

# **Amino-Functionalised Zr-MOF Nanoparticles for Adsorption of CO<sub>2</sub> and CH<sub>4</sub>**

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## **Abstract**

Amino functionalised Zr-MOF (amino-Zr-MOF) was synthesised using 2-aminoterephthalic acid as an organic linker. The physicochemical properties of the material were characterised by XRD, SEM, TGA, FTIR and N<sub>2</sub> adsorption to understand its crystalline structure, morphology, thermal stability, and porous structure. CO<sub>2</sub> adsorption isotherms on amino-Zr-MOF were obtained at 1 atm and different temperatures. In addition, CO<sub>2</sub> and CH<sub>4</sub> adsorption at high pressure (up to 10 atm) was also measured. CO<sub>2</sub> adsorption capacity on amino-Zr-MOF was 9 mmol/g at 988 kPa, 0 °C while CH<sub>4</sub> adsorption capacity was 3.7 mmol/g at 900 kPa and 0 °C. The heat of CO<sub>2</sub> adsorption on amino-Zr-MOF was estimated to be 29.4 kJ/mol. Continuous column tests of CO<sub>2</sub> adsorption were performed at different concentrations of CO<sub>2</sub> in nitrogen at 20 mL/min and 0.7 g adsorbent and total adsorbed amount of CO<sub>2</sub> within the column during the breakthrough time was calculated to be 4.55, 5.26 and 4.37 mmol/g at 10%, 15% and 20%CO<sub>2</sub>, respectively.

Keywords: Zr-MOF, Amino-Zr-MOF, Carbon dioxide, Methane, Adsorption.

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## **1. Introduction**

Over the last few decades, carbon dioxide removal technologies from industrial emissions have been dependent on traditional methods, absorption and adsorption, which are very expensive and energy-intensive due to low capacity and selectivity of materials [1]. Metal organic frameworks (MOFs) as new emerging porous materials have been investigated to be alternative adsorbents for various industrial and environmental applications [2]. The applications of MOFs have covered several areas including gas purification and separation [3], gas storage [4], drug delivery [5] and catalysis, etc [6,7]. Unprecedentedly, MOFs have many characteristics such as ability to be designed with required specifications, high rigidity and flexibility to be functionalised, and exceptional low density [8]. The synthesis of MOFs can result in a robust extended framework with permanent porosity and these porosities may be fulfilled when the molecular building nets are connected by strong bonds [9, 10]. Previously, some transition metals have been used in the synthesis of MOFs, but priority is given to Zn and Al [2, 11].

Zirconium oxide ( $ZrO_2$ ) has stable properties and the nature of zirconium atom connection with oxygen [12] makes it attractive in synthesis of MOFs. Moreover, it was observed that the synthesis of organic ligand of this framework could be controlled to get several Zr-MOFs (UiO-66, UiO-67 and UiO-68) [13]. Modification and application of Zr-MOFs have been also reported. Recently, Kandish et al. [14] synthesised Zr-MOF (UiO-66) with different functional groups by using different linker ligands,  $H_2N-H_2BDC$ ,  $O_2N-H_2BDC$ , and  $Br-H_2BDC$ . Silva et al. [15] used both UiO-66 and UiO-66- $NH_2$  as photocatalysts for hydrogen generation while Vermoortele et al. [16] selected amino-UiO-66 for the cross-aldol condensation. These researchers used different procedures and molar ratios in the synthesis of amino-modified Zr-MOF. However, few investigations have been reported in Zr-MOF application for  $CH_4$  and  $CO_2$  adsorption [17].

In this work, we synthesised an amino-Zr-MOF sample by using a different precursor ratio and activation process at varying temperatures. The main work focused on the characteristics of this material and its application for  $CO_2$  and  $CH_4$  adsorption. Carbon dioxide adsorption capacity was determined at both low and high pressures while methane adsorption capacity was obtained at high pressure. Continuous column testes were also carried out for  $CO_2$  adsorption.

## **2. Experimental**

### ***2.1 Chemicals***

All chemicals including 2-amino-terephthalic acids (NH<sub>2</sub>-BDC, 99%), terephthalic acid (BDC, 99.8%), zirconium chloride (ZrCl<sub>4</sub>, 99.9%), dimethylformamide (DMF, 98%), methanol (CH<sub>3</sub>OH, 99%), chloroform (CHCl<sub>3</sub>, 99.98%) were supplied by Sigma-Aldrich without further purification.

## **2.2 Synthesis and activation**

Synthesis of Zr-MOF (UiO-66) follows a reported procedure [13]. In detail, 0.053 g of ZrCl<sub>4</sub> (0.227 mmol) and 0.034 g of 1,4-benzenedicarboxylic acid (H<sub>2</sub>BDC, 0.227 mmol) were dissolved in 24.9 g of N,N-dimethylformamide (DMF, 340 mmol) at room temperature. The solution was placed in an autoclave of 45 mL and placed in a pre-heated oven at 120 °C for 24 h. The final product was activated by immersing in chloroform for 4 d and then dried and heated under vacuum at 190 °C overnight. This activation process was found to enhance the surface area and carbon dioxide adsorption [17].

In a typical procedure to synthesise amino-Zr-MOF (amino-UiO-66), 1.47 g (6.31 mmol) ZrCl<sub>4</sub> were dissolved in 15 mL DMF solution and stirring for around 20 min. Meanwhile, 1.06 g (5.85 mmol) NH<sub>2</sub>-BDC were dissolved in 68 mL DMF solution and stirring for 30 min. Then, the two prepared solutions were mixed and placed in an autoclave of 125 mL. The sealed autoclave was then placed in an oven at 120 °C for 48 h. Finally, the greenish yellow crystalline product was extracted from the solution by vacuum filtration. The crystalline product was activated by immersing in methanol for 5 d, and then it was filtered, dried and heated under vacuum at 200 or 300 °C for 12 h. Methanol activation could remove the non-coordinated amino-BDC from the pores.

## **2.3 Characterisation of materials**

A Perkin-Elmer 100 FT-IR spectrometer was used to investigate functional groups of crystalline materials. Scans were conducted from 650 to 4000 cm<sup>-1</sup> with a resolution of 4 cm<sup>-1</sup> using a universal ATR-Diamond/ZnSe as IR detector. Hydrothermal stability of the crystalline structure was checked by a TGA instrument (TGA/DSC1 STAR<sup>®</sup> system-METTLER TOLEDO). A sample of 10 mg was loaded into an alumina pan and placed automatically in the TGA furnace to heat in Argon gas of 20 mL/min at a heating rate of 10 °C/min from 35 to 900 °C. X-ray powder diffraction patterns were measured on a X-ray diffractometer (D8 Advanced Bruker Axs) with a transmission mode using Cu K $\alpha$  radiation at 2 $\theta$ =5-70° to evaluate the stability of the crystalline structure. N<sub>2</sub> physisorption measurements were carried out on a Quantachrome instrument (Autosorb-1) at -196 °C. The surface area was obtained by the BET method and pore size distribution was determined by the BJH

method. In general, 15 mg of samples were degassed at 200 °C under vacuum for 6 h and then the sample was used in adsorption/desorption measurement on the same instrument. Morphology of materials including the shape and the size of crystals was determined by a SEM (Zeiss NEON 40 EsB CrossBeam) with high resolutions.

#### ***2.4 Adsorption at low pressure***

A static volumetric technique with an apparatus of Micromeritics (Gemini I-2360) was used to determine adsorption isotherm of CO<sub>2</sub> (99.995%) at 0 and 23 °C, respectively, under pressure up to 100 kPa. In each test, about 0.3 g of amino-Zr-MOF was degassed by Vacprep 061 overnight at 200 °C under vacuum. The heat of CO<sub>2</sub> adsorption was calculated from the Clausius-Claperyron equation relying on the data of adsorption isotherms which were measured at different temperatures by the GemminiI-2360.

#### ***2.5 Adsorption at high pressure***

A Micromeritics-ASAP2050 was used to measure the adsorption isotherms of carbon dioxide (99.995%) and methane (99.995%) at high pressure up to 1000 kPa. A sample was first degassed on the same instrument by increasing the temperature to 100 °C at a heating rate of 10 °C/min and hold at this temperature for 2 h. Then the temperature was increased again at the same heating rate to 200 °C under vacuum (up to 0.4 kPa) and hold at these conditions for 8 h. Finally, the degassed sample was used for the analysis at 0 °C.

#### ***2.6 Breakthrough experiments of CO<sub>2</sub> adsorption***

CO<sub>2</sub> capture in a continuous column was tested by setting up breakthrough experiments. A mixture of N<sub>2</sub> (99.999%) and CO<sub>2</sub> (99.995%) in different molar concentrations (10, 15, and 20 mol% of CO<sub>2</sub>) was used. A total flow rate at 20 mL/min during all breakthrough tests was maintained and controlled by calibrated mass flow controllers (GFC17-CO<sub>2</sub> and GFC-N<sub>2</sub>, AALBORG). The stainless steel adsorption column has a length of 150 mm and inside diameter of 4 mm. In each run, 0.7 g of amino-Zr-MOF (particle size ranged from 0.1 to 1 μm) was packed in the column with quartz wool on the ends of the column to avoid suspending of packed materials. A gas chromatograph instrument (GC-17A, Shimadzu) was used to measure the concentration of carbon dioxide effluent from the column by a TCD detector using He as a carriage gas.

### **3. Results and discussion**

#### ***3.1 Characterisation of amino-Zr-MOFs***

Fig.1 shows FTIR spectra of amino-Zr-MOF as-synthesised and activated by methanol and heating at varying temperatures. In Fig. 1a, the bands of 1400-1767  $\text{cm}^{-1}$  are referred to carboxylic functional groups [18]. The carboxyl groups from free aromatic carboxylic acid were observed at 1656  $\text{cm}^{-1}$  [19] on the as-synthesized sample; however, the peak was removed after heat activation, due to solvent extraction and heat treatment. For the samples of as-synthesized and activated at 200  $^{\circ}\text{C}$ , the band of 1430-1533  $\text{cm}^{-1}$  was observed, indicating the presence of amino-carboxylate compounds that were coordinated with zirconium metal centre by  $-\text{CO}_2$  asymmetrical stretching with the peaks at 1497  $\text{cm}^{-1}$  and 1564  $\text{cm}^{-1}$  or  $-\text{CO}_2$  symmetrical stretching with the peaks at 1385 and 1424  $\text{cm}^{-1}$  [18, 20, 21]. From magnification part of the spectrum of as-synthesized amino-Zr-MOF (Fig.1b), two peaks with much lower intensities appearing at 3376 and 3457  $\text{cm}^{-1}$  were related to primary amines,  $-\text{NH}_2$ , on the organic linker. Low intensities of these peaks before heat activation may be attributed to the strong bonding between the amino groups on the coordinated acid with  $\text{C}=\text{O}$  groups of free  $\text{NH}_2$ -BDC inside the pores and bridging OH groups on the metal centre (bridging OH groups can interact with amino groups on the organic linker by hydrogen bond) [22, 23]. In addition, the amino group of organic linker was affected by moisture from the surroundings by hydrogen bonding interaction. After activation at 200  $^{\circ}\text{C}$ , a small shift in the position of amino-groups stretching can be observed and the primary amines,  $-\text{NH}_2$  on the organic linker can be clearly seen at 3394 and 3480  $\text{cm}^{-1}$ , referring to symmetric and asymmetric vibrations of  $\text{NH}_2$  groups, while the bridging OH group can be observed at 3675  $\text{cm}^{-1}$  [21].

**[Insert Fig.1]**

From the spectrum of amino-Zr-MOF after heating at 300  $^{\circ}\text{C}$ , it was found that the intensities of peaks at 1564, 1497, 1424 and 1384  $\text{cm}^{-1}$  were reduced. Also, the width of the band 1479-1785  $\text{cm}^{-1}$  was reduced, indication of collapse of the structure. This can be confirmed by TGA profiles of amino-Zr-MOF (Fig. 2), which showed a broad weight loss process at 350  $^{\circ}\text{C}$  on amino-Zr-MOF treated at 300  $^{\circ}\text{C}$ , due to breakup of the interconnection between the functional groups and oxygen of metal centre inside the pores. Meanwhile, the weight loss profile of amino-Zr-MOF treated at 200  $^{\circ}\text{C}$  showed two weight loss processes at 350 – 550  $^{\circ}\text{C}$ , suggesting the presence of the interconnection binding. By breaking this interconnection, the whole structure of Zr-MOF was collapsed and the most pores were blocked. In a previous report, such a weight loss was not found [14], due to a different activation process.

**[Insert Fig.2]**

XRD patterns of amino-Zr-MOFs are illustrated in Fig.3. The profiles of the as-synthesised amino-Zr-MOF and amino-Zr-MOF activated at 200 °C were similar and show a similar pattern of UiO-66 as reported in the previous investigation [13]. However, the sample XRD after activation at 300 °C was totally different. The XRD profile presented a broad peak at  $2\theta = 7.5^\circ$ , suggesting amorphous structure due to the collapse of crystalline structure. This further confirmed the FTIR and TGA results.

**[Insert Fig.3]**

The nitrogen isotherms and pore size distributions of amino-Zr-MOFs are presented in Fig.4. It can be seen that amino-Zr-MOF activated at 200 °C presented a highly developed microporous structure with much higher N<sub>2</sub> adsorption than the sample after heating treatment at 300 °C. The pore size distribution of amino-Zr-MOF activated at 200 °C showed two peaks centred at 1.5 and 3.3 nm, respectively. However, amino-Zr-MOF activated at 300 °C showed a shift of pore size to larger pore range. The two peaks were changed to 2.4 and 4.4 nm, respectively.

**[Insert Fig.4]**

The BET surface area of amino-Zr-MOF activated at 200 °C was 1220 m<sup>2</sup>/g while the Langmuir surface area was 1395 m<sup>2</sup>/g. The average pore radius was 0.99 nm and the total pore volume was 0.611 cm<sup>3</sup>/g. The surface area of this sample is similar to the result reported by Zlotea et al. [24]. However, the BET surface area of the sample activated at 300 °C was significantly reduced to 180 m<sup>2</sup>/g, which reflects the destruction of porous structure. A comparison of amino-Zr-MOF with other MOFs shows that amino-Zr-MOF still has high porosity (Table 1). These above results suggested that activation of as-synthesised amino-Zr-MOF could be only done below 300 °C. Above 300 °C, the crystalline structure of amino-Zr-MOF could be destroyed to produce amorphous phase.

Table 1. BET surface area of different amino functionalized metal organic frameworks

Sample	S <sub>BET</sub> (m <sup>2</sup> /g)	Reference
MIL-125(Ti)	1130	[24]
UiO-66(Zr)-NH <sub>2</sub>	1206(1280)	[24]
IRMOF-3 DEF	3683	[25]
Amino-MIL-53(Al)	675	[25]
Amino-MIL-101(Al)	2100	[26]
Amino-Zr-MOF	1220	This work

Fig. 5 shows SEM photo of amino-Zr-MOF. As seen, amino-Zr-MOF presented as symmetrical crystals with triangular base-pyramid shape and the particle size was about 200 nm.

**[Insert Fig.5]**

### 3.2 Adsorption study

Fig. 6 displays CO<sub>2</sub> adsorption isotherms on amino-Zr-MOF at 0 and 23 °C. Amino-Zr-MOF showed good adsorption of CO<sub>2</sub> and the capacity depended on temperature. At 1 atm, 0 °C, CO<sub>2</sub> adsorption could reach 4.46 mmol/g, which is higher than that on Zr-MOF (3.52 mmol/g)[17]. This could be attributed to the presence of basic amino groups in porous structure. In addition, the adsorption of CO<sub>2</sub> at 23 °C was reduced and the capacity was determined as 2.85 mmol/g. Heat of adsorption was calculated using the Clausius-Clapeyron equation ( $dp/p = \Delta H dT/RT^2$ ) and an average value from different CO<sub>2</sub> coverage on amino-Zr-MOF was obtained as 29.4 kJ/mol. Table 2 presents the heat of CO<sub>2</sub> adsorption on various amino-functionalised MOFs. The value of amino-Zr-MOF is reasonable in comparison with other amino-functionalised MOFs [27]. The relatively higher values of the heat of CO<sub>2</sub> adsorption is due to amino groups which could create an electric field inside the pores against more polarisable adsorbates [28]. In addition, the free primary amine present in amino-Zr-MOF would form a carbamate with the CO<sub>2</sub>, resulting in high energy.

**[Insert Fig.6]**

**Table 2. Heat of CO<sub>2</sub> adsorption on different amino-functionalised MOFs.**

Sample	CO <sub>2</sub> Heat of adsorption(kJ/mol)	Reference
IRMOF-3 DEF	20	[25]
Amino-MIL-53(Al)	38	[25]
Amino-MIL-101(Al)	28	[26]
Zr-MOF	28	[29]
Amino-Zr-MOF	29.3	This work

Amino-Zr-MOF was also tested at high pressure up to 980 kPa at 0 °C for CO<sub>2</sub> and CH<sub>4</sub> adsorption and the adsorption isotherms are presented in Fig.7. As seen, CH<sub>4</sub> and CO<sub>2</sub> adsorption increased with increasing pressure and CO<sub>2</sub> adsorption was higher than CH<sub>4</sub>. The adsorption values were 9.04 and 3.73 mmol/g for CO<sub>2</sub> and CH<sub>4</sub>, respectively at 980 kPa. The ideal selectivity of CO<sub>2</sub> to CH<sub>4</sub> was obtained based on ratio of the equilibrium adsorption at varying pressures and is shown in Fig.7. It is seen that the ideal selectivity of CO<sub>2</sub>/CH<sub>4</sub> on

amino-Zr-MOF decreased with increasing pressure and it ranged from 4.5 to 2.35 over the range of pressure between 0.3 and 9 atm, 0 °C.

**[Insert Fig.7]**

From adsorption column tests shown in Fig. 8, it was proved that this material had a good capacity to adsorb carbon dioxide in real conditions. The breakthrough time changed with the variation of CO<sub>2</sub> concentration. It was 40, 31, 20 min at 10%, 15% and 20% CO<sub>2</sub>, respectively, in CO<sub>2</sub>-N<sub>2</sub> gas mixture passing through the adsorption column. From the breakthrough curve it was shown that the slope of the curve increased sharply and with increase in CO<sub>2</sub> concentration, suggesting no diffusion limitations in the material during the adsorption. Also, the high slope of these curves may refer to the possibility of adsorbent regeneration directly after reaching saturation, which can be a good indication toward reducing the cost of gas regeneration [30]. By the integration of the curves, total adsorbed amount of CO<sub>2</sub> within the column during the breakthrough time could be calculated as around 4.55, 5.26 and 4.37 mmol/g at 10%, 15% and 20%CO<sub>2</sub>, respectively.

The adsorption capacities obtained at dynamic tests at 10% and 20%CO<sub>2</sub> are in a good agreement with the value obtained from the static volumetric adsorption (4.46 mmol/g) whereas the value is higher at 15% CO<sub>2</sub>. The high CO<sub>2</sub> adsorption obtained at 15%CO<sub>2</sub> was probably attributed to the dynamic exchange equilibrium of CO<sub>2</sub> with N<sub>2</sub>. Before CO<sub>2</sub> adsorption, Zr-MOF column was saturated with N<sub>2</sub>. At lower CO<sub>2</sub> concentration, the exchange of N<sub>2</sub> by CO<sub>2</sub> may take longer time, resulting in a fast effluent of CO<sub>2</sub> and overestimation of CO<sub>2</sub> adsorption at 15%CO<sub>2</sub>.

**[Insert Fig.8]**

#### **4. Conclusion**

An amino-Zr-MOF was synthesised with methanol activation and heating treatment at temperature below 300 °C. It was found that activating temperature played an important role for the porous structure of amino-Zr-MOF. The amino-Zr-MOF exhibited higher thermal stability and higher CO<sub>2</sub> adsorption than CH<sub>4</sub>. The CO<sub>2</sub>/CH<sub>4</sub> ideal selectivity decreased with increasing pressure. Static isothermal measurement and column tests showed that CO<sub>2</sub> adsorption at 1 atm is around 5 mmol/g. At higher pressure of 10 atm, CO<sub>2</sub> adsorption could reach 9 mmol/g.

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## List of Figures

Fig.1 FTIR spectra of amino-Zr-MOFs (a), magnification of the band at 2600-3800  $\text{cm}^{-1}$  of as-synthesised amino-Zr-MOF (b).

Fig. 2 TGA profiles of amino-Zr-MOF activated by methano and heating under 200 and 300  $^{\circ}\text{C}$ .

Fig.3 XRD patterns of amino-Zr-MOFs activated at different temperatures.

Fig.4  $\text{N}_2$  adsorption isotherm and pore size distributions of amino-Zr-MOFs activated at different temperatures.

Fig.5 SEM image of activated amino-Zr-MOF.

Fig.6  $\text{CO}_2$  adsorption on amino-Zr-MOF at different temperatures.

Fig.7  $\text{CO}_2$  and  $\text{CH}_4$  adsorption on amino-Zr-MOF and ideal selectivity.

Fig.8 Breakthrough curves of  $\text{CO}_2$  adsorption at different  $\text{CO}_2$  concentrations.

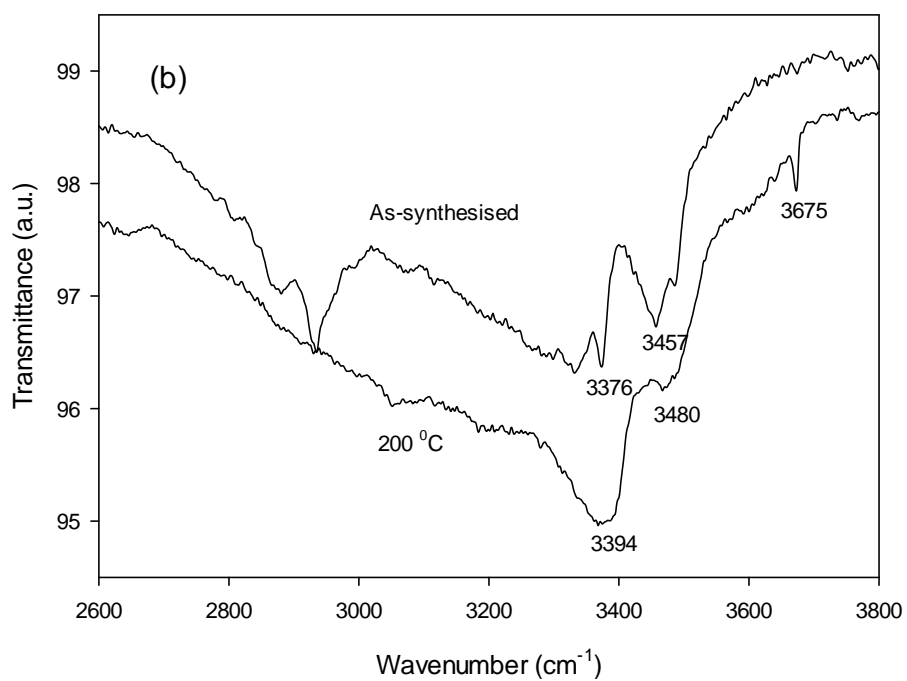
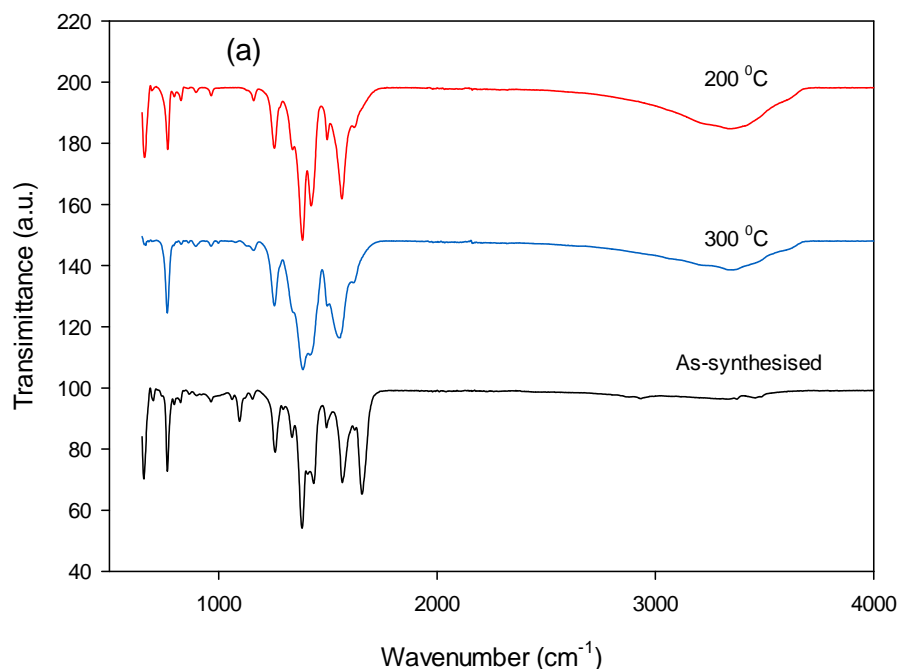


Fig.1 FTIR spectra of amino-Zr-MOFs (a), magnification of the band at 2600-3800  $\text{cm}^{-1}$  of as-synthesised amino-Zr-MOF (b).

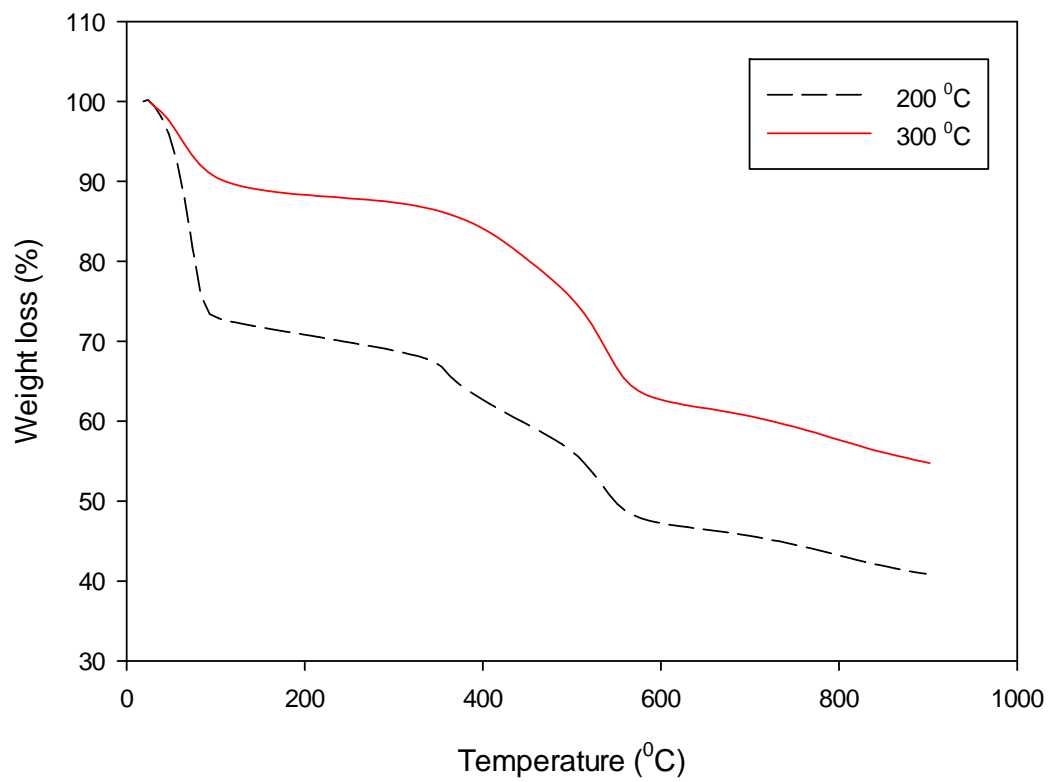


Fig. 2 TGA profiles of amino-Zr-MOF activated by methano and heating under 200 and 300 °C.

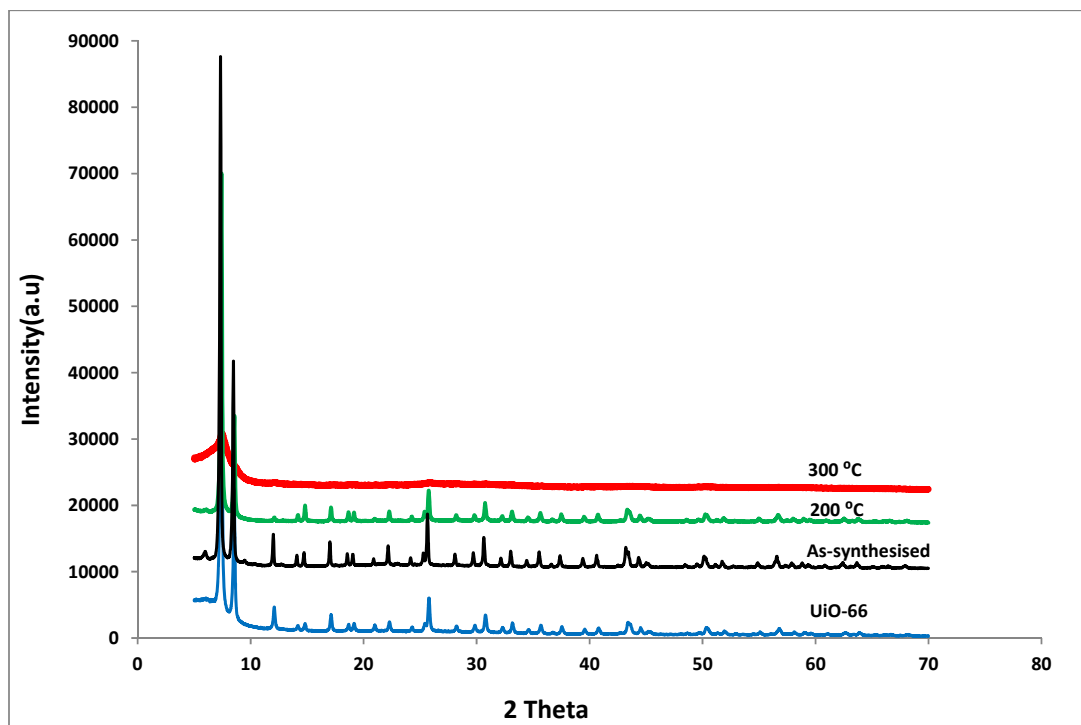


Fig.3 XRD patterns of amino-Zr-MOFs activated at different temperatures.

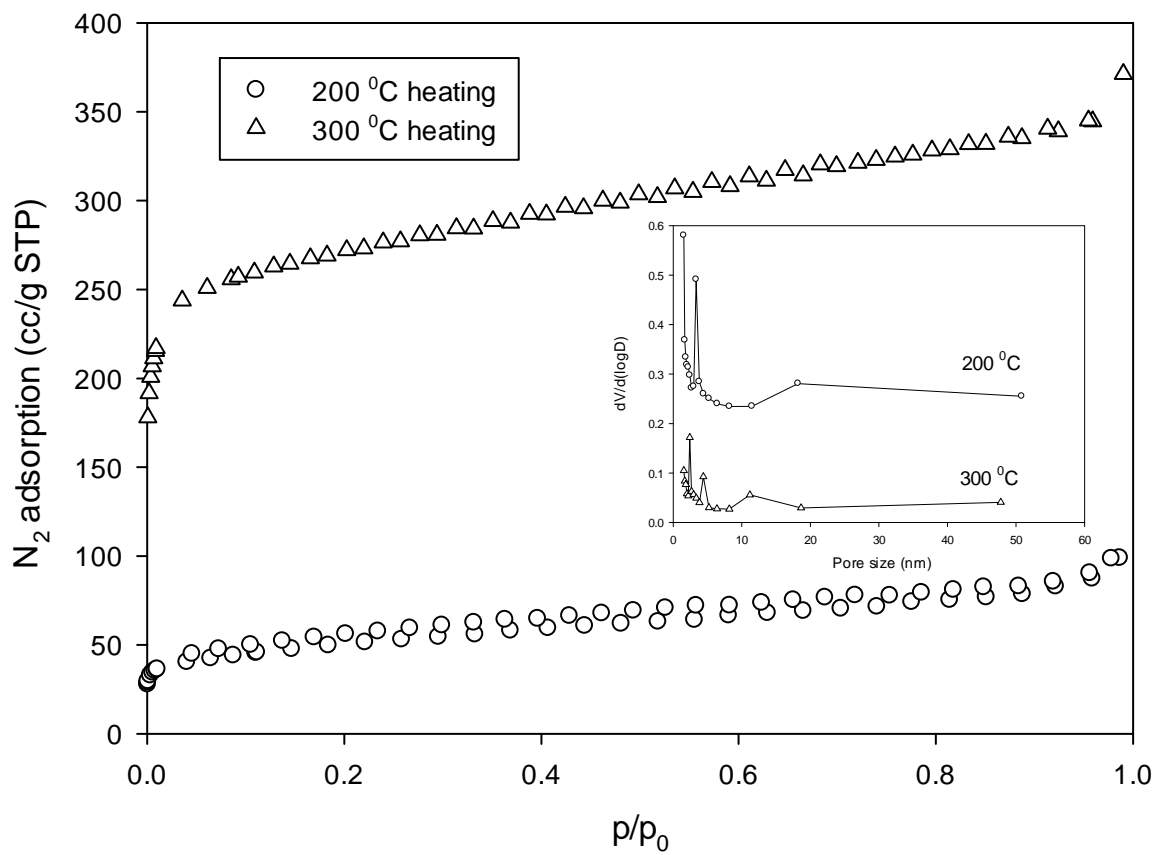


Fig.4  $N_2$  adsorption isotherm and pore size distributions of amino-Zr-MOFs activated at different temperatures.



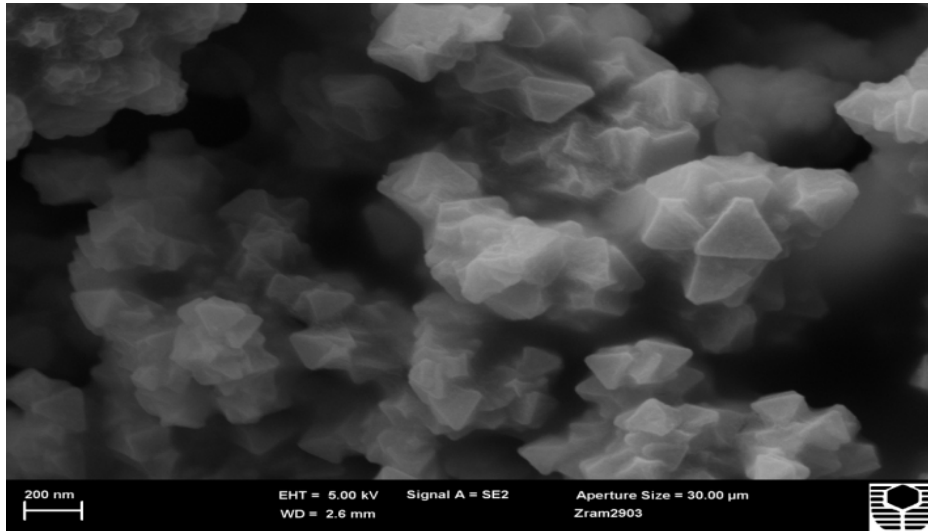


Fig.5 SEM image of activated amino-Zr-MOF.

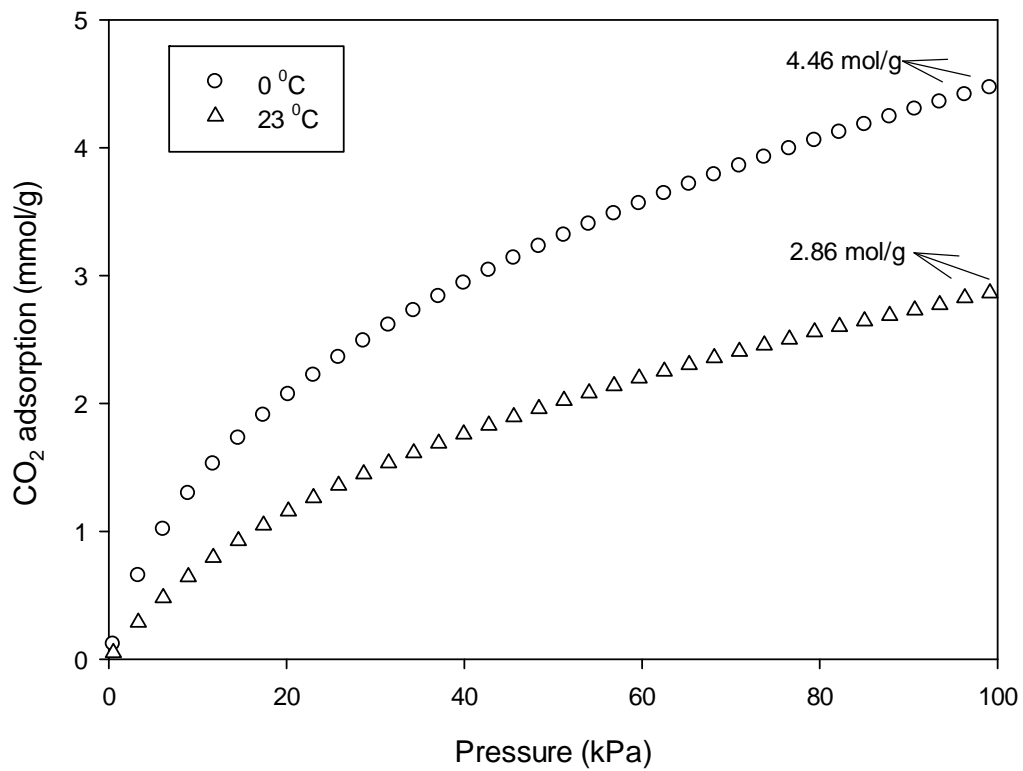


Fig.6 CO<sub>2</sub> adsorption on amino-Zr-MOF at different temperatures.

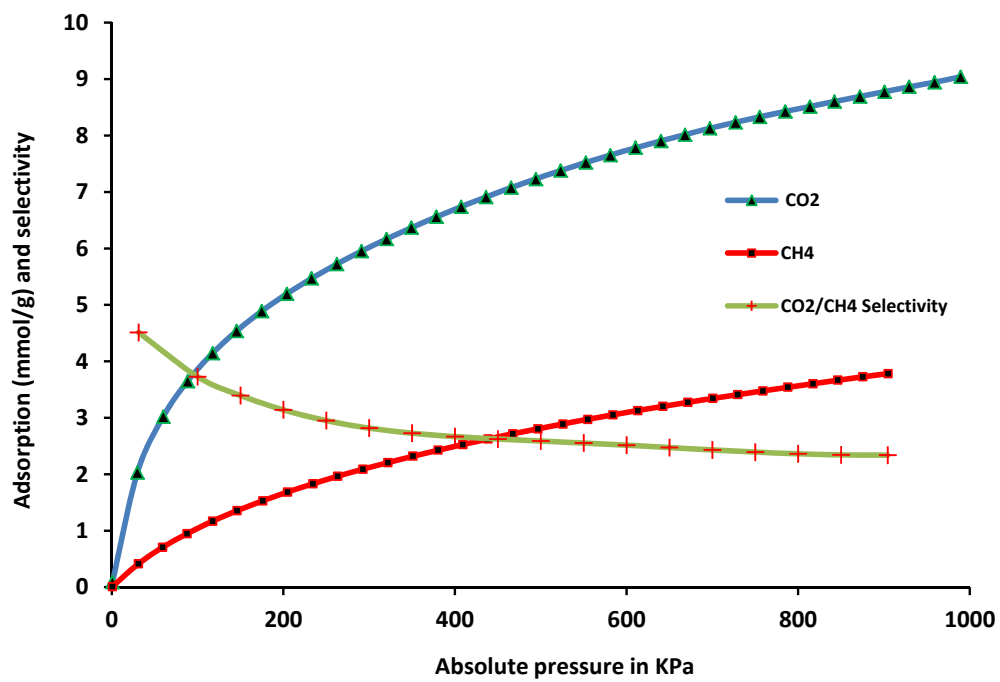


Fig.7 CO<sub>2</sub> and CH<sub>4</sub> adsorption on amino-Zr-MOF and ideal selectivity.

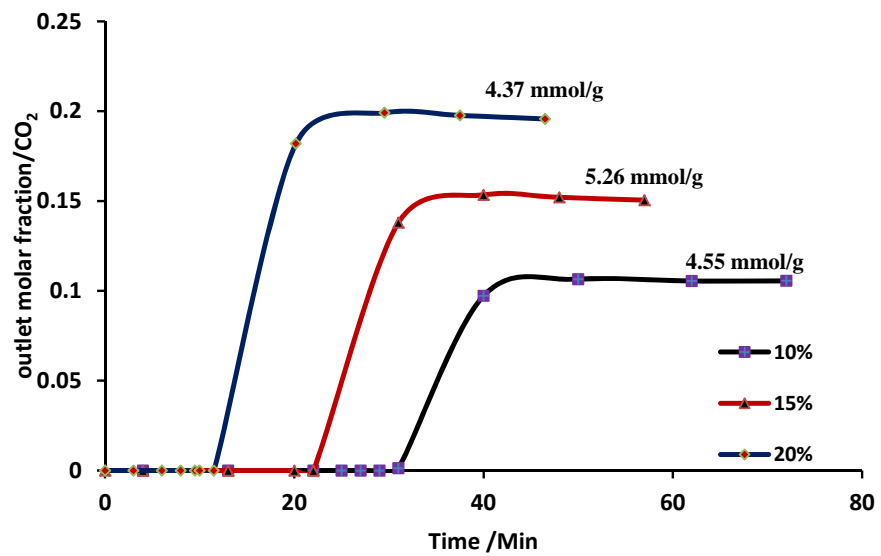


Fig.8 Breakthrough curves of CO<sub>2</sub> adsorption at different CO<sub>2</sub> concentrations.