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2 Cubic zirconia in >2370 °C impact melt records Earth's hottest crust

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Cubic zirconia in >2370 °C impact melt records the hottest crust on Earth

Highlights

- Zircon has partially dissociated in impact melt rock from a Canadian impact crater
- Former presence of cubic ZrO₂ is crystallographically encoded in reaction rims
- Cubic zirconia required >2370 °C melt, which is hottest recorded on Earth's surface
- Such superheated melt susceptible to devolatilization resulting in dry rigid crust
- Potential global effects for crustal evolution during bombardment of early Earth

23 **ABSTRACT**

24 Bolide impacts influence primordial evolution of planetary bodies because they can
25 cause instantaneous melting and vaporization of both crust and impactors.
26 Temperatures reached by impact-generated silicate melts are unknown because
27 meteorite impacts are ephemeral, and established mineral and rock thermometers have
28 limited temperature ranges. Consequently, impact melt temperatures in global
29 bombardment models of the early Earth and Moon are poorly constrained, and may
30 not accurately predict the survival, stabilization, geochemical evolution and cooling
31 of early crustal materials. Here we show geological evidence for the transformation of
32 zircon to cubic zirconia plus silica in impact melt from the 28 km diameter Mistastin
33 Lake crater, Canada, which requires super-heating in excess of 2370 °C. This new
34 temperature determination is the highest recorded from any crustal rock. Our phase
35 heritage approach extends the thermometry range for impact melts by several hundred
36 degrees, more closely bridging the gap between nature and theory. Profusion of >
37 2370 °C superheated impact melt during high intensity bombardment of Hadean Earth
38 likely facilitated consumption of early-formed crustal rocks and minerals, widespread
39 volatilization of various species, including hydrates, and formation of dry, rigid,
40 refractory crust.

41

42 **KEYWORDS:** cubic zirconia, zircon, phase transformation, impact melt, planetary
43 evolution, early Earth

44

45

46 **1. INTRODUCTION**

47 Shock wave propagation during hypervelocity impact can melt and vaporize
48 both the impactor and target rocks (Melosh, 1989). The immediate, post-impact,
49 thermal pulse far exceeds both ultra-high-temperature metamorphism in tectonic
50 settings (Heisinger and Head, 2006; Korhonen et al., 2014) and the liquidus
51 temperatures of target rocks, and is therefore important for the evolution of planet and
52 asteroid surfaces. The heat associated with impacts had a profound significance on the
53 early Earth, and has been shown to have affected processes in Earth's core (Arkani-
54 Hamed and Ghods, 2011; Monteux et al., 2015; Sleep, 2016), mantle (Watters et al.,
55 2009), crust and atmosphere (Marchi et al., 2016; Marchi et al., 2014; Marchi et al.,
56 2013; Melosh, 2008).

57 Volatile depletion and isotope fractionation of Earth, Moon and other planetary
58 bodies are attributed to giant basin-forming impacts (Albarede et al., 2013; Day and
59 Moynier, 2014; Moynier et al., 2010; Pringle et al., 2014). Vaporization of target and
60 impactor material during impacts is temperature dependant. Preferential mass-
61 dependant volatile loss and isotope fractionation from impact melts can therefore
62 locally influence subsequent remelting and rheological behaviour, and so giant
63 impacts and/or the cumulative effects of intense periods of impact bombardment have
64 important geodynamic consequences (Albarede, 2009). These effects, in turn, have
65 implications for the evolution and habitability of Earth's surface environment
66 (Abramov and Mojzsis, 2009; Ryder, 2002), and, by extension, the habitability of
67 extraterrestrial planetary bodies and exoplanets.

68 Initial thermal properties of impact melt also control their ability to digest pre-
69 existing solid material and entrained debris (Onorato et al., 1978), including metals
70 and sulphides essential for forming impact-generated economic ore deposits (Keays

71 and Lightfoot, 2004), their cooling rate (Onorato et al., 1978), and consequently the
72 preservation of all pre-existing material and high-P phases formed during the passage
73 of the shock wave (Tschauer et al., 2014). The highest temperature superheated
74 melts (i.e., above the liquidus - the equilibrium temperature for complete melting) and
75 vapor are produced during shock-release melting and frictional melting, whereas
76 exhumation-related decompression melting is limited to sub-liquidus temperatures
77 (Ahrens and O'Keefe, 1971; French and Koeberl, 2010; Melosh, 1989; Melosh, 2005;
78 Riller et al., 2010; Spray, 1998).

79 The critical melting temperature of target rocks depends on their composition,
80 proximity to the impact site, initial conditions (Gibbons et al., 1975; Hörz et al., 2005;
81 Stöffler, 1971) and porosity (Kieffer et al., 1976; Wünnemann et al., 2008).

82 Vaporization temperature is taken as the highest vaporization temperature of oxides
83 present in the system; in real systems, incongruent vaporization may occur over a
84 range of temperatures (Ahrens and O'Keefe, 1971; Grieve et al., 1977; Lamoreaux et
85 al., 1987; Lou et al., 1985). However, the complex behaviour of silicates has made
86 shock-related heating and, thus, the magnitudes of impact melt superheating and
87 vaporization difficult to predict with any degree of certainty (Ahrens and O'Keefe,
88 1971). Accordingly, our understanding of temperatures achieved by impact melts
89 necessarily relies on empirical constraints from the rock record via established
90 geothermometers, which have inherently limited temperature ranges (Asimow and
91 Ghiorso, 1998; Cherniak et al., 2007; Hart and Davis, 1978; Kubaschewski, 1982;
92 Lindsley and Andersen, 1983; Sack and Ghiorso, 1991; Taylor et al., 2015). As there
93 are large ranges between liquidus and vaporization temperatures for most rock types,
94 maximum impact melt temperatures are largely unaccounted for, thus the effects of
95 superheated melt may be underappreciated in planetary surface studies.

96 In the rock record, glass spherules and some types of microtektites contained
97 within impact ejecta horizons provide *a priori* evidence for atmospheric condensation
98 of silicate vapor (Glass and Simonson, 2012). In rare cases, nickel-rich
99 magnesiowüstite inclusions in spherules indicate condensation temperatures $>2300\text{ }^{\circ}\text{C}$
100 (Kyte and Bohor, 1995). However, no direct geological evidence has been reported
101 for high temperatures in contiguous impact melt rock, quenched ejected melt droplets
102 (tektites), or impact melt-bearing breccia (suevite) because no geothermometers
103 applicable to impact melt are calibrated beyond $\sim 2000\text{ }^{\circ}\text{C}$ (Asimow and Ghiorso,
104 1998; Cherniak et al., 2007; Hart and Davis, 1978; Kubaschewski, 1982; Lindsley and
105 Andersen, 1983; Sack and Ghiorso, 1991; Taylor et al., 2015). Furthermore, models
106 of impact melt formation often assume initial melt temperatures of $\sim 1700\text{ }^{\circ}\text{C}$
107 (Abramov et al., 2013; Onorato et al., 1978; Simonds, 1975), and in some cases allude
108 to temperatures $>2000\text{ }^{\circ}\text{C}$ (Wünnemann et al., 2008).

109 In this study, we investigate the microstructure of zirconia (ZrO_2) produced during
110 the dissociation of zircon (ZrSiO_4) in impact glass (quenched impact melt) sampled
111 from the 28-km-diameter, $37.83 \pm 0.05\text{ Ma}$ Mistastin Lake impact structure in
112 northern Labrador, Canada ($55^{\circ}53'\text{N}$; $63^{\circ}18'\text{W}$) (Marion and Sylvester, 2010;
113 Sylvester et al., 2013). We present new thermometry based on reconstruction of the
114 polymorphic transformation history of zirconia in a reaction rim around a zircon
115 grain, an approach we refer to as phase heritage. Crystallographic relationships reveal
116 the former presence of cubic zirconia, which definitively quantifies the minimum
117 impact melt temperature to $> 2370\text{ }^{\circ}\text{C}$, far exceeding commonly assumed impact melt
118 temperatures. These results thus present fundamental new constraints on the evolution
119 of impact melts, and influence our understanding of the evolution of early planetary
120 crust, when impact rates were orders of magnitude higher than now, substantial

121 resurfacing by impact structures comparable to or larger in size than Mistastin Lake
122 occurred (Marchi et al., 2014; Morbidelli et al., 2012), and all physical evidence of
123 cratering was subsequently destroyed.

124

125 **2. METHODS AND APPROACH**

126 We characterised the zircon grain and surrounding area *in situ* on a polished thin
127 section with cathodoluminescence (CL), backscattered electron (BSE) imaging and
128 wavelength dispersive spectroscopy (WDS). We used electron backscatter diffraction
129 (EBSD) to verify the mineral assemblage and characterize and quantify the
130 crystallographic orientation and microstructure of zirconia (ZrO_2) produced by the
131 dissociation of zircon ($ZrSiO_4$). See Supplementary File for further information about
132 data acquisition and processing. Crystallographic relationships among zirconia grains
133 preserve the phase transformation history, or phase heritage, of zirconia polymorphs
134 (Cayron et al., 2010; Kerschhofer et al., 2000). Transformations between mineral
135 phases commonly occur with systematic crystallographic orientation relationships
136 (e.g., Pearce et al., 2013). New phases tend to nucleate at multiple sites in one of
137 several symmetrically equivalent orientations. The new grain orientations in the
138 resulting polycrystalline microstructure are related by systematic misorientations,
139 described by specific angular rotations around specific crystallographic axes that are
140 strictly controlled by the symmetry relationships between the old and new phases.
141 Therefore, the original (parent) crystal orientation can be inferred by combining the
142 misorientations between the new (daughter) phase grains and the transformation
143 relationships, even when the old crystal has been completely transformed. This
144 concept of orientation-based phase heritage has been used to identify the former
145 presence of high-temperature zirconia polymorphs in manufactured ceramics (Cayron

146 et al., 2010; Chevalier et al., 2009) and kimberlites (Kerschhofer et al., 2000), and the
147 high-pressure ZrSiO₄ polymorph reidite in impactites (Cavosie et al., 2016; Timms et
148 al., 2017), and we use it here to investigate the thermal evolution of zircon entrained
149 into impact melt.

150 The orientation relationships (OR) for the transformations from cubic-
151 tetragonal-monoclinic zirconia due to cooling are well known from the literature on
152 refractory ceramics, and measured orientations of low-temperature monoclinic
153 zirconia (baddeleyite) can be used to reconstruct orientations of parent cubic grains,
154 the highest temperature zirconia polymorph (Cayron, 2007; Cayron et al., 2010). This
155 approach relies on $\langle a \rangle_{\text{cubic}} \rightarrow \langle a \rangle_{\text{tetragonal}} \text{ OR } \langle c \rangle_{\text{tetragonal}}$, generating up to three
156 possible unique tetragonal orientations from a single cubic grain. During subsequent
157 tetragonal-monoclinic transformation upon further cooling, the following
158 transformation rules apply:

159 $\langle a \rangle_{\text{tetragonal}} \text{ OR } \langle c \rangle_{\text{tetragonal}} \rightarrow \langle b \rangle_{\text{monoclinic}}$

160 plus either

161 $\langle a \rangle_{\text{tetragonal}} \text{ OR } \langle c \rangle_{\text{tetragonal}} \rightarrow \langle a \rangle_{\text{monoclinic}}$ (type 1 OR after Cayron et al., 2010)

162 or

163 $\langle a \rangle_{\text{tetragonal}} \text{ OR } \langle c \rangle_{\text{tetragonal}} \rightarrow \langle c \rangle_{\text{monoclinic}}$ (type 2 OR after Cayron et al., 2010)

164 allowing for up to four unique orientation variants from each tetragonal identity
165 (Chevalier et al., 2009), totalling twelve possible monoclinic variants from a single
166 cubic identity (Cayron et al., 2010).

167 The Python-based software combination of ARPGE and GenOVa were used to
168 perform a crystallographic orientation analysis of baddeleyite (monoclinic zirconia) to
169 reconstruct evidence for precursor zirconia polymorphs following the approach
170 applied by Cayron (2007) and Cayron et al. (2010) for ceramic applications. This

171 method automatically and objectively reconstructs ‘parent’ grains using known sets of
172 orientation relationships among ‘daughter’ grains that have been established to result
173 from phase transformations between specific zirconia polymorphs.

174

175 **3. RESULTS**

176 The zircon grain analyzed, MZRN-1, is in a glassy impact melt rock, exposed on
177 top of the 80 m thick columnar jointed impact melt body at Discovery Hill near the
178 crater wall (Marion and Sylvester, 2010). Mistastin impact melt rock composition
179 ranges from ~53 to ~59 wt % SiO₂ and is interpreted to reflect mixing of different
180 proportions of various crystalline igneous target rocks (Marion and Sylvester, 2010).
181 Field relations indicate the melt ponded at Earth’s surface at ambient atmospheric
182 pressure (Marion and Sylvester, 2010). The zircon records a pre-impact U-Pb
183 crystallization age of 1403 +/- 10 Ma (2σ error, n = 3, MSWD = 0.85), consistent with
184 derivation from Mesoproterozoic bedrock in the region (Zanetti, 2015). Other phases
185 in the matrix include ballen-textured silica, which is also characteristic of high-
186 temperature (>1200 °C) impact melt (Ferrière et al., 2009).

187 Zircon grain MZRN-1 has an oscillatory zoned core in cathodoluminescence (CL)
188 that is truncated by a ~2 μm wide, bright CL rim that contains zirconia particles with
189 interspersed sub-μm silicate melt inclusions (Fig. 1). The grain is cut by irregular
190 fractures and contains two voids, all filled with glassy silicate impact melt (Fig. 1B).
191 The zircon does not preserve any diagnostic shock-deformation microstructures, such
192 as twins or the high-pressure ZrSiO₄ polymorph reidite (Timms et al., 2017)
193 (Supplementary File item 2). Minor lattice misorientations (<5°) are associated with
194 rigid block rotation across brittle fractures (Supplementary File item 2), but in general
195 the core of the zircon appears undeformed.

196 The core is surrounded by a ~40 μm thick corona of vermicular ZrO_2 crystals
197 ranging from ~0.5 to ~5 μm across and up to ~20 μm long (Fig. 1). Elongate crystals
198 tend to be aligned at high angles to the zircon core, and form domains of
199 morphologically similar clusters up to ~50 μm wide. Individual crystals generally do
200 not impinge on each other, and are separated by glassy silicate impact melt similar in
201 composition to the surrounding crystal-free impact melt (Fig. 1). ZrO_2 contains lower
202 trace elements abundances (e.g., ~350 ppm Ti, ~275 ppm Th, ~90 ppm Y) than zircon
203 (e.g., ~900 ppm Ti, ~600 ppm Th, ~1900 ppm Y) (Supplementary File item 4). Most
204 (>99%) of the ZrO_2 grains index as baddeleyite (monoclinic ZrO_2). No crystalline
205 SiO_2 phases were detected (Figs 1, 2).

206 Baddeleyite grains are commonly twinned and have a wide range of
207 crystallographic orientations (Fig. 2A, C). Morphologically distinct clusters of
208 baddeleyite grains preserve up to twelve unique crystallographic orientation variants
209 with a systematic relationships among them (Fig. 2D, Supplementary File item 5).
210 Groups of grains are crystallographically orientated approximately orthogonal to one
211 another, and systematic deviations from orthogonality of ~20° form cross-shaped
212 patterns on pole figures of <100> (Fig. 2D).

213 The occurrence of spatially-clustered baddeleyite grains with these distinctive
214 patterns of twelve orientations uniquely identifies the former presence and orientation
215 of a precursor cubic zirconia polymorph from the two-stage transformation from
216 cubic to tetragonal to monoclinic zirconia (Fig. 2B, E) (Cayron et al., 2010). Initial
217 processing of the EBSD maps of MZRN-1 involved dilation of daughter monoclinic
218 zirconia grains by an iterative nearest neighbour extrapolation routine so that they
219 impinge on one another (Supplementary File Fig. S3). This was required in order to
220 perform neighbour-pair disorientation analysis within the ARPGE software.

221 Observed peaks we identified in disorientation analysis of MZRN-1 at 90°,
222 115°, and 180° are consistent with cubic-monoclinic transformation twinning (Fig. 3).
223 We identified symmetry operators for disorientations between daughter monoclinic
224 zirconia grains at these specific angles in crystal reference frame (Fig. 3).

225 We used the software GenOVa to develop a list of theoretical symmetry
226 operators for the cubic-monoclinic transformation according to OR type 1 or type 2
227 OR (Cayron et al., 2010) to reconstruct the parent cubic zirconia grains (Figs 2, 3).
228 Parent grain reconstruction involved a quadruplet search method with a minimum of
229 five daughter grains per parent grain (Cayron et al., 2010). The disorientations we
230 observed (angles +axes) fit very well with rotations by the operators expected from
231 OR type 2, within an angular tolerance range of <5 ° (Fig. 3). We conclude that OR
232 type 2 is the OR in the samples: there is no evidence that OR type 1 is present. The
233 observed spread of the orientations (Fig. 3) is not due to a mixture of OR types 1 and
234 2, and may be due to limited crystal-plasticity in the dynamic impact melt
235 environment. We used 15-20° as the tolerance angle for the parent cubic grain
236 reconstruction (Fig. 2B, D, Supplementary File Fig. S5).

237 Reconstruction of parent cubic zirconia grains from baddeleyite electron
238 backscatter diffraction (EBSD) data shows that the dissociation corona was once
239 comprised of large (~5 to ~50 µm across) domains of vermicular single crystals of
240 cubic zirconia (Fig. 2B). The morphology of the cubic parent grains indicates that the
241 vermicular ZrO₂ grains formed a connected 3D network that broadly pseudomorphed
242 the original rim of the zircon grain. A second zircon from the same sample (MZRN-2)
243 displays the same partial dissociation texture and also successfully yields
244 reconstructed parent cubic zirconia grains (Supplementary File items 3 and 5),
245 indicating the same cubic zirconia phase heritage as MZRN-1.

246

247 **4. DISCUSSION**

248 The microstructural and geochronological data, field relations, and thermodynamic
249 constraints all support the interpretation that the zircon grains described here record
250 an extreme high-temperature history at low-pressure conditions. The Mesoproterozoic
251 U-Pb age and preserved igneous zoning of MZRN-1 indicate that the grain is a
252 xenocryst derived from one of the plutonic igneous target rocks. The complete lack of
253 crystal-plastic deformation, shock-twins and high-pressure polymorphs are consistent
254 with the grains not having been affected by high-pressure shock conditions. The
255 grains were, however, entrained into impact melt in the dynamic crater environment
256 after passage of the shock wave. The ZrO_2 coronae are therefore interpreted to have
257 developed solely due to thermal effects from immersion in super-heated impact melt
258 at ambient pressure. The inference of ambient pressure is also consistent with
259 available thermodynamic constraints that indicate that the zircon dissociation reaction
260 line has a steep Clapeyron slope ($29\text{ }^\circ\text{C}/\text{bar}$) and therefore is inevitably a low-pressure
261 process (Timms et al., 2017).

262 A minimum temperature achieved by the impact melt can be constrained via
263 analysis of phases in the zirconia corona, and we use the ZrO_2 - SiO_2 binary phase
264 diagram at 1 atm to establish the temperature of zircon dissociation (Fig. 4) (Kaiser et
265 al., 2008). The presence of a thick zirconia corona clearly records the melt achieving
266 high enough temperature for zircon dissociation to proceed (Fig. 4) (Kaiser et al.,
267 2008). Phase equilibria predict that zircon dissociates to tetragonal zirconia and
268 cristobalite at $1673\text{ }^\circ\text{C}$ (Fig. 4). The presence of silicate melt and absence of
269 cristobalite indicates the sample exceeded $1687\text{ }^\circ\text{C}$; liquid silica released by
270 dissociation would have dissolved into surrounding melt.

271 Clusters of morphologically distinct zirconia crystals that contain up to twelve
272 unique orientations with systematic misorientations are consistent with a two-stage
273 solid-state transformation from cubic zirconia via tetragonal zirconia. The former
274 presence of cubic zirconia requires that the zircon, and therefore the melt that
275 surrounds it, reached a minimum temperature of 2370 °C, but did not exceed 2700 °C
276 above which cubic zirconia would have melted (Fig. 4), and thus prevented
277 crystallographic phase heritage relations from being preserved. Upon cooling below
278 2370 °C, the displacive transformation of cubic to tetragonal zirconia results in up to
279 three distinct orientations of tetragonal zirconia whereby (001)_{tetragonal} is parallel to a
280 {100}_{cubic} (Heuer, 1987). Further cooling below ~1180 °C results in a stable
281 assemblage of baddeleyite + zircon (Fig. 4). The tetragonal –monoclinic zirconia
282 transformation is martensitic, and resulted in four orientation variants of baddeleyite
283 from each tetragonal parent (Cayron et al., 2010). Thus, the transformation of higher-
284 temperature cubic ZrO₂ polymorphs in the corona to the low temperature ZrO₂
285 polymorph baddeleyite produced systematic orientation relationships among
286 baddeleyite grains. These are interpreted as transformation (reversion) twins formed
287 upon cooling (Supplementary File item 6).

288 Even though cubic zirconia is the highest temperature ZrO₂ polymorph, its
289 stability is pressure-dependant, such that it is stabilized at lower temperature with
290 increasing pressure (Bouvier et al., 2000; Kerschhofer et al., 2000) (Supplementary
291 File item 7). However, zirconia in the Mistastin sample is a product of zircon
292 dissociation, which ostensibly is a low-pressure phenomenon that occurs at post-
293 shock conditions (see Timms et al., 2017 for further details and thermodynamic
294 calculations). All available field evidence, theoretical constraints, microstructural and
295 geochemical data supports zircon dissociation in an unconfined impact melt at

296 ambient, post-shock pressure. Therefore, the transformations among zirconia
297 polymorphs that took place after zircon dissociation necessarily also occurred in the
298 melt sheet at post-shock low-pressure conditions, which constrains the stability of
299 cubic zirconia to extremely high temperatures. Impurities, such as REE and Y, can
300 also stabilize cubic zirconia at lower temperatures (Swab, 2001) (Supplementary File
301 item 7). However, the levels of trace elements present in the ZrO₂ are much less than
302 0.1 wt% (Supplementary File item 4), and so the effects of impurities on the cubic to
303 tetragonal zirconia transformation temperature were negligible (Swab, 2001)
304 (Supplementary File item 7). Therefore, the former presence of cubic zirconia in the
305 Mistastin Lake zircon indicates that the impact melt temperature was in excess of
306 2370 °C (Fig. 3) (Kaiser et al., 2008).

307 Our study documents forensic geological evidence for a minimum temperature
308 of 2370 °C in a sample of impact melt. This temperature constraint is the highest
309 recorded by any rock on Earth's surface, and is several hundred degrees higher than
310 previous estimates for average temperatures of superheated impact melts (Fig. 5)
311 (Onorato et al., 1978; Simonds, 1975; Wünnemann et al., 2008).

312 The ability to assess the phase heritage of ZrO₂ polymorphs by
313 crystallographic orientation relationships is a new methodology for establishing the
314 thermal evolution of superheated melt. Textural evidence of zircon dissociation and
315 associated zirconia has been reported in numerous examples, including impact melt
316 rocks, tektites, lunar melt breccias, kimberlites, as well as anthropogenic settings,
317 such as slag from metal smelting and nuclear test sites (Timms et al., 2017). Given
318 that zircon is a common accessory mineral in many rocks, this approach can be used
319 to assess minimum temperatures of zircon-bearing impact melt rocks from Earth,

320 Moon, and meteorites. The broader use of determining phase heritage can potentially
321 be applied to a wide range of polymorphic transformations in geological systems.

322 Our results begin to constrain the conditions under which zircon and its U-Pb
323 isotope systematics may survive during impact cratering. During high temperature
324 and low shock pressure events, zircon is capable of surviving extreme conditions
325 without necessarily recording microstructural evidence typical of shock deformation
326 (Erickson et al., 2017; Timms et al., 2017; Timms et al., 2012). The Mesoproterozoic
327 U-Pb protolith zircon age preserved in the ~38 Ma impact melt investigated here
328 indicates that, despite being entrained in impact melt in excess of 2370 °C, the
329 residence time at high temperature was not sufficient to cause significant Pb-loss from
330 the core of the grain by volume diffusion (Cherniak and Watson, 2001). The glassy
331 matrix is further evidence that duration at extremely high temperatures was short and
332 the sample quenched rapidly. However, increased melt volumes in larger impact
333 structures with slower cooling rates may lead to complete digestion of zircon
334 xenocrysts, erasing mineralogical evidence of superheated temperatures. Early during
335 Earth's history when impact rates were high, digestion of zircon by superheated
336 impact melt may have contributed to the paucity of Hadean zircon preserved in the
337 geological record (Marchi et al., 2014).

338 The formation of >2370 °C impact melt during moderate sized impacts has
339 several significant, global-scale consequences during the early evolution of the Earth
340 and Moon when the frequency and magnitude of meteorite flux was much higher
341 (Gomes et al., 2005; Kring and Cohen, 2002). The extent of compositional
342 modification by differential volatile loss of major and trace elements from impact
343 melts by selective vaporization at temperatures >2370 °C has not been fully
344 investigated, yet are sufficiently high to cause significant volatilization of many key

345 oxide constituents of silicate melts (Lamoreaux et al., 1987). Preferential
346 volatilization of SiO₂, Na₂O and K₂O components from hot impact melt bodies could
347 provide a mechanism for localised densification of residual crust, potentially with
348 geodynamic consequences. Molecular species essential for terrestrial life, such as
349 H₂O and CO₂, are potentially lost due to volatilization of superheated impact melt.
350 High-temperature impact-melting could locally form dry, rigid, refractory residual
351 crust, resistant to subsequent melting. Moderate sized impacts, comparable to
352 Mistastin, are predicted to have completely resurfaced the Earth within the first tens
353 of millions of years after the Moon-forming event, irrespective of the choice of the
354 suggested bombardment models (Marchi et al., 2014). Therefore, the resultant net
355 effect of hypothesised high impact flux early during Earth's history, combined with
356 commonplace high temperatures of impact melts, potentially led to widespread
357 depletion of hydrous minerals from the lithosphere and therefore governed
358 rheological properties of early crust. Further investigation of the geodynamic
359 consequences of this phenomenon for early Earth would require accurate predictions
360 of impact melt temperatures, flux, size, impact angles, and velocity distributions of
361 impactors, and appropriate scaling relationships for melt volumes linked to these
362 variables. Furthermore, long term effects would need to account for rehydration
363 effects by interaction with Earth's hydrosphere. Our results on high temperature melt
364 and attendant volatilization processes have implications for the early evolution of the
365 geochemical reservoirs and multi-element isotope systematics of planetary bodies
366 (Drake and Righter, 2002) and that extremely high melt temperatures can be achieved
367 even in moderate-sized impact events, and are not limited to giant, basin-forming
368 impacts.

369

370

371 **FIGURE CAPTIONS**

372 Figure 1. A. Panchromatic cathodoluminescence (CL) image of zircon (Zrn)
373 xenocryst with a corona of baddeleyite (Bdy) and silicate glass. B. Backscattered
374 electron image of area shown by white box in A showing the zircon-baddeleyite-glass
375 interface.

376

377 Figure 2. A. Electron backscatter diffraction map coloured for measured
378 crystallographic orientation of baddeleyite. Zircon is shown in grayscale based on
379 EBSD pattern quality (band contrast). B. Map showing crystallographic orientations
380 of reconstructed parent cubic zirconia grains. Inverse pole figure (IPF) orientation
381 colour scheme. C-F. Pole figures showing measured ‘daughter’ baddeleyite
382 orientations and reconstructed cubic zirconia ‘parent’ orientations. Lower hemisphere
383 plots in sample x-y-z reference frame.

384

385 Figure 3. Results from orientation analysis of daughter monoclinic zirconia grains of
386 MZRN-1 using the software ARPGE (Cayron, 2007; Cayron et al., 2010). A.
387 Histogram of disorientation angles between daughter monoclinic grains. B.
388 Disorientation symmetry operator statistics for adjacent daughter grains for operators
389 for the type 2 orientation relationship (OR) of Cayron et al. (2010). C. Pole figures
390 showing the distribution of 90°, 115°, and 180° disorientation axes between daughter
391 grains in the crystal reference frame.

392

393 Figure 4. T-X phase diagram for ZrO_2 - $ZrSiO_4$ - SiO_2 system modified after Kaiser et
394 al. (2008). Tridymite stability after Swamy et al. (1994). Crosshatch shows zircon-
395 bearing fields, stipple shows melt-bearing fields.

396

397 Figure 5. Ranges for various types of high-temperature thermometers available in
398 geosciences. Cubic zirconia (blue bar) related to zircon dissociation (blue arrow) is
399 the thermometer for highest known temperatures (numbers on left indicate
400 references). Vaporization field is approximate and based on rates for common species
401 present in silicate melts (Lamoreaux et al., 1987).

402

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412

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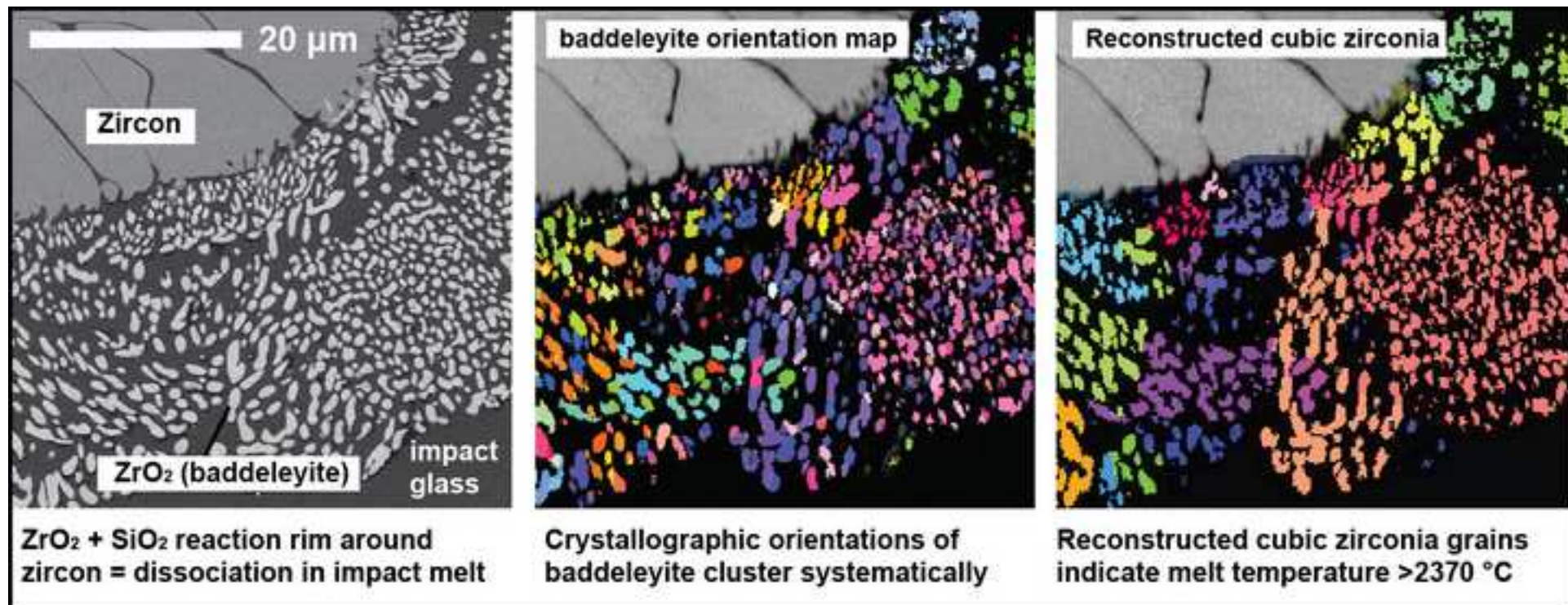
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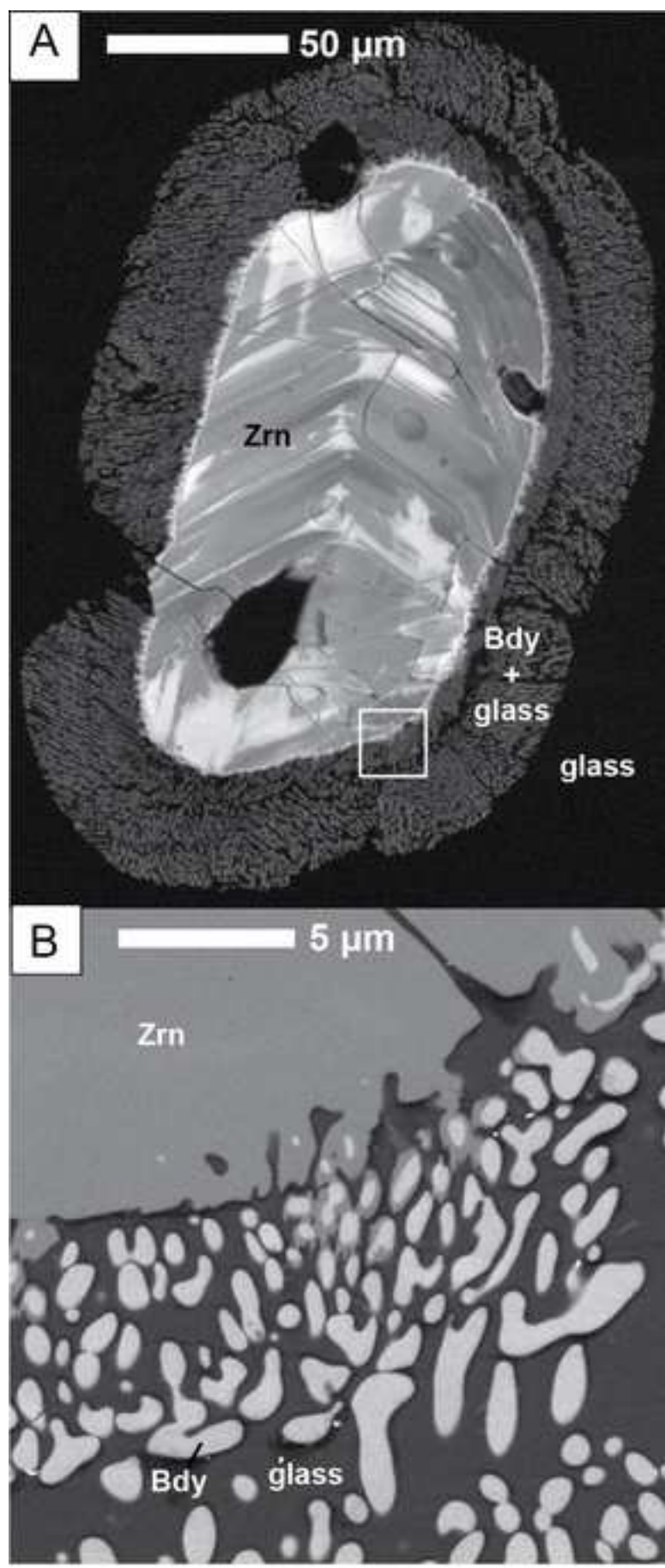
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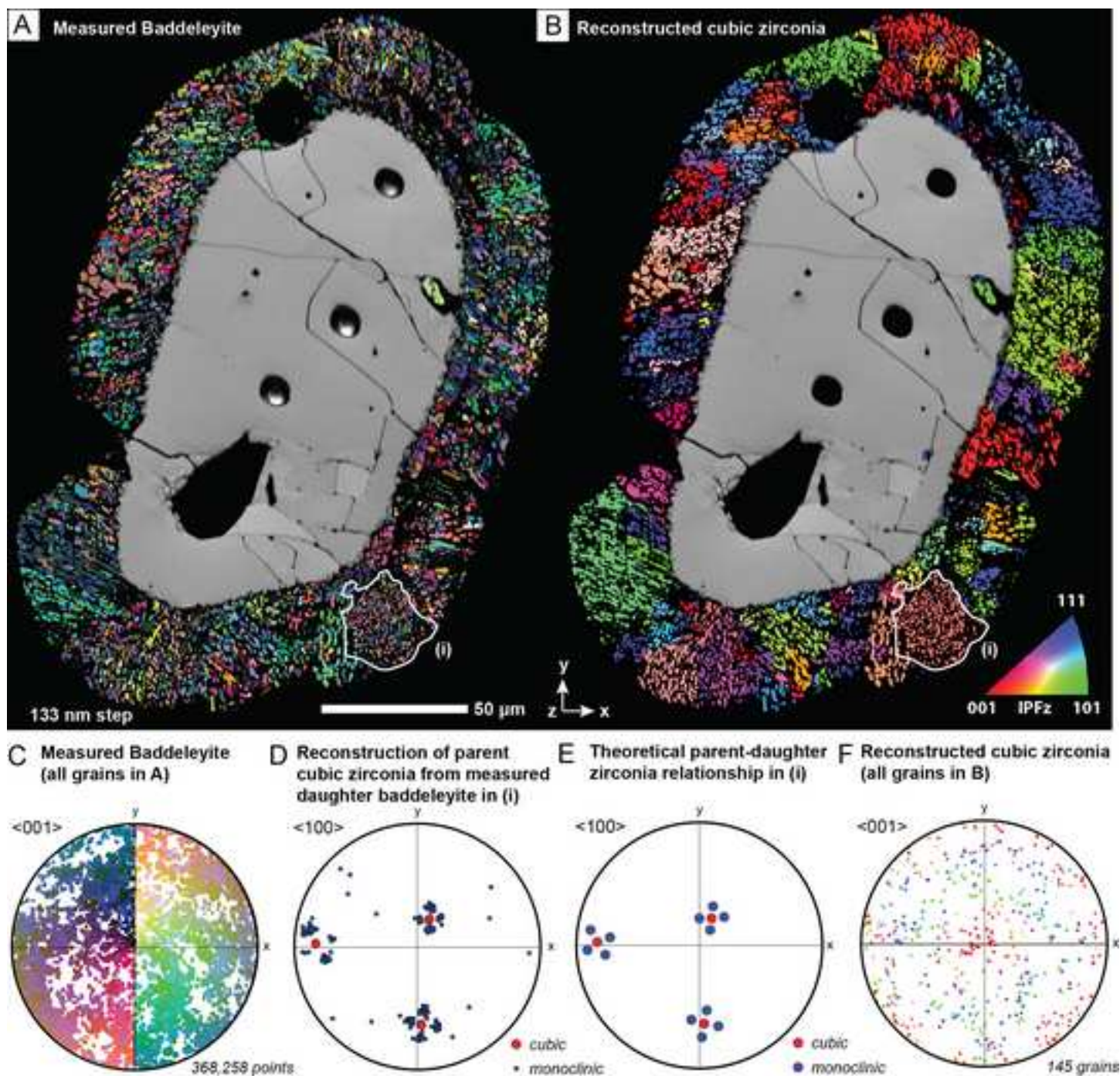




Timms et al. Figure 1

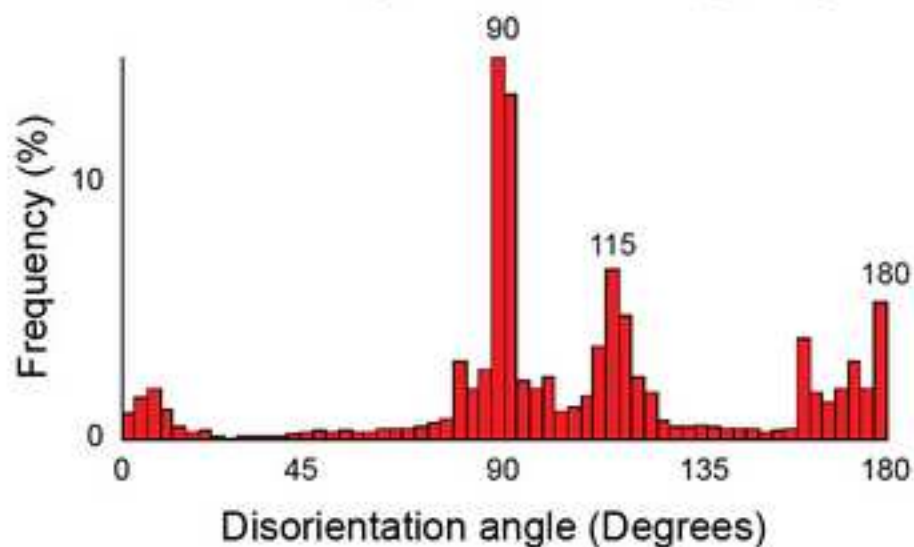
Figure 2

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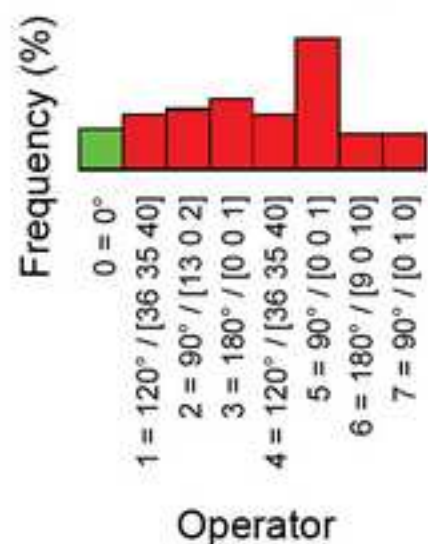
Timms et al. Figure 2

A. Disorientation angles between daughter grains



B. Operator statistics

(minimum rotation for OR type 2)



C. Distribution of disorientation axes between daughter grains (crystal reference frame)

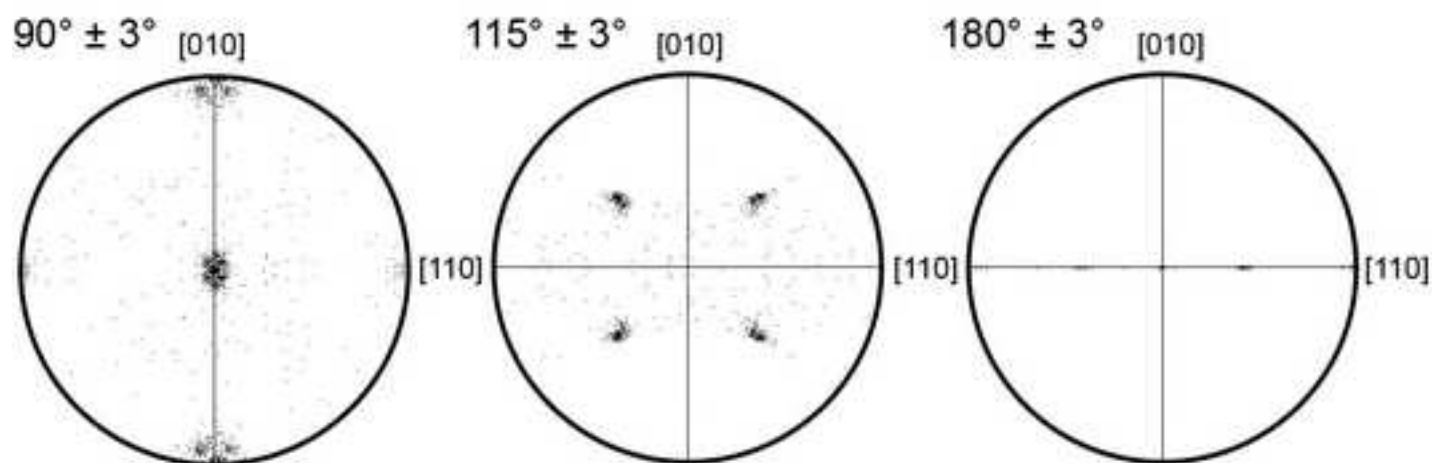
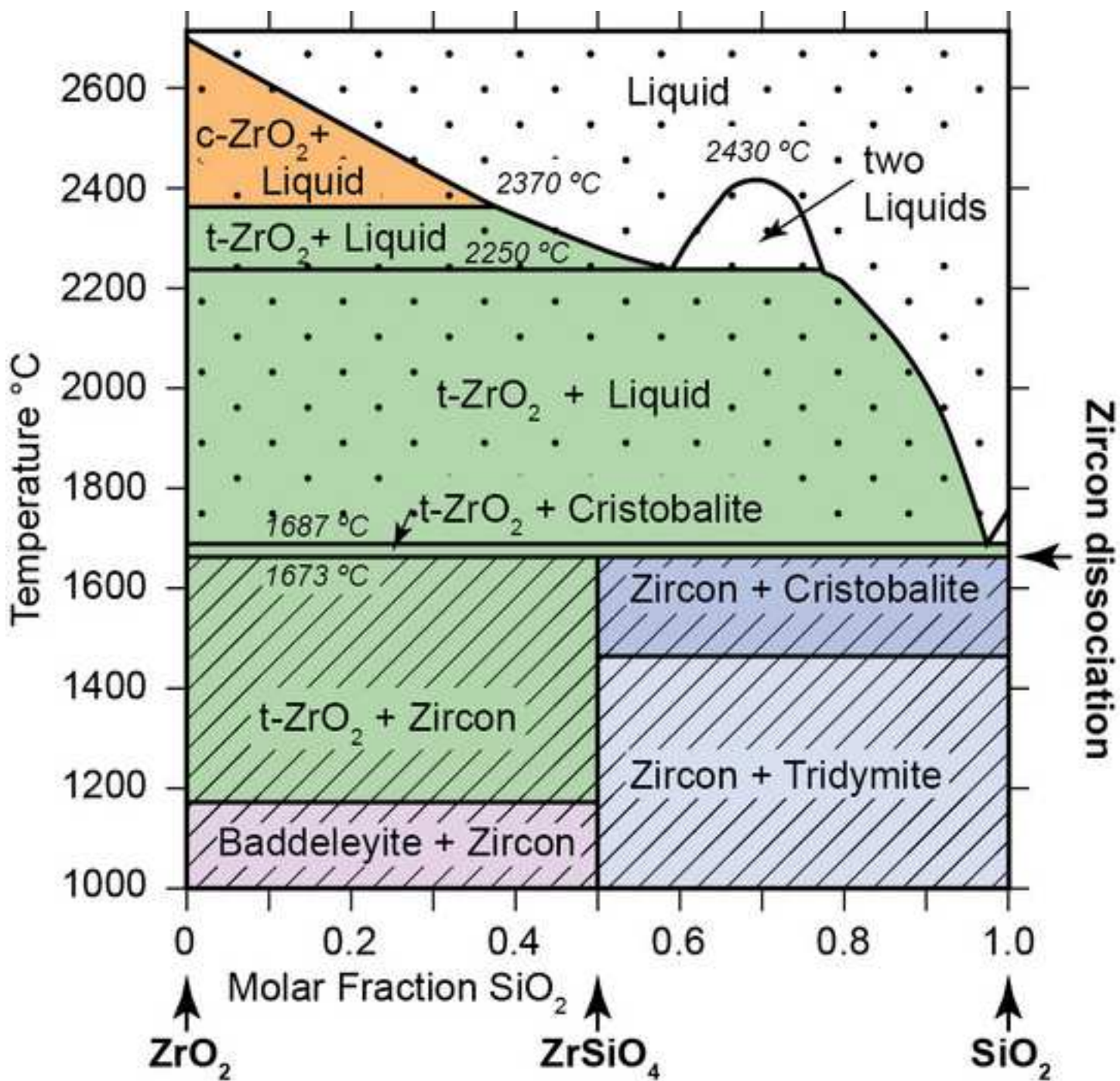
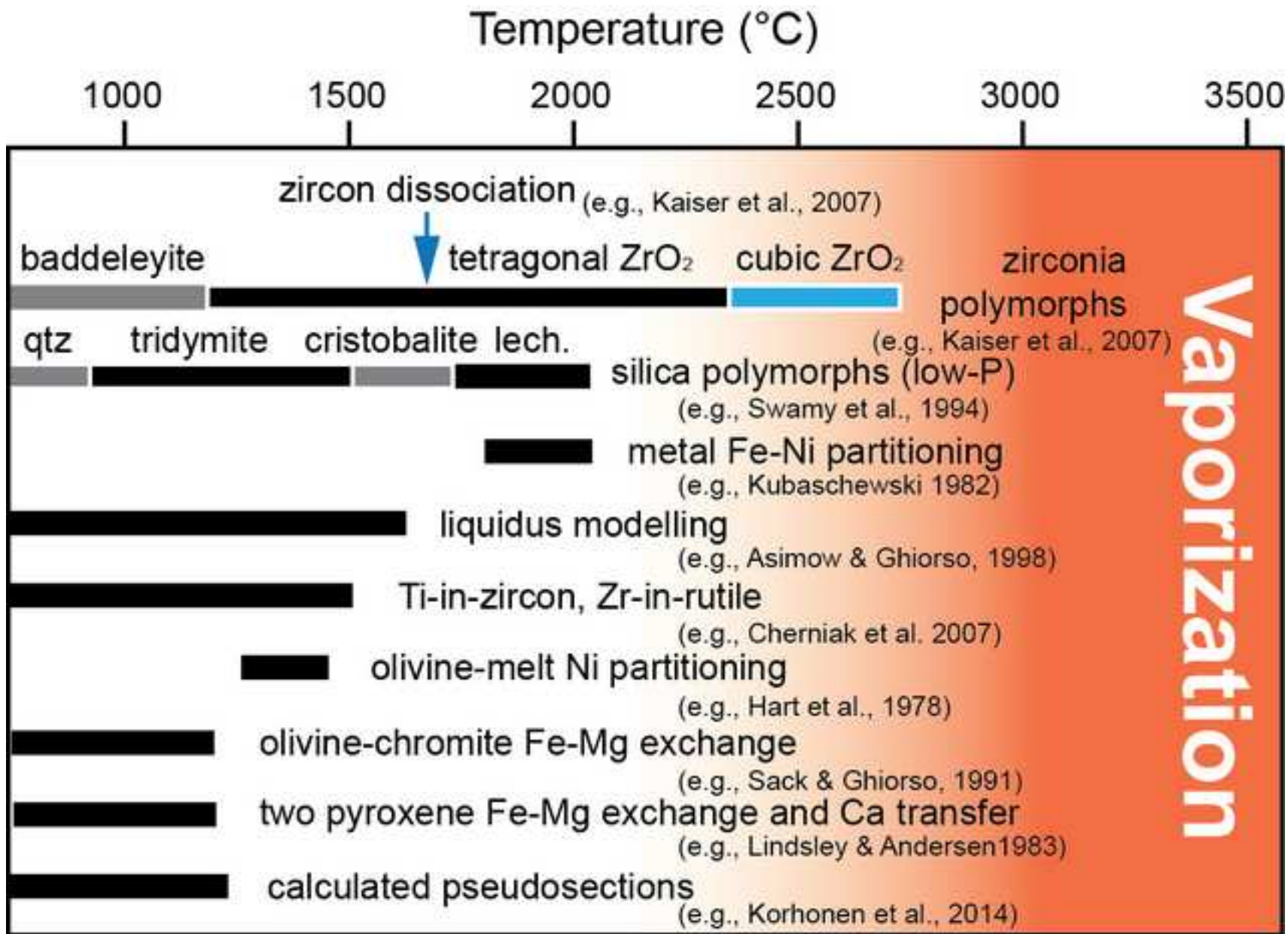


Figure 4
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Timms et al. Figure 4

Figure 5
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Supplementary File

Cubic zirconia in >2370 °C impact melt records Earth's hottest crust

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- Item 6. Transformation twin relationships in the baddeleyite corona
- Item 7. Effects of impurities and pressure on zirconia polymorph stability
- Item 8. Supplemental Reference List

Item 1. Sample preparation and scanning electron microscopy (SEM) data collection methods and settings

Sample preparation

We polished the petrographic sections of the sample with 1 μm diamond paste prior to mounting regions of interest that include the zircon grains into an epoxy disk. The sample was given a further chemical-mechanical polish for EBSD analysis using colloidal silica in NaOH for three hours on a Buehler Vibromet II polisher. A thin coat of carbon was applied to mitigate charging during electron microscopy yet permit EBSD patterns to be acquired.

Cathodoluminescence and backscatter electron imaging

At Curtin University, we performed scanning electron microscopy (SEM) using a Tescan MIRA3 field emission (FE-)SEM and atomic number contrast imaging with a pole piece backscatter detector, an accelerating voltage of 5 kV, a beam current of 22 nA, and a working distance of 15 mm. Moreover, we did cathodoluminescence imaging at 15 kV acceleration voltage and 15 mm working distance.

Electron backscatter diffraction (EBSD) mapping

We did orientation mapping via EBSD, using a Tescan MIRA3 FE-SEM with an Oxford Instruments AZTEC EDS/EBSD acquisition system at Curtin University. Operation conditions were optimised for EBSD and include a stage tilt of 70° around a horizontal axis and beam acceleration voltage of 20 kV (Prior et al., 1999). We processed EBSD patterns using 4 x 4 pixel binning and indexed using match units for zircon, cubic, orthorhombic, tetragonal and monoclinic ZrO_2 (Table S1). However, only zircon and baddeleyite could be indexed. The mean 'mean angular deviation' for zircon and baddeleyite indexing solutions was 0.41° and 0.50°, respectively.

We processed the EBSD data using Oxford Instruments Channel 5.12 software. Maps were noise reduced by removing isolated erroneous data points ('wildspike' correction) followed by a zero solution extrapolation to six nearest neighbours. Visual inspection shows that no significant artefacts were generated during noise reduction. We generated thematic maps using the Tango module of Channel 5, and include phase, EBSD pattern quality (band contrast), and crystallographic orientation maps (Figs S1 and S2). Pole figures were generated using the Mambo module of Channel 5 (Figs S1 and S2).

Table S1. Scanning electron microscopy settings and electron backscatter diffraction analysis acquisition and processing parameters.

SEM	
Make/model	Tescan Mira3 FEG-SEM
EBSD acquisition system	Oxford Instruments Aztec / Nordlys EBSD Detector
EBSD Processing software	Oxford Instruments Channel 5.10
Acceleration Voltage (kV)	20
Working Distance (mm)	~20.5
Tilt	70°
EBSD match units	
Zircon	Zircon5260, 1 atm (Hazen and Finger, 1979)
Reidite	Reidite632, 0.69 GPa (Farnan et al., 2003)
Baddeleyite (monoclinic ZrO ₂)	(Bondars et al., 1995), (Hill and Cranswick, 1994)
Tetragonal ZrO ₂	(Teufer, 1962)
Cubic ZrO ₂	ICSD card 53998 (Böhm, 1925)
Orthorhombic ZrO ₂	ICSD card 77716
Quartz	'Quartznew', HKL database (Sands, 1969)
Cristobalite	(Downs and Palmer, 1994)
Coesite	(Kirfe et al., 1979)
EBSA Acquisition, Indexing and Processing	
Sample Location	Discovery Hill, Mistastin Lake, Canada
Grain ID	MZRN-1 MZRN-2
Figures	2, S1, S3, S5 S2, S4, S6, S7
EBSA Acquisition Speed (Hz)	40 40
EBSA Background (frames)	64 64
EBSA Binning	4 x 4 4 x 4
EBSA Gain	High High
Hough resolution	60 60
Band detection (min / max)	6 / 8 6 / 8
Mean angular deviation (zircon)	<1° <1°
Mean angular deviation (reidite)	n/a n/a
Mean angular deviation (baddeleyite)	<1° <1°
Map step size (nm)	133 80
Map size (X steps / Y steps)	1428 / 1862 1495 / 1435
EBSA noise reduction routine	
Wildspike correction	Yes Yes
Nearest neighbour zero solution extrapolation	6 6

Item 2. Summary of images and electron backscatter diffraction (EBSD) data from MZRN-1

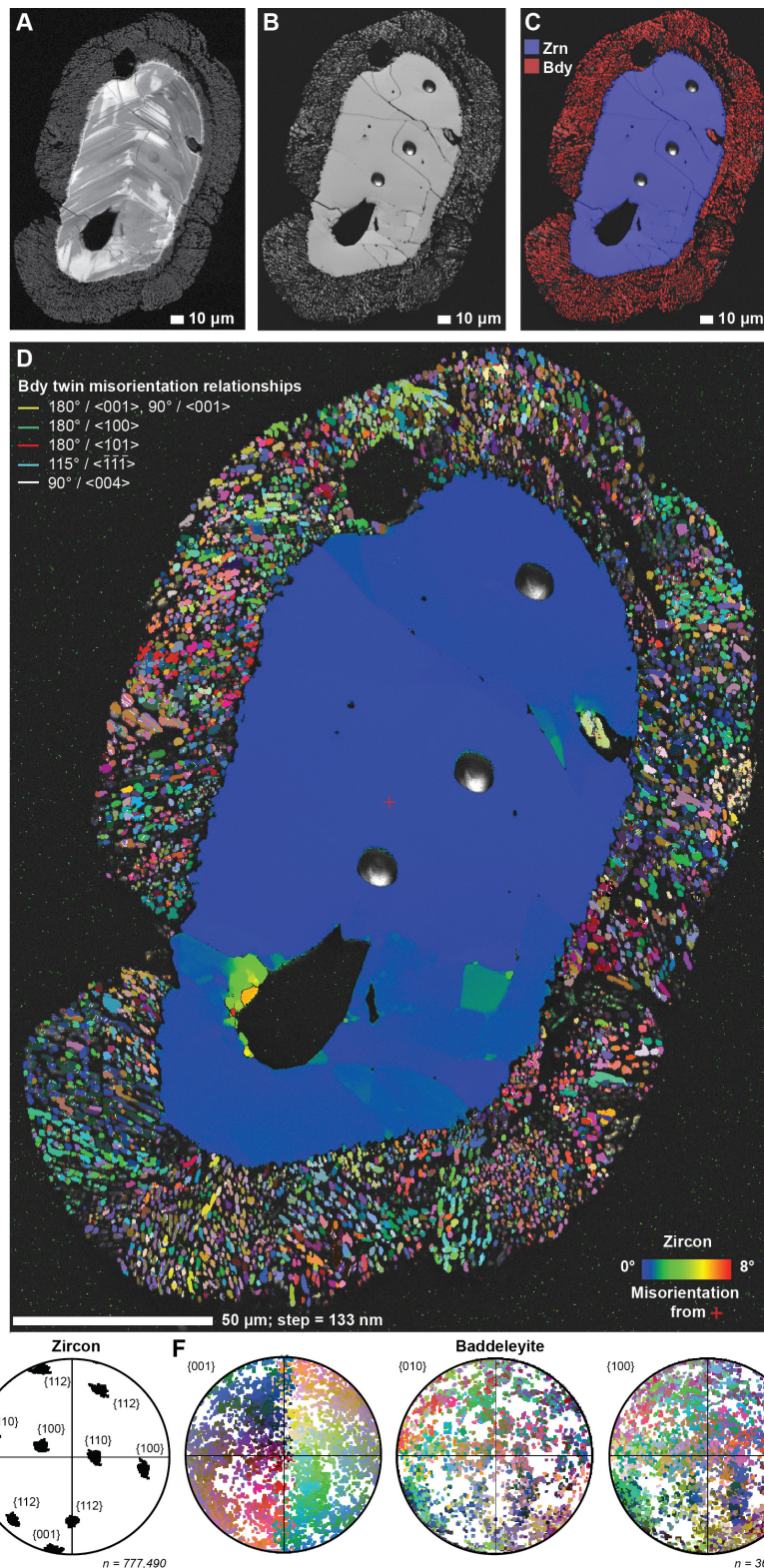


Figure S1. MZRN-1 **A.** CL image. **B.** EBSD pattern quality (band contrast) map. **C.** Phase map from EBSD data. **D.** EBSD map showing $\sim 8^\circ$ cumulative misorientations recorded by MZRN-1, solely attributed to brittle fracture block rotation. No crystal-plastic microstructures, twinning or reidite were detected. Baddeleyite in corona coloured for orientation using All Euler scheme. **E.** Pole figure of principal planes of zircon. **F.** Pole figures of principal planes of baddeleyite. **E** and **F** are lower hemisphere equal area projections in the map x-y-z reference frame.

Item 3. Summary of images and electron backscatter diffraction (EBSD) data from MZRN-2

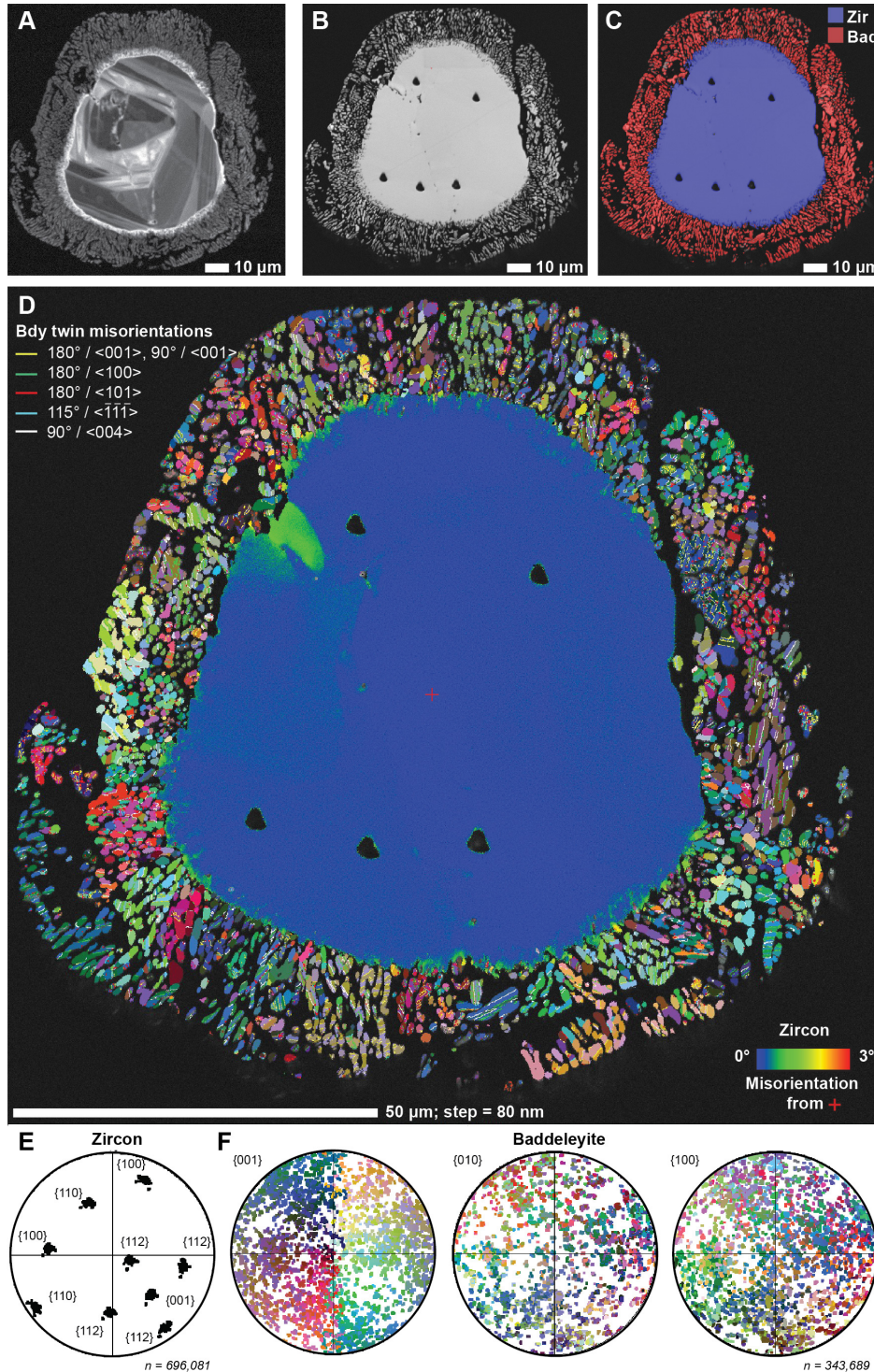


Figure S2. MZRN-2 A. CL image. B. EBSD pattern quality (band contrast) map. C. Phase map from EBSD data. D. EBSD map showing $<3^\circ$ cumulative misorientation recorded by MZRN-2, solely attributed to brittle fracture block rotation. A minor amount of crystal-plastic strain is preserved near the dissociation interface. No other crystal-plastic microstructures, twinning or reidite indicative of shock metamorphism were detected. Baddeleyite in corona coloured for orientation using All Euler scheme. E. Pole figure of principal planes of zircon. F. Pole figures of principal planes of baddeleyite. E and F are lower hemisphere equal area projections in the map x-y-z reference frame.

Item 4. Electron microprobe microanalysis (EPMA) methods and settings

We measured the concentration of selected trace elements of zircon and zirconia using a JEOL JXA-8200 electron probe microanalyser at the Department of Earth and Planetary Sciences of Washington University in St Louis, USA. This instrument is equipped with 5 wavelength-dispersive spectrometers, and an e2v silicon-drift energy-dispersive X-ray spectrometer. The WDS spot analyses included Ca, Fe, P, Zr, Si, Hf, Ti, Dy, Er, Th, Y. We used a time-dependent intensity correction routine for the WDS analyses to quantify element concentrations using the Probe for EPMA software package (Table S2).

Table S2. Selected major and trace element concentrations in zirconia, zircon and impact melt glass from time dependent intensity WDS analysis, after Zanetti (2015). Concentrations in parts per million (ppm). Bdl = below detection limit.

Acc. V (kV)	150	150	50	50
Probe current (nA)				
Element	Zircon core (bright CL)	Zircon core (dark CL)	Zircon rim	Baddeleyite
Fe	250-475	360-800	840-1000	~300
Ti	150-250	170-330	200-400	~350
Th	~35	200-340	20?	~275
Y	Bdl	1000-1780	670	~90
Dy	Bdl	60?	10-220	Bdl
Er	Bdl	Bdl	≤ 150	Bdl
P	280	300-400	-	~640

Item 5. Reconstruction of parent cubic zirconia grains from daughter monoclinic zirconia crystallographic orientation data

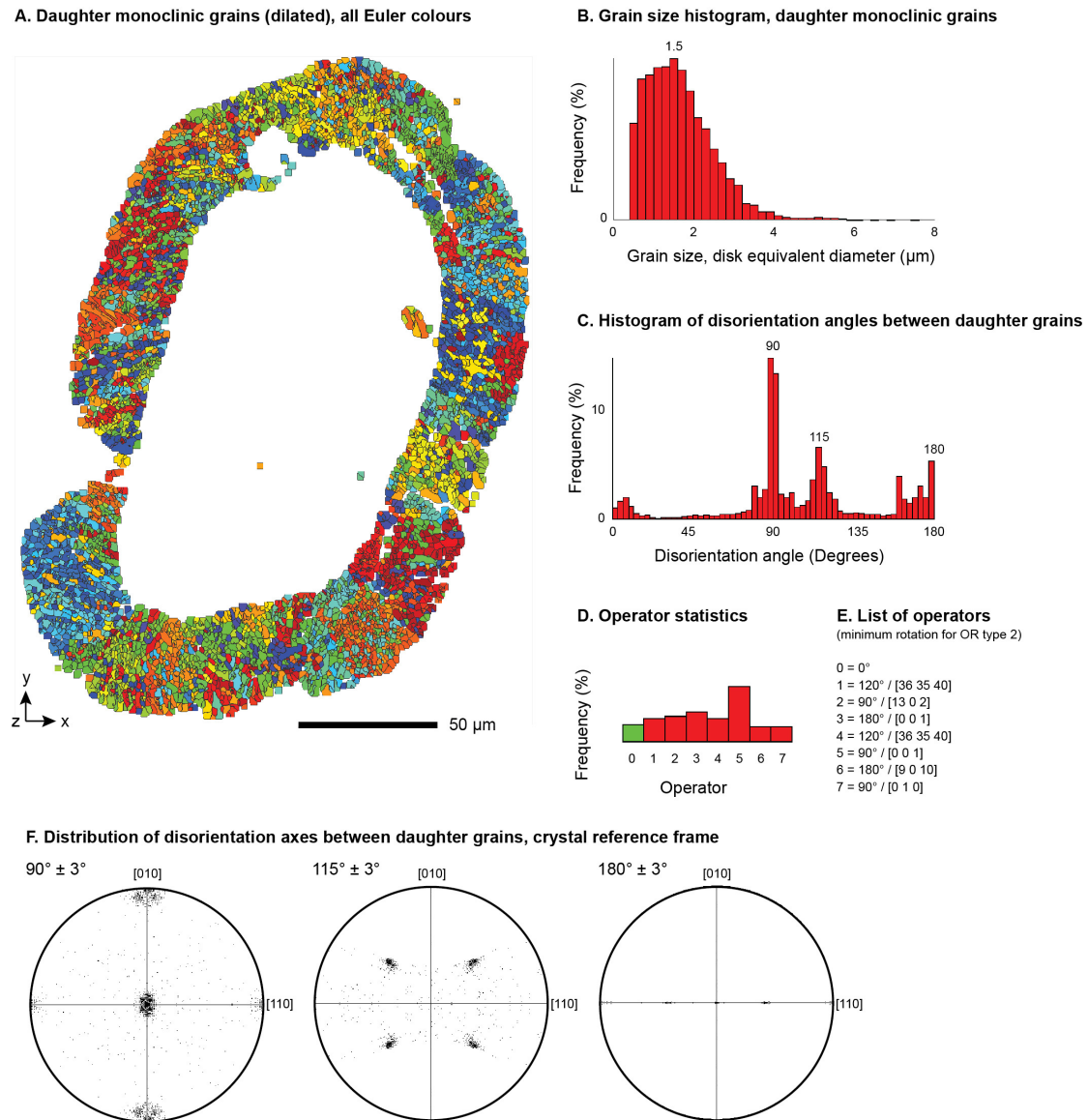
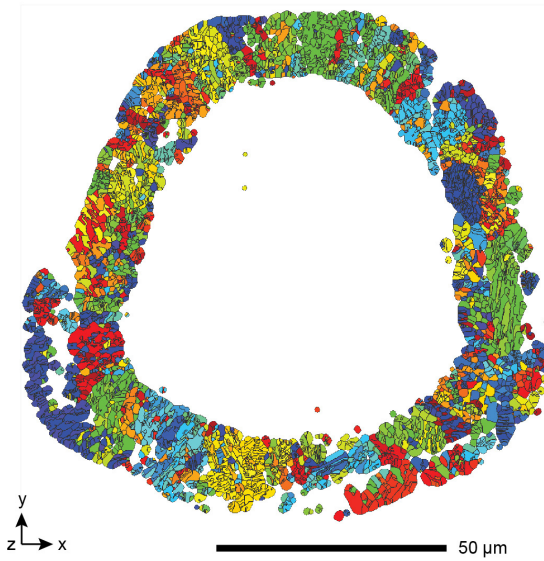
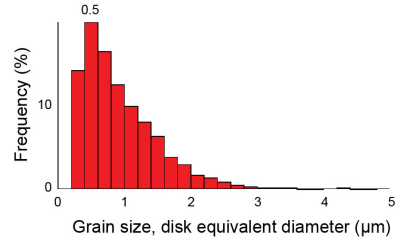


Figure S3. Results from orientation analysis of daughter monoclinic zirconia grains of MZRN-1 using the software ARPGE (Cayron, 2007; Cayron et al., 2010). A. Map showing the daughter monoclinic grains after grain the dilation procedure. All Euler colours represent different crystallographic orientations. B. Grain size histogram of daughter monoclinic grains shown in A. C. Histogram of disorientation angles between daughter monoclinic grains shown in A. D. Disorientation symmetry operator statistics for adjacent grains in A. E. List of disorientation operators for the type 2 orientation relationship (OR) of Cayron et al. (2010). F. Pole figures showing the distribution of 90°, 115°, and 180° disorientation axes between daughter grains in the crystal reference frame.

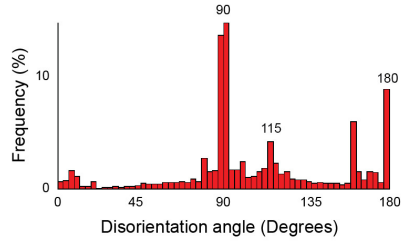
A. Daughter monoclinic grains (dilated), all Euler colours



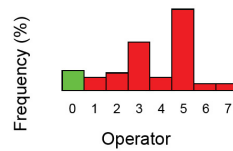
B. Grain size histogram, daughter monoclinic grains



C. Histogram of disorientation angles between daughter grains



D. Operator statistics



E. List of operators

(minimum rotation for OR type 2)

- 0 = 0°
- 1 = 120° / [36 35 40]
- 2 = 90° / [13 0 2]
- 3 = 180° / [0 0 1]
- 4 = 120° / [36 35 40]
- 5 = 90° / [0 0 1]
- 6 = 180° / [9 0 10]
- 7 = 90° / [0 1 0]

F. Distribution of disorientation axes between daughter grains, crystal reference frame

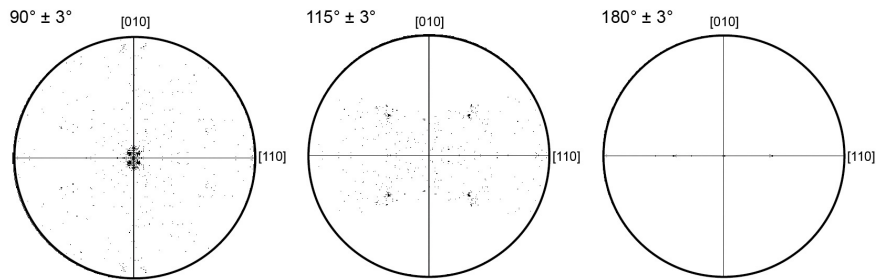
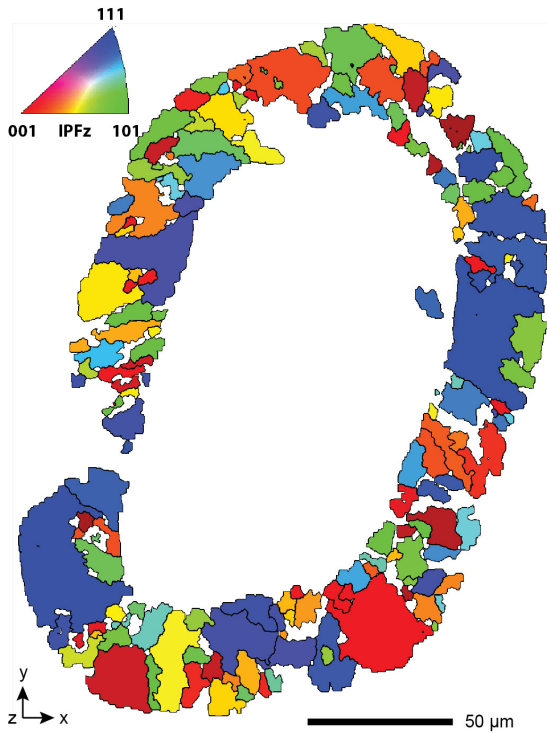
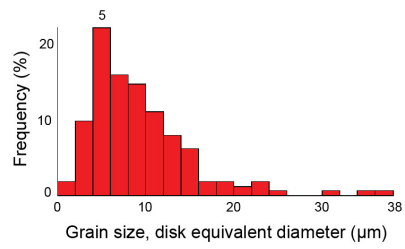


Figure S4. Results from orientation analysis of daughter monoclinic zirconia grains of MZRN-2 using the software ARPGE (Cayron, 2007; Cayron et al., 2010). A. Map showing the daughter monoclinic grains after grain the dilation procedure. All Euler colours represent different crystallographic orientations. B. Grain size histogram of daughter monoclinic grains shown in A. C. Histogram of disorientation angles between daughter monoclinic grains shown in A. D. Disorientation symmetry operator statistics for adjacent grains in A. E. List of disorientation operators for the type 2 orientation relationship (OR) of Cayron et al. (2010). F. Pole figures showing the distribution of 90°, 115°, and 180° disorientation axes between daughter grains in the crystal reference frame.

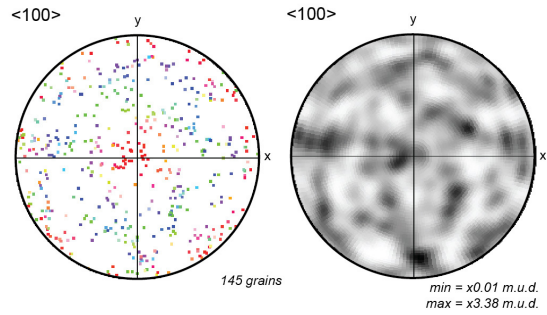
A. Parent cubic grains (dilated), IPFz colours



B. Grain size histogram, parent cubic grains



C. Pole figure of parent cubic grains, sample reference frame



D. Pole figures of selected reconstructed parent cubic grains with daughter monoclinic grains, sample reference frame

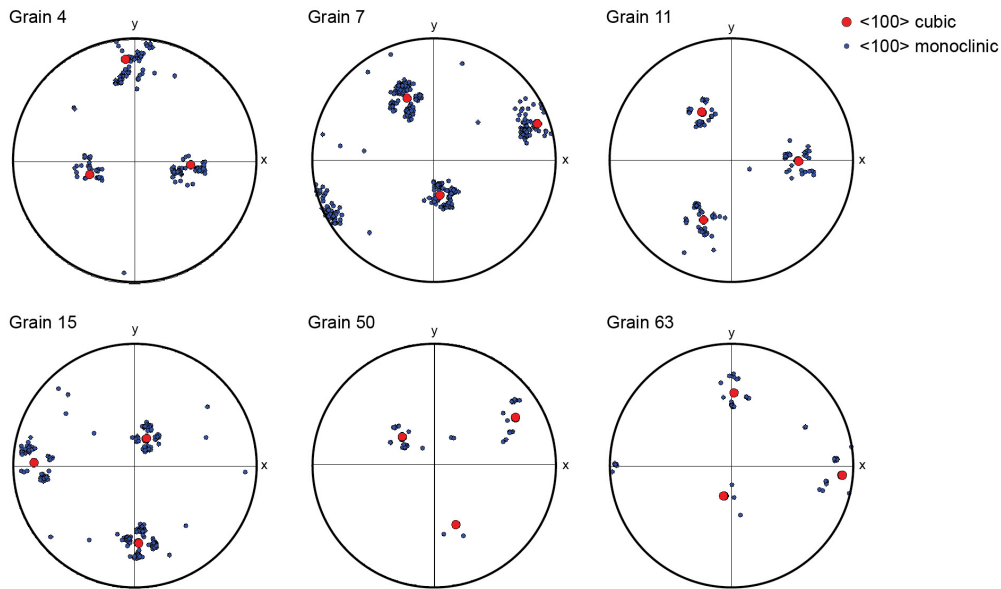


Figure S5. Reconstructed dilated parent cubic zirconia grains for MZRN-1 via the ARPGE software using type 2 OR after Cayron et al. (2010) (see Fig. S3). A. Map of parent cubic zirconia grains, all Euler orientation colour scheme. B. Grain size histogram of daughter monoclinic grains shown in A. C. Pole figure of $\langle 100 \rangle$ of the reconstructed cubic zirconia grains in the sample x-y-z reference frame. Left: colour scheme as in A; right: contoured at one point per grain. D. $\langle 100 \rangle$ Pole figures of selected reconstructed parent cubic zirconia grains (red) with daughter monoclinic zirconia grains (blue). Pole figures are lower hemisphere equal area projections in the map x-y-z reference frame.

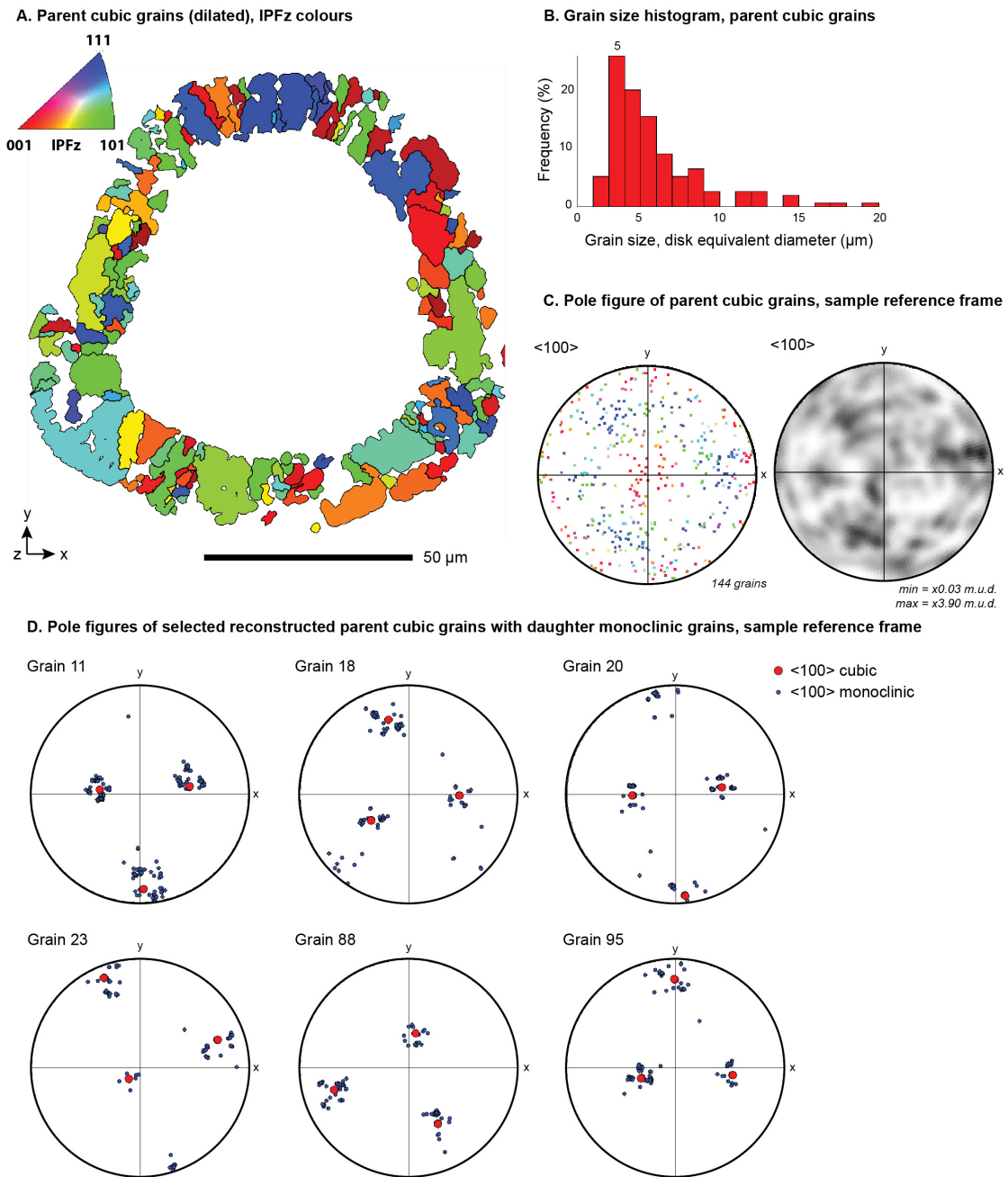


Figure S6. Reconstructed dilated parent cubic zirconia grains for MZRN-2 via the ARPGE software using type 2 OR after Cayron et al. (2010) (see Fig. S4). A. Map of parent cubic zirconia grains, all Euler orientation colour scheme. B. Grain size histogram of daughter monoclinic grains shown in A. C. Pole figure of $\langle 100 \rangle$ of the reconstructed cubic zirconia grains in the sample x-y-z reference frame. Left: colour scheme as in A; right: contoured at one point per grain. D. $\langle 100 \rangle$ Pole figures of selected reconstructed parent cubic zirconia grains (red) with daughter monoclinic zirconia grains (blue). Pole figures are lower hemisphere equal area projections in the map x-y-z reference frame.

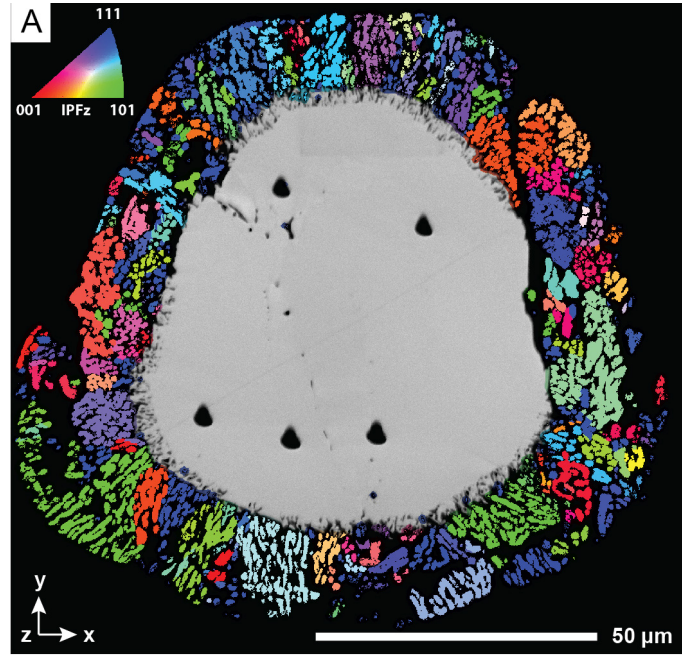


Figure S7. A. Map showing fully reconstructed parent cubic zirconia grains from MZRN-2. IPFz colour scheme. EBSD pattern quality (band contrast) for zircon shown in grayscale, and non-indexed points are black.

Item 6. Transformation twin relationships in the baddeleyite corona

Table S3. Transformation twin relationships in the baddeleyite corona. K_1 = composition plane; η_1 = shear direction.

Misorientation relationship	K_1	η_1	Type
180° <001>	(100)	[001]	Compound
180° <100>	(001)	[100]	Compound
180° <101>	($\bar{1}0\bar{1}$)	[$10\bar{1}$]	Compound
90° <004>		[$\bar{1}\bar{1}0$]	Type 2
90° <001>	($1\bar{1}0$)		Type 1
90° <001>		[$0\bar{1}\bar{1}$]	Type 2
85° <100>	($01\bar{1}$)		Type 1
115° < $\bar{1}\bar{1}\bar{1}$ >	?	?	unknown

Item 7. Effects of impurities and pressure on zirconia polymorph stability

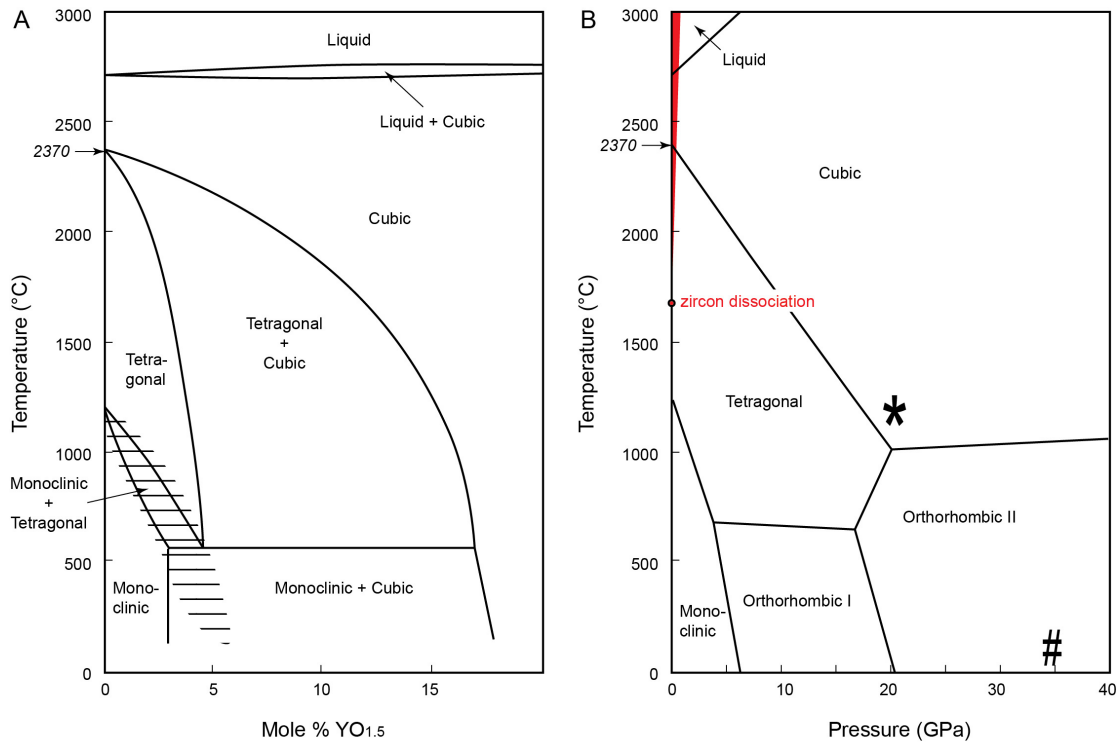


Figure S8. A. The effects of YO_{1.5} impurities on the stability of ZrO₂ polymorphs, after Swab (2001). B. The effects of pressure on the stability of ZrO₂ polymorphs, after Bouvier et al. (2000) and references therein. The red dot and red shaded area show the conditions at which zircon dissociates to zirconia + silica after Kaiser et al. (2008) and Timms et al. (2017), respectively.

Item 8. Supplemental Reference List

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