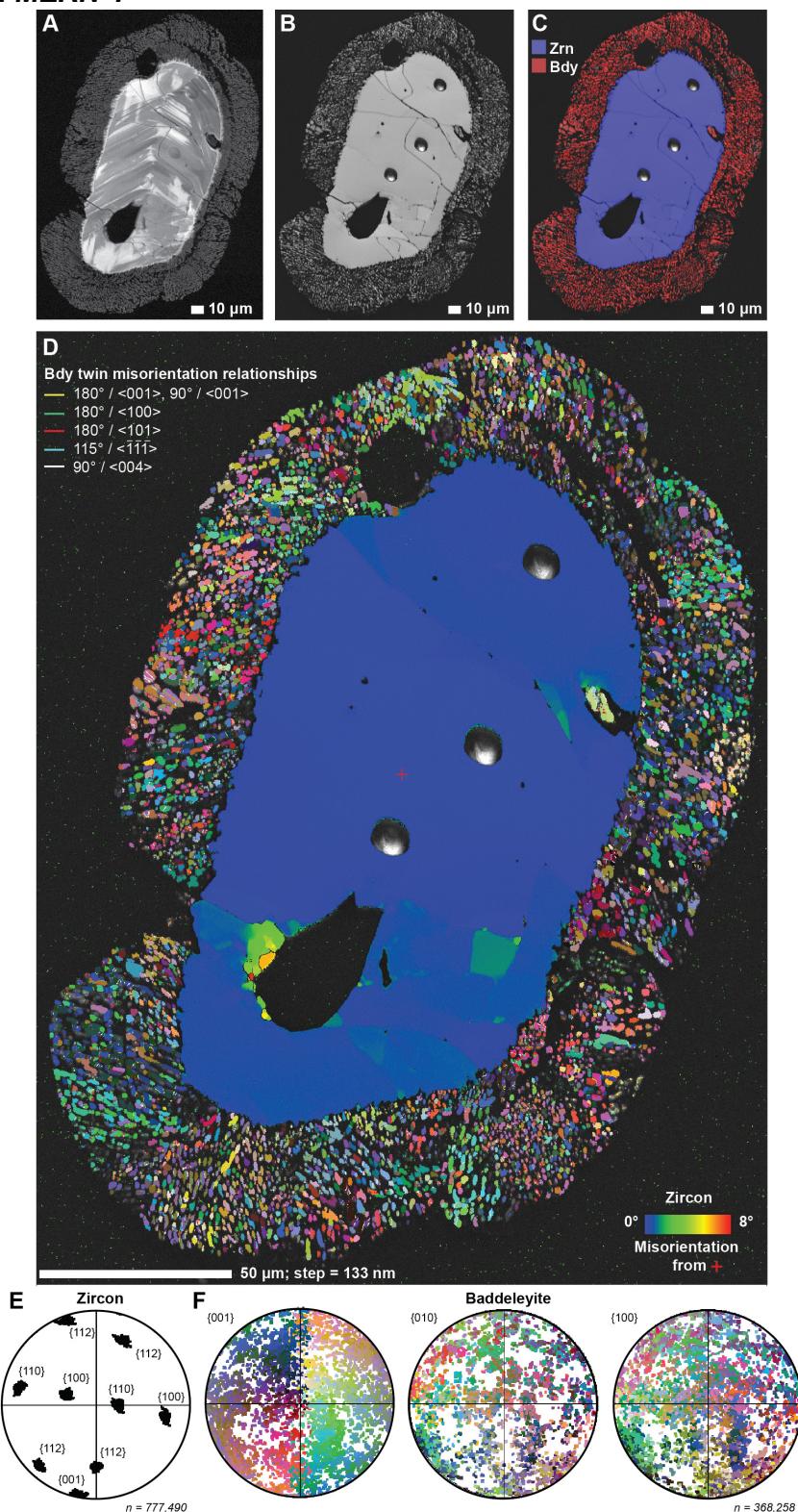
We did orientation mapping via EBSD, using a Tescan MIRA3 FE-SEM with an Oxford Instruments AZTEC EDS/EBSD acquisition system at Curtin University. Operation conditions were optimised for EBSD and include a stage tilt of 70° around a horizontal axis and beam acceleration voltage of 20 kV (Prior et al., 1999). We processed EBSD patterns using 4 x 4 pixel binning and indexed using match units for zircon, cubic, orthorhombic, tetragonal and monoclinic ZrO<sub>2</sub> (Table S1). However, only zircon and baddeleyite could be indexed. The mean ‘mean angular deviation’ for zircon and baddeleyite indexing solutions was 0.41° and 0.50°, respectively.

We processed the EBSD data using Oxford Instruments Channel 5.12 software. Maps were noise reduced by removing isolated erroneous data points ('wildspike' correction) followed by a zero solution extrapolation to six nearest neighbours. Visual inspection shows that no significant artefacts were generated during noise reduction. We generated thematic maps using the Tango module of Channel 5, and include phase, EBSD pattern quality (band contrast), and crystallographic orientation maps (Figs S1 and S2). Pole figures were generated using the Mambo module of Channel 5 (Figs S1 and S2).

**Table S1.** Scanning electron microscopy settings and electron backscatter diffraction analysis acquisition and processing parameters.

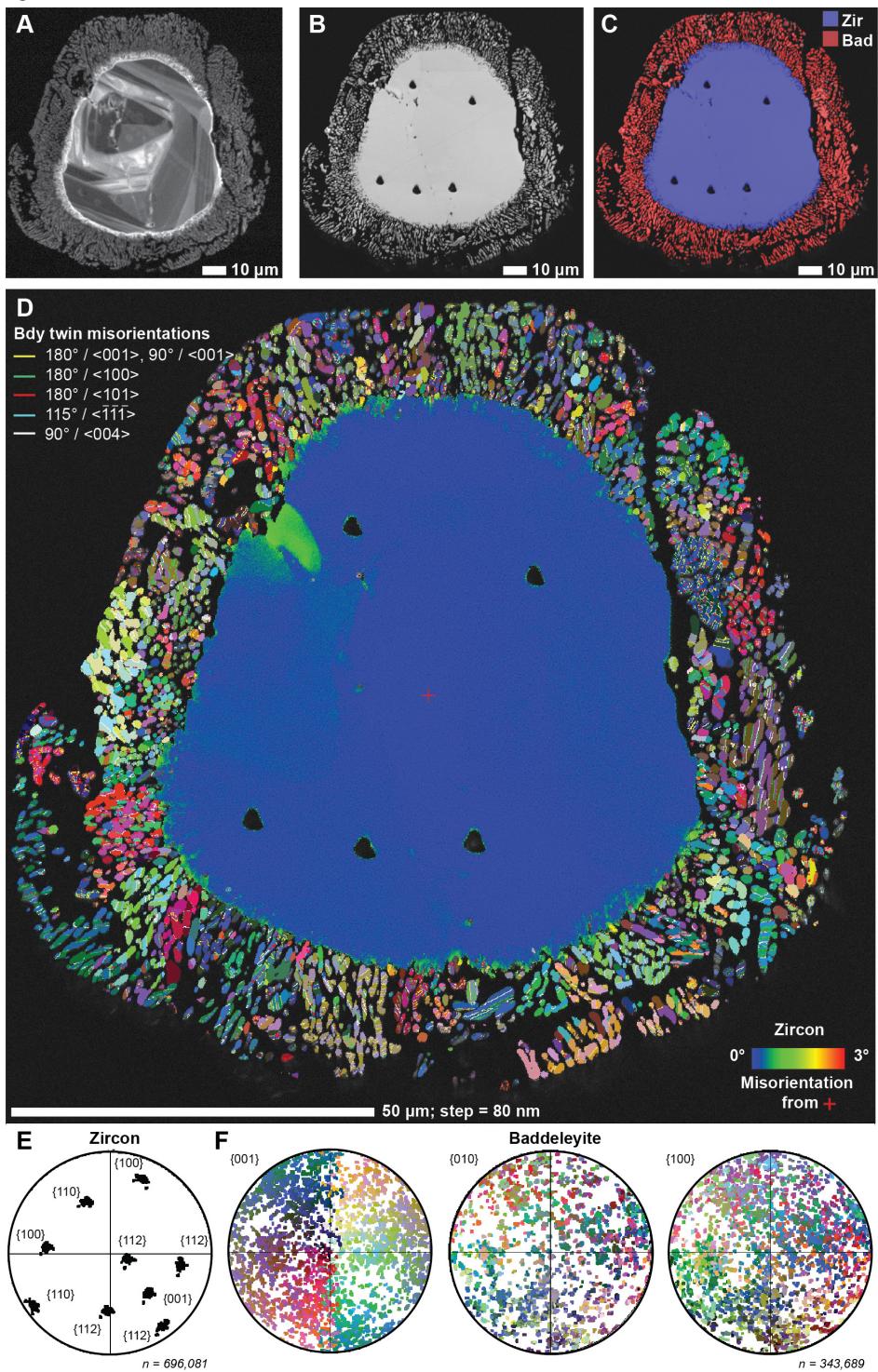
<b>SEM</b>				
Make/model	Tescan Mira3 FEG-SEM			
EBSD acquisition system	Oxford Instruments Aztec / Nordlys EBSD Detector			
EBSD Processing software	Oxford Instruments Channel 5.10			
Acceleration Voltage (kV)	20			
Working Distance (mm)	~20.5			
Tilt	70°			
<b>EBSD match units</b>				
Zircon	Zircon5260, 1 atm (Hazen and Finger, 1979)			
Reidite	Reidite632, 0.69 GPa (Farnan et al., 2003)			
Baddeleyite (monoclinic ZrO <sub>2</sub> )	(Bondars et al., 1995), (Hill and Cranswick, 1994)			
Tetragonal ZrO <sub>2</sub>	(Teufer, 1962)			
Cubic ZrO <sub>2</sub>	ICSD card 53998 (Böhm, 1925)			
Orthorhombic ZrO <sub>2</sub>	ICSD card 77716			
Quartz	'Quartznew', HKL database (Sands, 1969)			
Cristobalite	(Downs and Palmer, 1994)			
Coesite	(Kirfe et al., 1979)			
<b>EBSP Acquisition, Indexing and Processing</b>				
<b>Sample Location</b>				
	<b>Discovery Hill, Mistastin Lake, Canada</b>			
Grain ID	MZRN-1	MZRN-2		
Figures	2, S1, S3, S5	S2, S4, S6, S7		
EBSP Acquisition Speed (Hz)	40	40		
EBSP Background (frames)	64	64		
EBSP Binning	4 x 4	4 x 4		
EBSP Gain	High	High		
Hough resolution	60	60		
Band detection (min / max)	6 / 8	6 / 8		
Mean angular deviation (zircon)	<1°	<1°		
Mean angular deviation (reidite)	n/a	n/a		
Mean angular deviation (baddeleyite)	<1°	<1°		
Map step size (nm)	133	80		
Map size (X steps / Y steps)	1428 / 1862	1495 / 1435		
<b>EBSD noise reduction routine</b>				
Wildspike correction	Yes	Yes		
Nearest neighbour zero solution extrapolation	6	6		

**Item 2. Summary of images and electron backscatter diffraction (EBSD) data from MZRN-1**



**Figure S1.** MZRN-1 A. CL image. B. EBSD pattern quality (band contrast) map. C. Phase map from EBSD data. D. EBSD map showing ~8° cumulative misorientations recorded by MZRN-1, solely attributed to brittle fracture block rotation. No crystal-plastic microstructures, twinning or reidite were detected. Baddeleyite in corona coloured for orientation using All Euler scheme. E. Pole figure of principal planes of zircon. F. Pole figures of principal planes of baddeleyite. E and F are lower hemisphere equal area projections in the map x-y-z reference frame.

**Item 3. Summary of images and electron backscatter diffraction (EBSD) data from MZRN-2**



**Figure S2.** MZRN-2 A. CL image. B. EBSD pattern quality (band contrast) map. C. Phase map from EBSD data. D. EBSD map showing <3° cumulative misorientation recorded by MZRN-2, solely attributed to brittle fracture block rotation. A minor amount of crystal-plastic strain is preserved near the dissociation interface. No other crystal-plastic microstructures, twinning or reidite indicative of shock metamorphism were detected. Baddeleyite in corona coloured for orientation using All Euler scheme. E. Pole figure of principal planes of zircon. F. Pole figures of principal planes of baddeleyite. E and F are lower hemisphere equal area projections in the map x-y-z reference frame.

#### **Item 4. Electron microprobe microanalysis (EPMA) methods and settings**

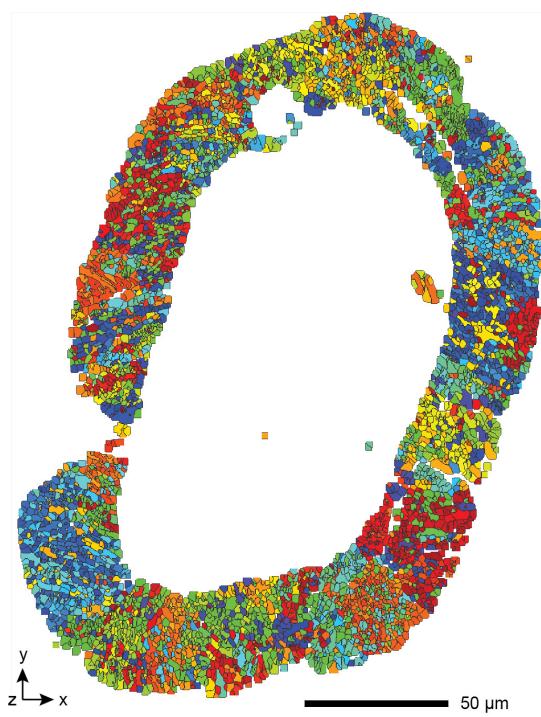
We measured the concentration of selected trace elements of zircon and zirconia using a JEOL JXA-8200 electron probe microanalyser at the Department of Earth and Planetary Sciences of Washington University in St Louis, USA. This instrument is equipped with 5 wavelength-dispersive spectrometers, and an e2v silicon-drift energy-dispersive X-ray spectrometer. The WDS spot analyses included Ca, Fe, P, Zr, Si, Hf, Ti, Dy, Er, Th, Y. We used a time-dependent intensity correction routine for the WDS analyses to quantify element concentrations using the Probe for EPMA software package (Table S2).

**Table S2.** Selected major and trace element concentrations in zirconia, zircon and impact melt glass from time dependent intensity WDS analysis, after Zanetti (2015). Concentrations in parts per million (ppm). Bdl = below detection limit.

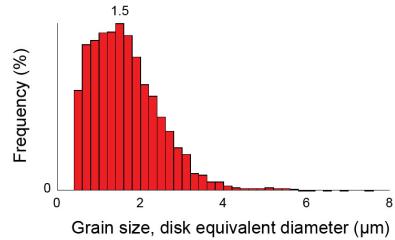
Acc. V (kV) Probe current (nA)	150	150	50	50
Element	Zircon core (bright CL)	Zircon core (dark CL)	Zircon rim	Baddeleyite
Fe	250-475	360-800	840-1000	~300
Ti	150-250	170-330	200-400	~350
Th	~35	200-340	20?	~275
Y	Bdl	1000-1780	670	~90
Dy	Bdl	60?	10-220	Bdl
Er	Bdl	Bdl	≤ 150	Bdl
P	280	300-400	-	~640

## Item 5. Reconstruction of parent cubic zirconia grains from daughter monoclinic zirconia crystallographic orientation data

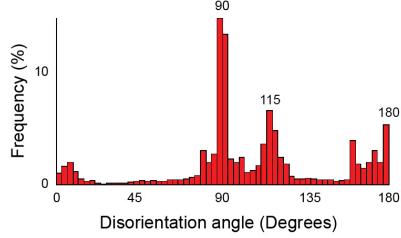
A. Daughter monoclinic grains (dilated), all Euler colours



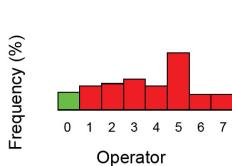
B. Grain size histogram, daughter monoclinic grains



C. Histogram of disorientation angles between daughter grains



D. Operator statistics

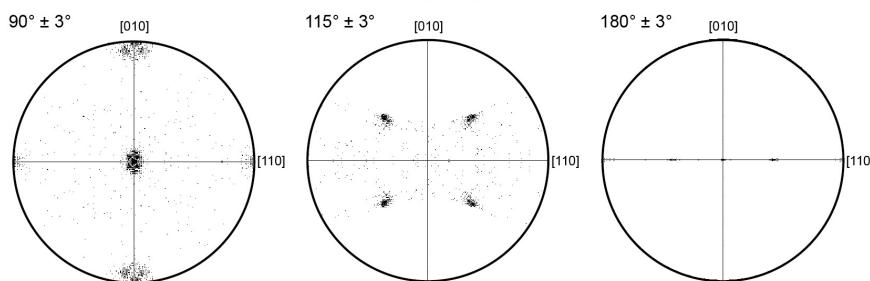


E. List of operators

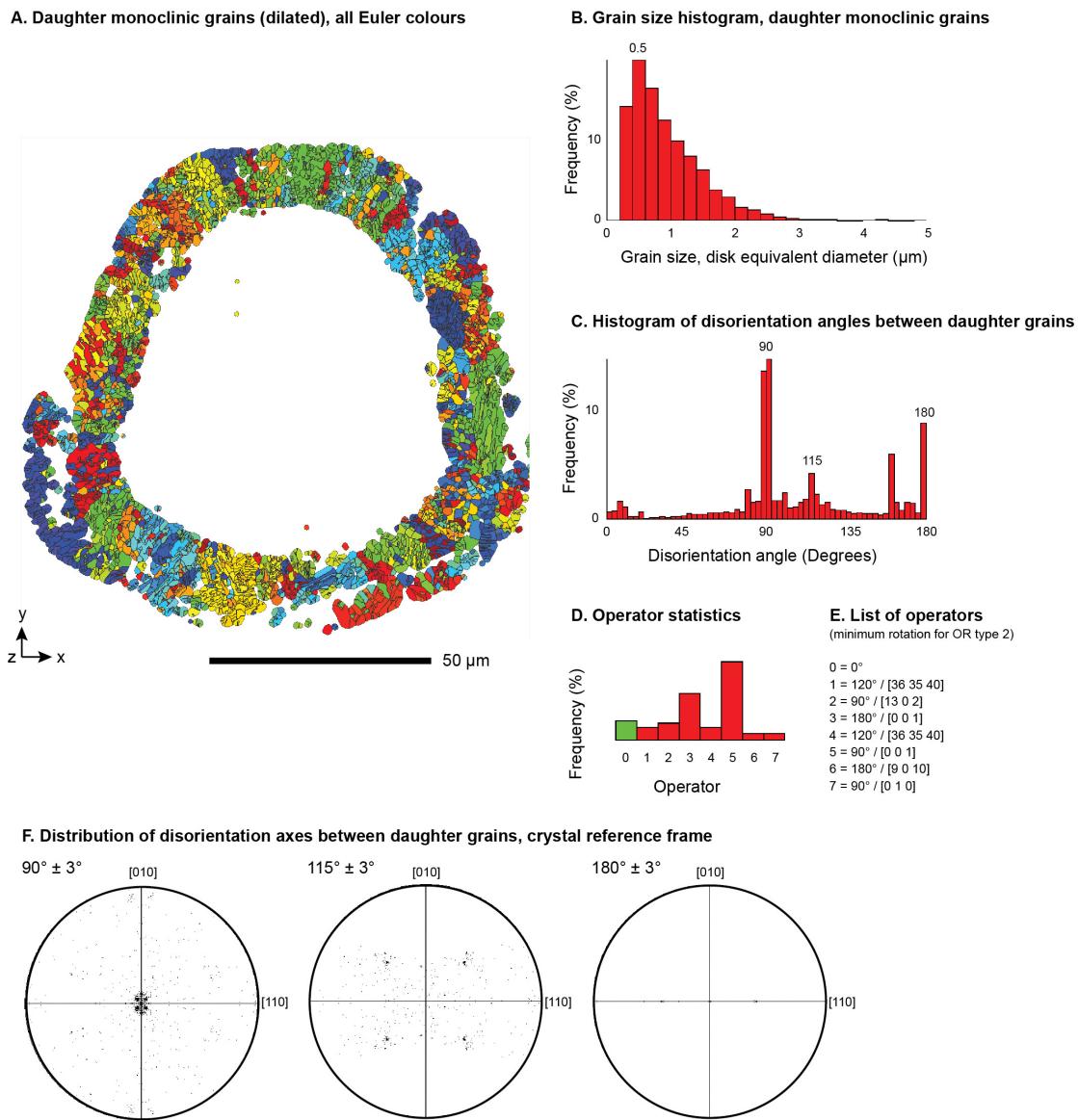
(minimum rotation for OR type 2)

0	= 0°
1	= 120° / [36 35 40]
2	= 90° / [13 0 2]
3	= 180° / [0 0 1]
4	= 120° / [36 35 40]
5	= 90° / [0 0 1]
6	= 180° / [9 0 10]
7	= 90° / [0 1 0]

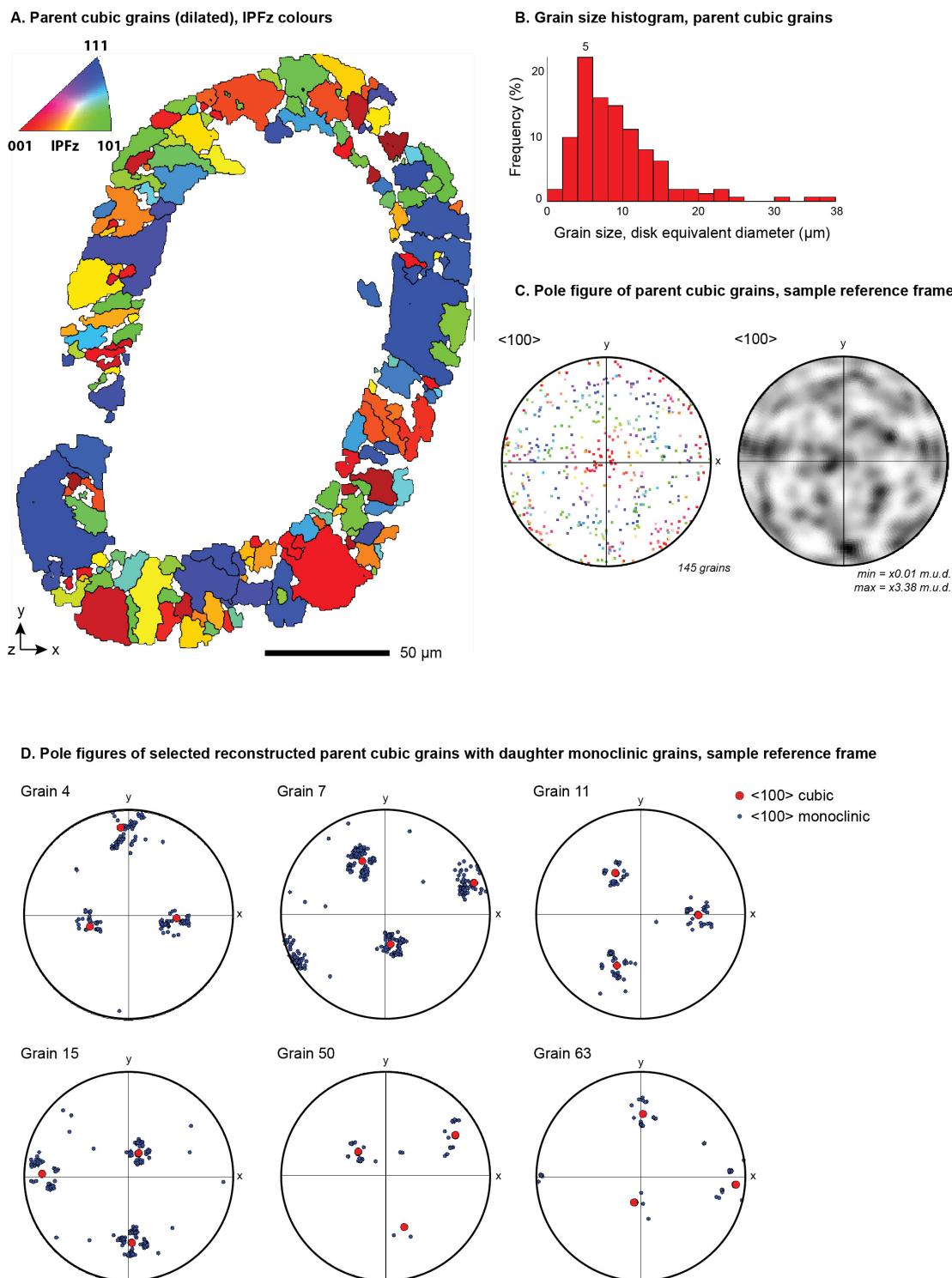
F. Distribution of disorientation axes between daughter grains, crystal reference frame



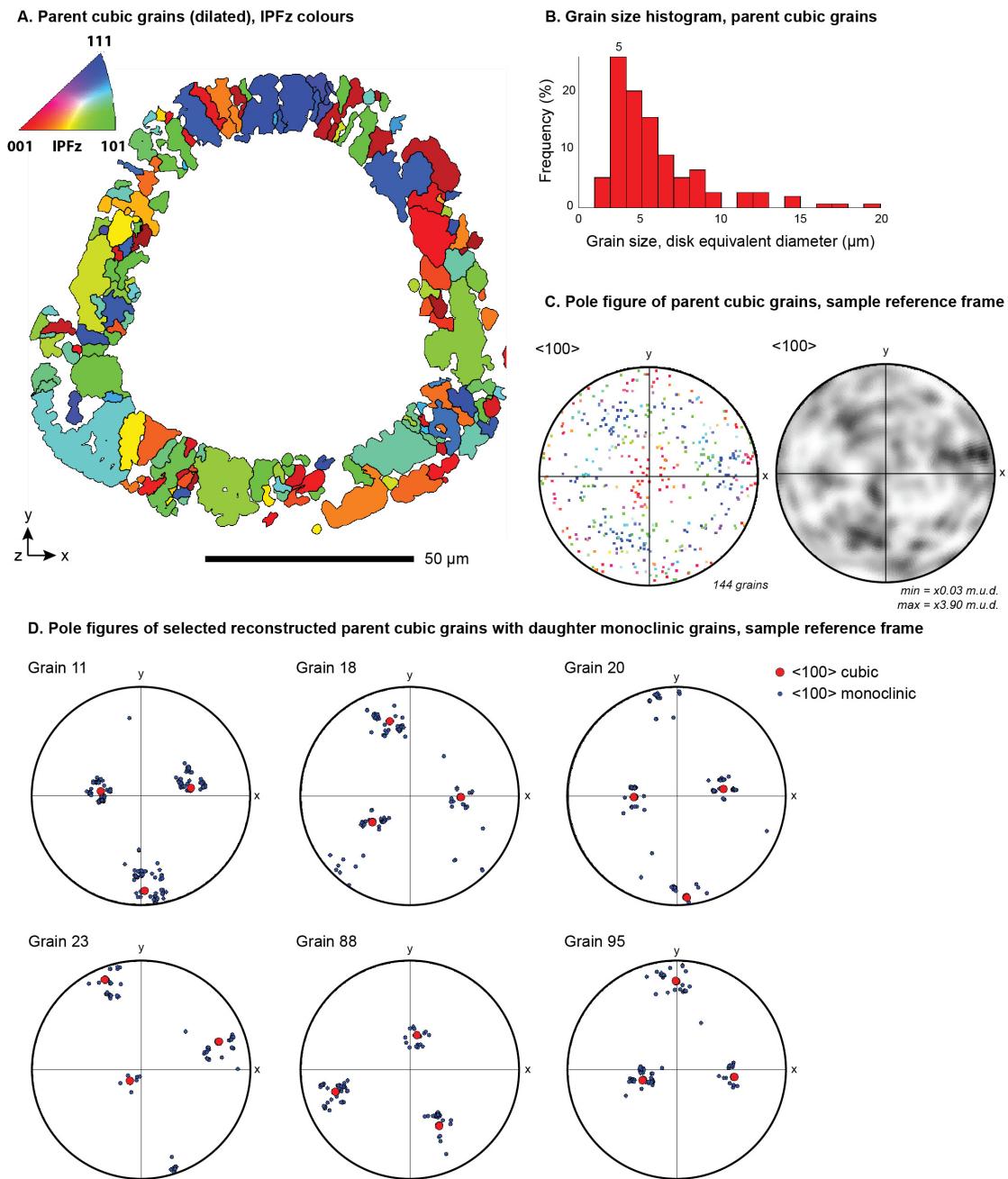
**Figure S3.** Results from orientation analysis of daughter monoclinic zirconia grains of MZRN-1 using the software ARPGE (Cayron, 2007; Cayron et al., 2010). A. Map showing the daughter monoclinic grains after grain the dilation procedure. All Euler colours represent different crystallographic orientations. B. Grain size histogram of daughter monoclinic grains shown in A. C. Histogram of disorientation angles between daughter monoclinic grains shown in A. D. Disorientation symmetry operator statistics for adjacent grains in A. E. List of disorientation operators for the type 2 orientation relationship (OR) of Cayron et al. (2010). F. Pole figures showing the distribution of 90°, 115°, and 180° disorientation axes between daughter grains in the crystal reference frame.



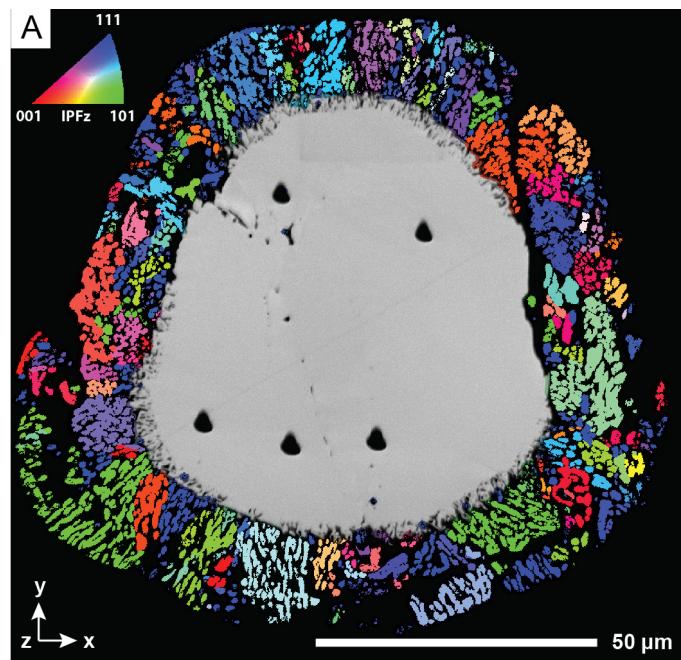
**Figure S4.** Results from orientation analysis of daughter monoclinic zirconia grains of MZRN-2 using the software ARPGE (Cayron, 2007; Cayron et al., 2010). A. Map showing the daughter monoclinic grains after grain dilation procedure. All Euler colours represent different crystallographic orientations. B. Grain size histogram of daughter monoclinic grains shown in A. C. Histogram of disorientation angles between daughter monoclinic grains shown in A. D. Disorientation symmetry operator statistics for adjacent grains in A. E. List of disorientation operators for the type 2 orientation relationship (OR) of Cayron et al. (2010). F. Pole figures showing the distribution of 90°, 115°, and 180° disorientation axes between daughter grains in the crystal reference frame.



**Figure S5.** Reconstructed dilated parent cubic zirconia grains for MZRN-1 via the ARPGE software using type 2 OR after Cayron et al. (2010) (see Fig. S3). A. Map of parent cubic zirconia grains, all Euler orientation colour scheme. B. B. Grain size histogram of daughter monoclinic grains shown in A. C. Pole figure of  $<100>$  of the reconstructed cubic zirconia grains in the sample x-y-z reference frame. Left: colour scheme as in A; right: contoured at one point per grain. D.  $<100>$  Pole figures of selected reconstructed parent cubic zirconia grains (red) with daughter monoclinic zirconia grains (blue). Pole figures are lower hemisphere equal area projections in the map x-y-z reference frame.



**Figure S6.** Reconstructed dilated parent cubic zirconia grains for MZRN-2 via the ARPGE software using type 2 OR after Cayron et al. (2010) (see Fig. S4). A. Map of parent cubic zirconia grains, all Euler orientation colour scheme. B. Grain size histogram of daughter monoclinic grains shown in A. C. Pole figure of <100> of the reconstructed cubic zirconia grains in the sample x-y-z reference frame. Left: colour scheme as in A; right: contoured at one point per grain. D. <100> Pole figures of selected reconstructed parent cubic zirconia grains (red) with daughter monoclinic zirconia grains (blue). Pole figures are lower hemisphere equal area projections in the map x-y-z reference frame.



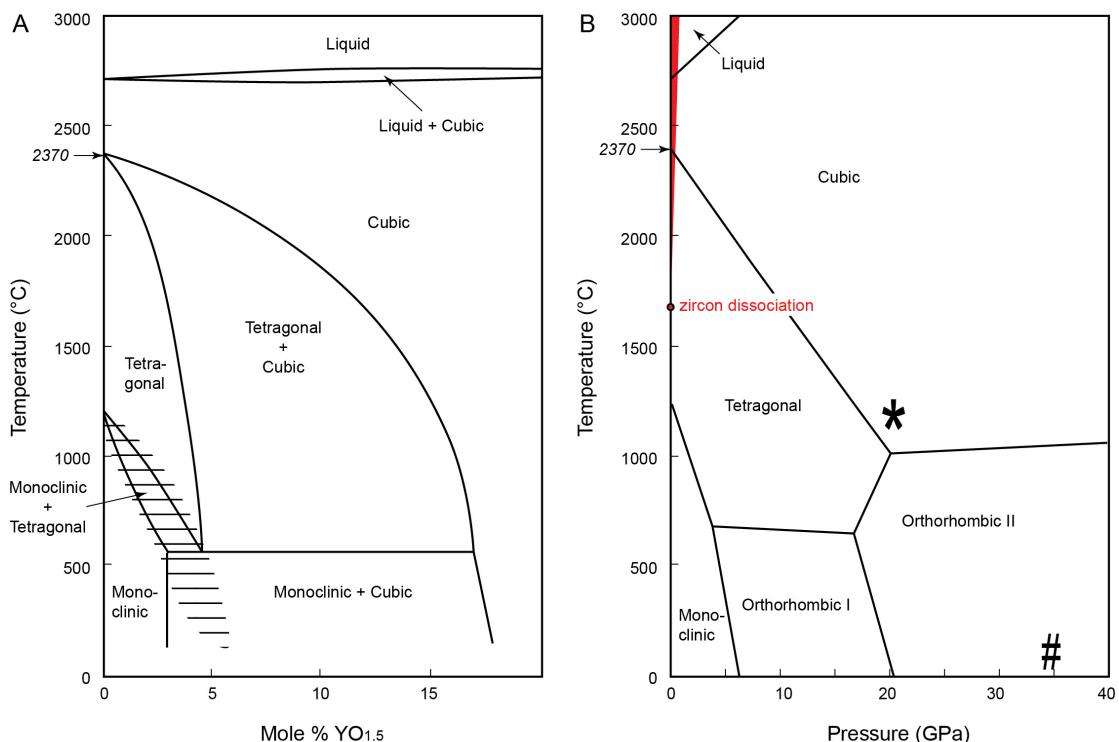
**Figure S7.** A. Map showing fully reconstructed parent cubic zirconia grains from MZRN-2. IPFz colour scheme. EBSD pattern quality (band contrast) for zircon shown in grayscale, and non-indexed points are black.

### Item 6. Transformation twin relationships in the baddeleyite corona

**Table S3.** Transformation twin relationships in the baddeleyite corona.  $K_1$  = composition plane;  $\eta_1$  = shear direction.

Misorientation relationship	$K_1$	$\eta_1$	Type
$180^\circ <001>$	(100)	[001]	Compound
$180^\circ <100>$	(001)	[100]	Compound
$180^\circ <101>$	( $\bar{1}0\bar{1}$ )	[ $10\bar{1}$ ]	Compound
$90^\circ <004>$		[ $\bar{1}\bar{1}0$ ]	Type 2
$90^\circ <001>$	( $1\bar{1}0$ )		Type 1
$90^\circ <001>$		[ $0\bar{1}\bar{1}$ ]	Type 2
$85^\circ <100>$	( $01\bar{1}$ )		Type 1
$115^\circ <\bar{1}\bar{1}1>$	?	?	unknown

## Item 7. Effects of impurities and pressure on zirconia polymorph stability



**Figure S8.** A. The effects of YO<sub>1.5</sub> impurities on the stability of ZrO<sub>2</sub> polymorphs, after Swab (2001). B. The effects of pressure on the stability of ZrO<sub>2</sub> polymorphs, after Bouvier et al. (2000) and references therein. The red dot and red shaded area show the conditions at which zircon dissociates to zirconia + silica after Kaiser et al. (2008) and Timms et al. (2017), respectively.

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