

# 1 **Microelectrode arrays with active-area geometries defined by spatial light modulation**

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10  
11 **ABSTRACT.** Microelectrode arrays form the basis of electrochemical sensing devices because  
12 of their unique properties, such as enhanced mass transport and steady-state diffusion currents.  
13 However, they demand a predefined and rigid geometry, and require a connecting pad for each  
14 element of the array. Here it is reported the formation of microelectrode arrays whose geometry  
15 is defined by the shape of a light pattern projected on an unstructured silicon electrode.  
16 Spatiotemporally resolved fluxes of charge carriers are used to confine a model electrochemical  
17 reaction only to the illuminated areas. Using spatial light modulators, microelectrode geometry  
18 is adjusted instantaneously, at will, on a homogeneous semiconductor electrode carrying a  
19 single electrical connection. By developing a theoretical model to analyze the current–potential  
20 data, it is revealed within which limits spatial light modulation can be used to enhance, on  
21 silicon, the mass transport of a diffuse redox system.

22  
23 **Keywords:** microelectrode arrays; semiconductor electrochemistry; spatial light modulation;  
24 light addressable electrochemistry; monolayers on silicon

## 25 26 **1. Introduction**

27 Spatial light modulators (SLM) are devices that impose some form of spatial modulation to a  
28 light beam. Originally developed as video projection devices for entertainment purposes, SLM  
29 are now used in areas as diverse as photolithography [1], microscopy [2], 3D printing [3], and  
30 micromanipulation [4]. More recently, SLM have found applications in electrochemical  
31 research: a spatially modulated light beam can confine electrochemical reactions at a specific

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1 microscopic location within a macroscopic semiconductor–electrolyte interface [5]. Light  
2 generates charge carriers in the semiconductor, and these are driven by the space-charge electric  
3 field towards redox species present at the solid–liquid interface, hence localizing the redox  
4 process in correspondence to the illuminated area (Fig. 1A) [5]. This principle served as  
5 platform to develop mask-free lithography techniques [6-8], and has been successfully applied  
6 to interrogate the redox activity of a surface [9, 10], to develop surface chips for the capture of  
7 rare cells [11], to generate chemical gradients [12], and as a light-driven DNA sensor [13].  
8 However, no reports are yet available on methods to harness spatially structured light to  
9 generate transient microelectrode arrays on readily available (unstructured) semiconductive or  
10 photoconductive surfaces.

11 Microelectrode arrays, owing in large part to developments in the technology required for the  
12 fabrication of integrated circuitry, are now readily available, relatively inexpensive, and the  
13 core of several chemical sensing devices [14, 15]. They offer the advantages of miniaturized  
14 electrodes, such as fast response time and increased sensitivity [16], and when multiple units  
15 are organized in an array they open up to multi-analyte sensing and mapping of a redox  
16 environment [14, 15]. However, microelectrode arrays are not free from constrains: a  
17 considerable amount of space is lost between conductive pads or to fit bonding wires and, most  
18 importantly, the geometry of the array is fixed and predetermined by a choice made prior to the  
19 experiment.

20 Here we begin to address some of these shortcomings by exploring the use of spatially  
21 modulated light to define the active area of multiple microscopic sites across an unstructured  
22 silicon electrode. We explore the parallel aspects of light activated electrochemistry,  
23 specifically testing the possibility of generating microelectrode arrays that have no bonding  
24 wires, no conductive pads, and of unrestricted and dynamic geometry. Microelectrode arrays  
25 are transiently formed by projecting light patterns on a silicon–electrolyte interface using a  
26 ferroelectric liquid crystal on silicon micromirror device as SLM. Our current setup allows

[Type here]

1 projecting visible light patterns with a spatial resolution of 4.5  $\mu\text{m}$  (Fig. 1B). Specifically, we  
2 have studied the voltammetric current–potential behaviour of macroscopic electrodes, with a  
3 single electrical connection, where only microscopic band-shaped regions are illuminated. We  
4 demonstrate experimentally, and model theoretically, to what extent increasing the width of  
5 dark regions can fine-tune the overlap between adjacent diffusion fronts, hence defining the  
6 conditions under which cyclic voltammograms show a progressive transition from macro- to  
7 micro-electrode behaviour. Finally, by analysing two silicon allotropes we demonstrate that the  
8 extent of electrochemical confinement depends strongly on charge carriers' diffusion lengths.

## 10 **2. Experimental**

### 11 **2.1 Chemicals and materials**

12 Unless otherwise specified chemicals were of analytical grade and used as received. Sulphuric  
13 acid (Sigma-Aldrich, Puranal, 95–97%), ammonium fluoride (Sigma-Aldrich, Puranal, 40%),  
14 hydrofluoric acid (Sigma Aldrich, 48%), and 1,8-nonadiyne (Sigma-Aldrich, 98%), were used  
15 in wafer cleaning, etching, and hydrosilylation procedures. Milli-Q water with a resistivity  
16  $>18.2 \text{ M}\Omega \text{ cm}$ , redistilled dichloromethane and 2-propanol were used for surface modification  
17 procedures and washings. Ferrocene (Sigma-Aldrich, 98%), tetrabutylammonium perchlorate  
18 (Sigma-Aldrich, 98%) and acetonitrile (Ajax Finichem, 99.7%) were used to prepare  
19 electrolytic solutions. Copper sulphate (Chem-Supply, 99.5%) and potassium sulphate (Ajax  
20 Finichem, 99%) were used to prepare the electrolytic solutions used for patterning experiments.  
21 Gallium–indium eutectic (Sigma Aldrich, 99.99%) was used to form an ohmic contact to the  
22 silicon electrode. Single crystal silicon wafers, 500  $\mu\text{m}$  thick, were purchased from Siltronix  
23 (Archamps, France). The wafers were n-type (phosphorous-doped) Si(111), and p-type (boron-  
24 doped) Si(100), with a nominal resistivity of 8–12  $\Omega \text{ cm}$  unless otherwise stated.

### 25 **2.2 Preparation of the amorphous silicon substrates**

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1 Highly doped (0.001–0.003  $\Omega$  cm) p-type (boron-doped) Si(100) wafers were cleaned in an  
2 ozone chamber immersed for 3 min in a 5% hydrofluoric acid solution and then blown dry  
3 under a nitrogen flow. The amorphous silicon film was grown on Si(100) by plasma-enhanced  
4 chemical vapour deposition until reaching a thickness of 1  $\mu$ m.

### 5 **2.3 Surface functionalization**

6 All silicon samples were chemically modified according to the following procedure. Substrates  
7 were rinsed with dichloromethane, 2-propanol and water, immersed for 10 min in a  
8 deoxygenated 40% aqueous  $\text{NH}_4\text{F}$  solution, washed with water and dichloromethane and then  
9 blown dry under a nitrogen flow. A few drops of 1,8-nonadiyne were then placed on the top of  
10 the sample, which was immediately contacted with a quartz slide to limit the evaporation of the  
11 diyne. Samples were then kept for 2 h under nitrogen in front of a 312 nm light source (Vilber,  
12 VL-215.M, 30 W nominal power output) positioned approx. 20 cm away from the sample.  
13 Samples were rinsed thoroughly with dichloromethane before analysis.

### 14 **2.4 Direct measurement of electrochemical confinement**

15 The extent to which the electrochemical reactions were confined in 2D to match the illumination  
16 pattern was assessed by scanning electron microscopy (SEM) analysis of  $\text{Cu}_2\text{O}$  patterns  
17 electrochemically deposited on silicon photocathodes. In brief, a line-shaped feature of known  
18 width was projected on a diyne-modified silicon substrate while applying a negative potential  
19 of  $-0.5$  V (vs Ag/AgCl,  $\text{KCl}_{\text{sat}}$ ). The electrolytic solution contained  $\text{Cu}_2\text{SO}_4$  ( $50 \times 10^{-3}$  M) and  
20  $\text{K}_2\text{SO}_4$  (0.5 M). The amount of charge transferred per unit of area was kept constant at 1  $\mu\text{C}$   
21  $\mu\text{m}^{-2}$ . SEM images of the  $\text{Cu}_2\text{O}$  patterns were acquired on a Zeiss Neon 40EsB FESEM  
22 equipped with a Schottky field emission gun operating at 5 kV and a chamber pressure of ca. 4  
23  $\times 10^{-6}$  mbar.

### 24 **2.5 Voltammetry simulations**

25 The analytical model developed to simulate cyclic voltammograms is based on that of  
26 Santangelo *et al.* [17], but expanded to account for the microelectrode and band geometries.

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1 This model is described in detail in Section S2. Simulation experiments were written and  
2 performed in GFortran and Mathcad 14.

3

### 4 **3. Results and discussion**

#### 5 **3.1. Electrochemical switching by a light stimulus**

6 Proving the formation of transient microelectrode arrays upon local illumination is the main  
7 purpose of this study. Therefore, we first proceeded to investigate the experimental conditions  
8 where a light stimulus becomes a “switch” of the electrode kinetics. To prevent anodic  
9 decomposition of the electrode, and to allow for reproducible electrochemical measurements  
10 across a wide potential window (*vide infra*), hydrogen-terminated silicon substrates were  
11 modified with a monolayer of 1,8-nonadiyne [18], which prevents silicon oxidation under  
12 positive voltages [19], but still allows for efficient electron transfer [7]. The  
13 ferrocene/ferrocenium (Fc/Fc<sup>+</sup>) redox couple was chosen for its ideal and well-known  
14 electrochemical behaviour. In order to ensure minority carriers depletion under dark and at a  
15 potential required for a fast Fc oxidation, we selected lowly doped (LD) n-silicon with a flat  
16 band potential more negative than the formal redox potential of the Fc/Fc<sup>+</sup> couple. The formal  
17 redox potential of the Fc/Fc<sup>+</sup> couple, measured on highly doped (HD) p-type Si(111) electrodes  
18 (*i.e.* metal-like electrodes), is around 0.2 V (vs Ag/AgCl/sat. KCl, Fig. S1A). The LD n-type  
19 Si(111) substrates have a flat band potential of -0.3 V (vs Ag/AgCl/Sat. KCl, Fig. S1B).  
20 Therefore under dark the LD n-type Si(111) electrode is depleted of charge carriers at the  
21 potential where Fc oxidation dominates the redox process. When the semiconductor is  
22 illuminated, electron-hole pairs are generated in the solid and holes, driven both by the space-  
23 charge layer (SCL) electric field and by a concentration gradient, are trapped by Fc molecules  
24 in solution, Fig. 2A. In order to confirm this, the potential was scanned in cyclic voltammetry  
25 experiments performed both under dark and under red light (Fig. 2B, symbols and solid line,  
26 respectively). Data in Fig. 2B reveal the flow of an electrochemical current only when the

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1 semiconductor is illuminated, indicating the redox reaction occurs at an appreciable rate only  
2 upon electrode illumination.

3

### 4 **3.2 Cyclic voltammetry of freely diffusing redox species on a photoanode**

5 The main characteristic of a microelectrode is its radial diffusion profile. This is reflected on an  
6 enhanced mass transport of diffusible redox species, which on a metal electrode results in a  
7 steady-state electrochemical current [20, 21]. However, the electrochemical behaviour of  
8 semiconductor electrodes differs substantially from that of metal electrodes [17, 22-25]. The  
9 potential drop across a semiconductor–liquid interface is distributed between the diffuse layer  
10 (DL) and the SCL, the latter extending inside the semiconductor (Fig. 2A). When an external  
11 potential is applied across the interface, a large fraction of this potential can drop inside the  
12 SCL. In contrast, in a metal any applied bias drops exclusively across the diffuse layer [17, 22-  
13 25].

14 Because of these differences in the potential profile along the interface, the voltammetric  
15 responses of semiconductors and metals differ considerably [17, 22-25]. We therefore  
16 developed a theoretical model based on that of Santangelo *et al.* [17] in order to describe, and  
17 analyze, voltammetric photocurrent–potential curves of our silicon electrodes (see Section S2,  
18 for a full description of the model). Our model differs from that of Santangelo *et al.* in that it  
19 takes into account the geometry of the electrode, allowing for the interpretation of voltammetric  
20 responses of microelectrodes and microelectrode arrays.

21 Fig. 2C shows the theoretical influence of photocurrent magnitude on the voltammetric  
22 responses expected for freely diffusing redox species on a photoanode. At low photocurrent  
23 values, a steady state current is observed in the forward scan because the reaction kinetics is  
24 limited by the availability of photogenerated carriers. A stationary state arises because all  
25 photogenerated carriers are being consumed by the available electroactive species. This  
26 situation is for example experimentally observed when the light intensity is sufficiently low

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1 (Fig. 2D). However, the same response could also arise when the illuminated area gets  
2 sufficiently small (*vide infra*). Therefore, one has to exert caution when trying to interpret  
3 steady state currents at photoelectrodes. In fact, the voltammetric curves at low photocurrent  
4 values resemble that of a metal electrode under radial or convective mass transport, such in the  
5 case of a microelectrode (Fig. S2) or a rotating disk electrode, respectively [20]. Because our  
6 principal aim is to demonstrate the formation of microelectrode arrays on semiconductors, a  
7 correct interpretation and analysis of the current–potential relationship is fundamental.

### 8 9 **3.3 Voltammetric photocurrent–potential behaviour of freely diffusing Fc/Fc<sup>+</sup> on Si(111)** 10 **electrodes under local illumination**

11 The digital simulations of Fig. S2 show voltammetric curves of diffusing redox species on a  
12 metallic single band microelectrode as a function of the width of the band ( $w$ ). As the band gets  
13 narrower, simulations anticipate a semi-stationary current and a progressive increase in current  
14 density. We then performed digital simulations for a single band photoanode microelectrode,  
15 again as a function of  $w$  (Fig. 3A and Section S2). Similarly to a metal electrode, there is an  
16 increase in current density as  $w$  decreases. However, a steady state current emerges when the  
17 band is sufficiently broad, a situation exactly opposite to a metal electrode. As pointed in the  
18 previous section, this steady state is a consequence of the electrode kinetics being limited by  
19 the supply of photogenerated carriers, and cannot be assigned unambiguously to a mass  
20 transport effect. Simulated cyclic voltammetry curves on band microelectrode photoanodes as  
21 a function of the photocurrent are shown in Fig. 3B. Similar to the macroelectrode photoanodes,  
22 the current density is proportional to the photocurrent, and a steady state is observed at low  
23 photocurrents.

24 Having developed the theoretical framework that allows us to interpret electrochemical  
25 responses of freely diffusing redox species on semiconductor microelectrodes, we then  
26 investigated the experimental response for the Fc/Fc<sup>+</sup> couple on LD Si(111) electrodes under

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1 local illumination. This is to demonstrate the formation of transient microelectrodes arrays with  
2 a structured light stimulus on a homogeneous substrate with a single electrical connection. A  
3 custom-built SLM device was used to project light micropatterns at the silicon–electrolyte  
4 interface (Fig. 1B). The illuminated area is expected to act as a transient electrode, able to  
5 confine in space an electrochemical reaction and to enhance mass transport. Fig. 3C shows  
6 cyclic voltammetry curves obtained when projecting lines of variable width. We chose a band  
7 geometry for our transient electrodes because it allows a broader range of currents, up to  
8 microamperes, while still retaining the microelectrode response [15]. Similar to the simulated  
9 voltammograms, the experimental curves of Fig. 3C show an increase in anodic current density  
10 as  $w$  gets narrower. However, and opposite to the simulated response (Fig. 3A), a steady state  
11 current is observed when  $w$  decreases and the cathodic current also increases. This striking  
12 contrast between simulations and experiments is probably caused by the assumption that the  
13 active electrode area is an exact match of the illumination pattern. However, equating the active  
14 area to the illuminated area is an approximation valid only when the diffusion length of the  
15 photogenerated carriers ( $L$ ) is much smaller than the width of  $w$  (see schematics in Fig. 3D). If  
16  $L$  is of comparable size, or bigger than  $w$  (Fig. 3D, right panel), one has to take into account the  
17 contribution of the photogenerated carriers diffusion length to the electrochemical current. The  
18 direct consequences of photogenerated carriers diffusing significantly outside the illuminated  
19 areas are two: 1) it leads to a wrong estimate of the active electrode area (and therefore of the  
20 current density), and 2) a distortion of the voltammetric curves. This latter point (2) is due to  
21 the anisotropic character of carriers' diffusion, and their concentration decreasing with the  
22 square of the distance, as does the local photocurrent. Consequently, the microelectrode  
23 response in Fig. 3C does not directly arise nor match the actual geometry of the illumination  
24 pattern.

25 In order to avoid the undesirable distortion effects mentioned in the above paragraph, we tried  
26 to adjust the overlap and separation of diffusion fronts occurring on microelectrode arrays of

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1 variable density but keeping the active area constant [21]. This can be done by generating band  
2 electrodes where the inter-band gap is systematically varied. When the bands approach each  
3 other, their diffusion fronts overlap (Fig. 4A), and the electrochemical current decreases (digital  
4 simulations of a semiconductor band microelectrode array, Fig. 4B and Section S2). In this way,  
5 the illuminated area is kept constant so that an estimation of the current density based on the  
6 active area is not necessary and effects arising from the diffusion of charge carrier effects are  
7 kept constant. By projecting band electrode patterns with different band separations as shown  
8 in Fig. 4A we were able to determine and quantify diffuse effects on our LD Si(111)  
9 photoanodes. Fig. 4C shows identical voltammetric experimental responses for an array of 8  
10 bands (22.5  $\mu\text{m}$  wide) where the inter-band gap was varied between 225 and 0  $\mu\text{m}$ . A low  
11 photocurrent could limit the anodic current, but the cathodic current should show a variation  
12 with the inter-band separation current, see Fig. 4D. Further decreasing the  $w$  (13.5  $\mu\text{m}$ ) did not  
13 result in a microelectrode behaviour, Fig. S3. Therefore, as discussed above, the most likely  
14 explanation is that the electrochemical reaction is not confined exclusively in correspondence  
15 of the projected light pattern. This experiment also proves that the current difference between  
16 bands of different width shown in Fig. 3C can be unambiguously assigned to photoeffects  
17 arising when the illuminated area is decreased. The larger current density for the narrower bands  
18 arises because a wrong estimation of the electrode active area and a large  $L/w$  ratio, and a  
19 plateau is observed because the current is limited by the photogenerated charge carriers (*i.e.* is  
20 not directly related to a radial diffusion front).

21

### 22 **3.4 Lateral diffusion of carriers as function of diffusion lengths**

23 To confirm our hypothesis, that is, minority carriers diffusion preventing us from observing  
24 truly diffuse effects on light-activated microelectrodes, we next quantified the extent of  
25 electrochemical confinement on functionalized LD Si(111) electrodes. This was done by  
26 photoelectrochemically depositing patterns of  $\text{Cu}_2\text{O}$ :  $\text{Cu}^{2+}$  ions are reduced to  $\text{Cu}^+$  only where

[Type here]

1 photogenerated charge carriers reach the interface, and cuprous ions then deposit as  $\text{Cu}_2\text{O}$   
2 crystals [6, 7]. By means of imaging (ex-situ) the  $\text{Cu}_2\text{O}$  patterns, this method can be used to  
3 directly determine the mismatch between the electrochemically active area and the light pattern.  
4 For these photoelectrochemical reduction experiments we used chemically functionalized LD  
5 p-type Si(111) surfaces as a photocathode. The cyclic voltammograms of Fig. 5A show that the  
6  $\text{Cu}^{2+}$  reduction to  $\text{Cu}^+$  occurs only under illumination. This is because the  $\text{Cu}^{2+}$  reduction  
7 potential lies in the depletion layer region, that is at potentials below the electrode flat-band  
8 potential ( $E_{fb} = -0.25$  V vs Ag/AgCl/KCl sat., Mott-Schottky plot in Fig. 5B and in more details  
9 in Fig. S4). We projected ‘line’ patterns of different widths and compared the known width of  
10 the projected band with the width of the  $\text{Cu}_2\text{O}$  deposit estimated from scanning electron  
11 micrographs (Figs. 5C–E). As expected, the  $\text{Cu}_2\text{O}$  features extend several micrometres beyond  
12 the illuminated region, supporting a non-negligible lateral diffusion of minority carriers. A  
13 diffusion length of ca. 30–45  $\mu\text{m}$  was roughly estimated by subtracting the width of the  
14 projected light feature from the actual pattern width. This value is in good agreement with  
15 previously reported values for crystalline silicon [26]. When digital simulations of cyclic  
16 voltammetry on band microelectrodes are performed taking into account this correction to the  
17 active area (Fig. S5), their agreement with the experimental results improves significantly (Fig.  
18 4C).

19 To reinforce on this point – the importance of the  $w/L$  ratio in the performance and analysis of  
20 light activated microelectrodes – we tested photoelectrodes with a significantly shorter  
21 diffusion length. Amorphous silicon is a silicon allotrope with a diffusion length not exceeding  
22 few nanometres [27]. Cyclic voltammetry reveals the electrodeposition on amorphous silicon  
23 happens only under illumination (Fig. S6), and, most importantly,  $\text{Cu}_2\text{O}$  patterns accurately  
24 match (within 1  $\mu\text{m}$ ) the projected light feature, as shown in the SEM images of Figs. 5F–H,  
25 indicative of the short diffusion length of charge carriers in amorphous silicon.

26

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### 1 **3.5 Microelectrode arrays generated on amorphous silicon electrodes**

2 As we have demonstrated in the previous section, for a material with a sufficiently short  
3 diffusion length, such as amorphous silicon, a photoelectrochemical reaction is confined  
4 satisfactorily within the illuminated area. Moreover, as for LD crystalline wafers, in these  
5 amorphous electrodes the  $\text{Fc}/\text{Fc}^+$  redox reaction happens only under illumination (Fig. S7) [28].  
6 Therefore, amorphous silicon should be an ideal substrate to gain enhanced mass transport by  
7 forming microelectrode arrays with light. Fig. 6A are voltammograms obtained on amorphous  
8 silicon photoelectrodes when projecting arrays of light bands with changes to the inter-band  
9 separation. These data show an increase of the electrochemical current and a progressive loss  
10 of the voltammograms hysteresis as the bands are progressively separated, both indicating an  
11 effective separation of the diffusion fronts, hence a more efficient mass transport to the  
12 electrode leading to a true stationary of quasi-stationary behaviour [15]. Fig. 6B shows the  
13 effect of changes to the scan rate for the band microelectrode array. As expected for  
14 microelectrodes, the scan rate has little effect on the electrochemical current for the case of  
15 isolated microband electrodes since a pseudo-stationary state has been attained under these  
16 conditions. We note that the current–potential curves in Fig. 6A and B show a clear distortion  
17 of the current–potential shape. Moreover, the appearance of a shift towards more positive  
18 potentials, which for the curves in Figs. 6A is of 65 mV for the half-wave potential (see Fig.  
19 S8), is a typical feature of non-reversible redox processes, but finite kinetics factors can be here  
20 discarded due to the practically null shift of the voltammograms with the scan rate (Fig. 6B).  
21 Thus, a cause of this distortion could be ohmic factors since this distortion is more evident as  
22 the current increases (see also Fig. S7 corresponding to a macroelectrode which clearly shows  
23 a distorted voltammogram as compared with an ohmic-free current–potential curve). That is,  
24 although the increase in current can be directly related to a “pure” mass transport effect, a  
25 quantitative evaluation of the observed behaviour should include an ohmic drop contribution to

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1 the applied potential. This refinement to the analytical models will be the subject of a future  
2 work.

3

#### 4 **4. Conclusion**

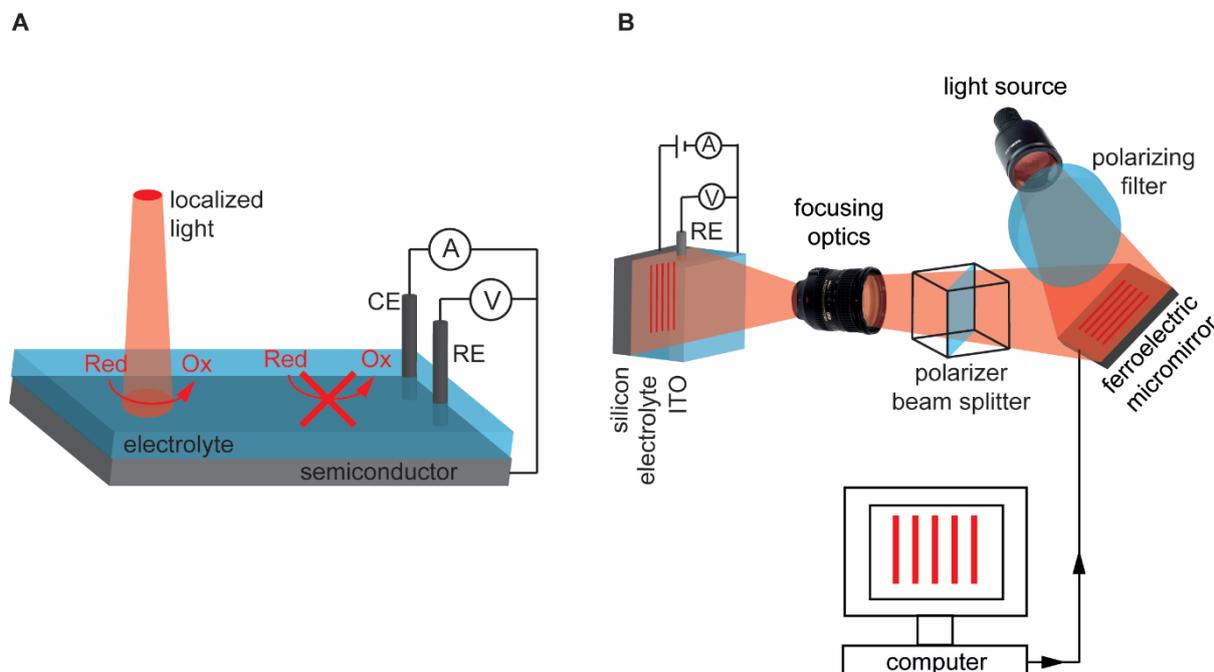
5 We have shown that transient microelectrode arrays can be formed on an unstructured  
6 semiconductor substrate by projecting light patterns of user-defined size and geometry.  
7 Enhanced mass transport to the electrode surface was demonstrated by separating the diffusion  
8 fronts feeding the active areas of transient (light-addressed) microelectrode band arrays. This  
9 method allows to differentiate between true microelectrode response (due to radial mass  
10 transport), and apparent microelectrode response (due to distortion of the voltammograms at  
11 high ratios of diffusion length to width of the illuminated area). The cautionary note is that  
12 steady state currents are not necessarily indicating efficient mass transport, but can instead be  
13 a consequence of the electrode kinetics being limited by the supply of photogenerated carriers.  
14 We have defined the influence of the diffusion length on the photoelectrochemical responses of  
15 these transient microelectrodes by comparing two different silicon allotropes of very different  
16 carriers' diffusion length. Ideal microelectrode responses are achievable in amorphous silicon,  
17 but not in crystalline silicon. This is due to an enhanced electrochemical confinement ensured  
18 in amorphous silicon by a short (few nanometres) diffusion length. We have developed an  
19 analytical model to account for microelectrode geometry and for the splitting of the potential  
20 applied at the semiconductor–electrolyte solution into two contributions: a diode–solution and  
21 a metal–solution potential drops. The theoretical description of the influence of light intensity,  
22 separation between band electrodes and scan rates, qualitatively agrees with the experimental  
23 data, yielding deeper physical insights on the process of guiding electrochemical activity at  
24 semiconductors with spatially structured light.

25 Transient microelectrode arrays created using a spatial light modulator and unstructured  
26 macroscopic photoconductors, or semiconductors, can be a tool for chemical analysis with

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1 performances comparable to those of conventional microelectrodes [29], but with the advantage  
2 of lifting almost entirely geometrical restrictions on the active area.

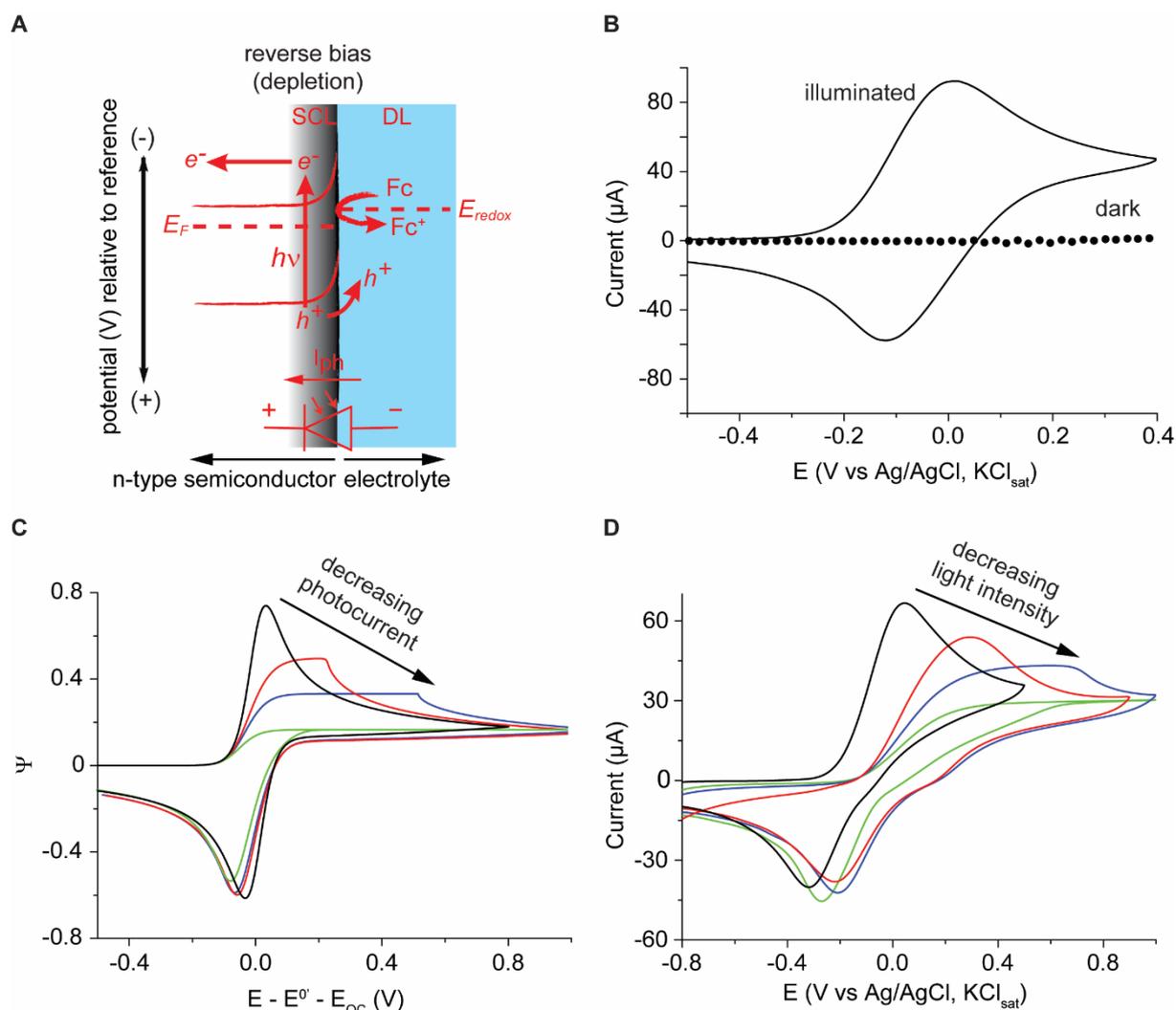
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4  
5 **Fig. 1.** Microelectrode arrays with active-area geometries defined by spatial light modulation  
6 (A) Depiction of the light-activated electrochemistry principle at a depleted semiconductor–  
7 electrolyte interface. (B) Schematics of the ferroelectric liquid crystal on silicon spatial light  
8 modulator (SLM) used to project light micropatterns at the silicon–electrolyte interface.

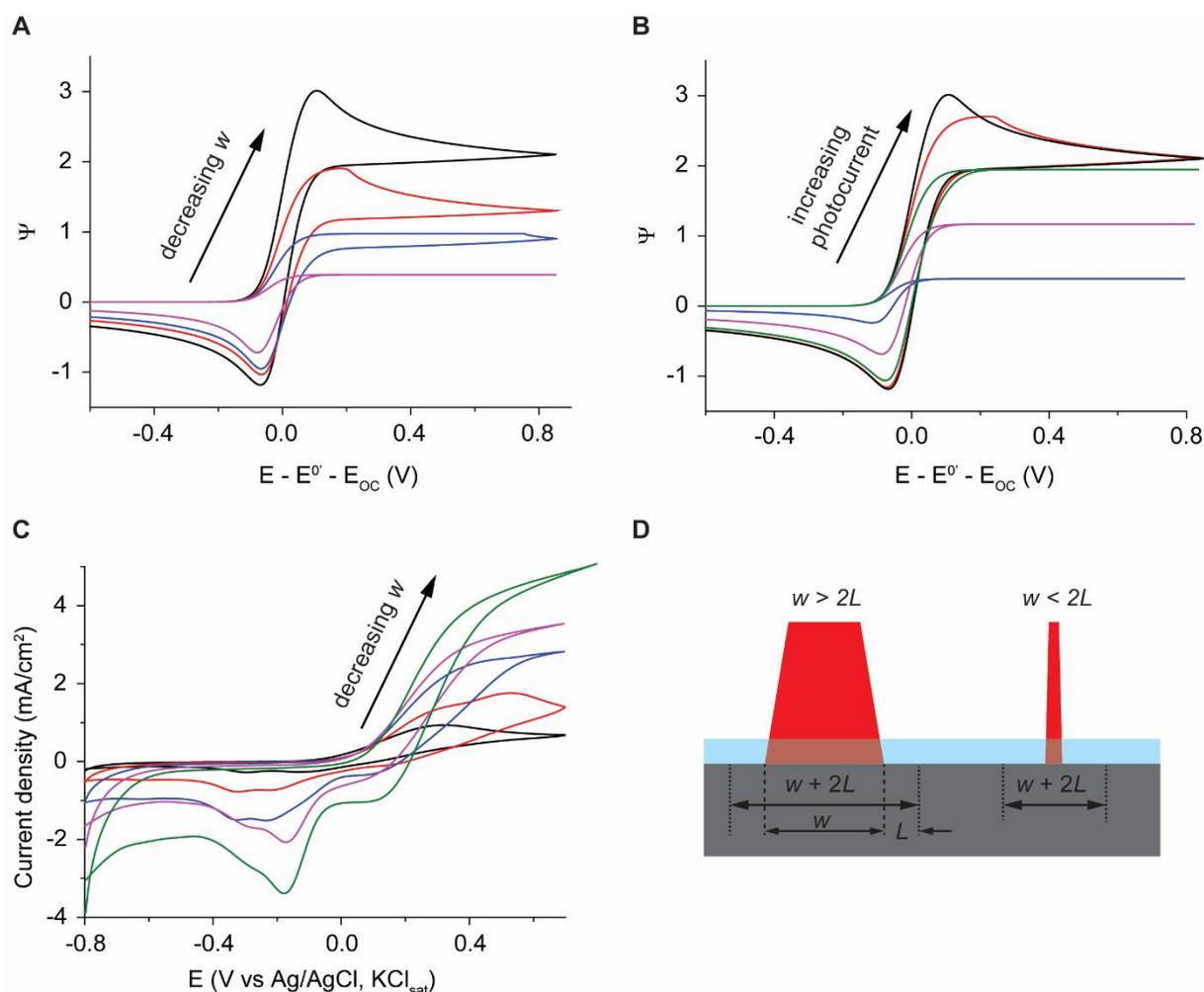
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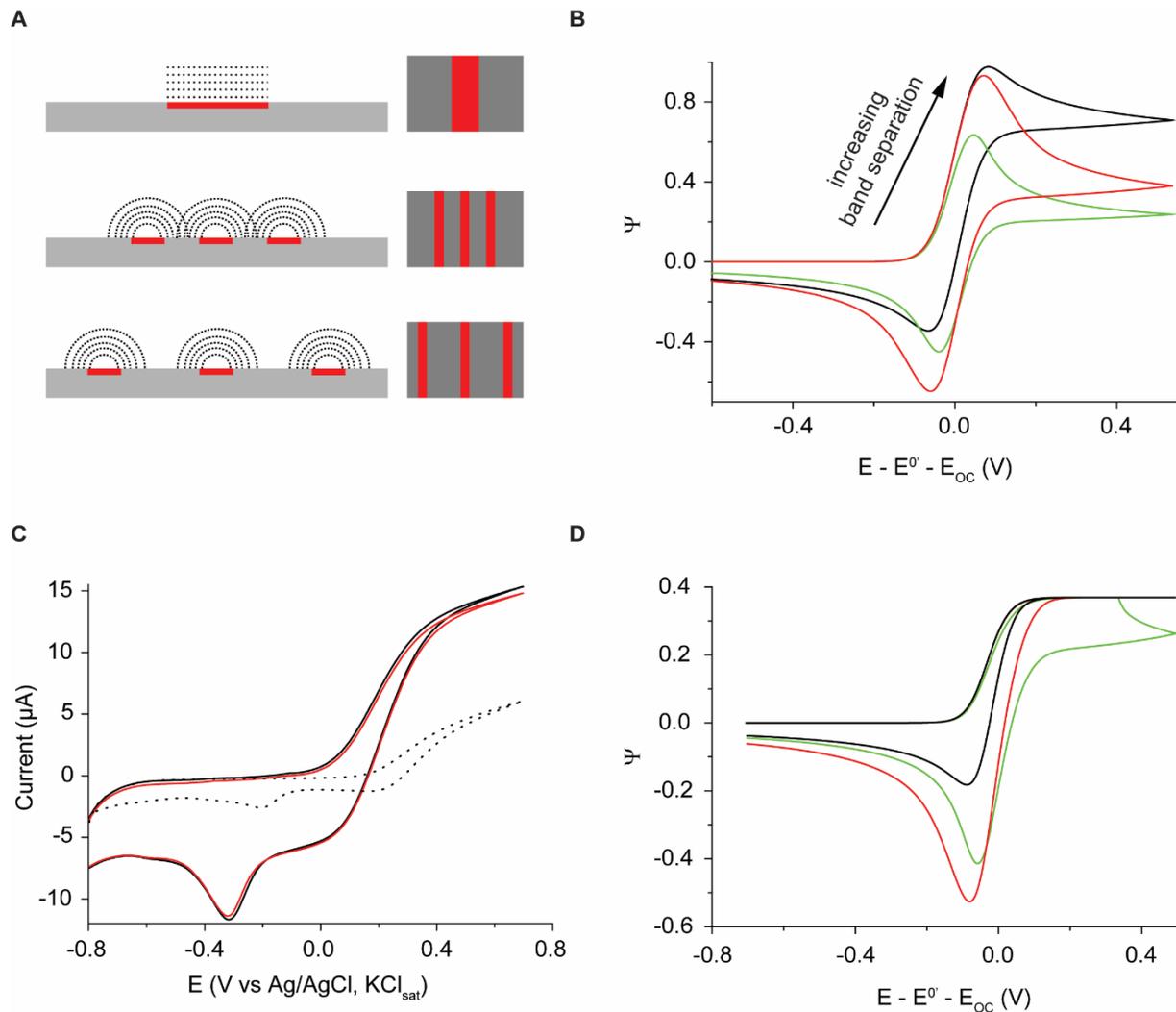
1  
2 **Fig. 2.** Electrochemical switching by a light stimulus and cyclic voltammetry of freely diffusing  
3 redox species on a photoanode  
4 (A) Representation of the potential profile across an n-type semiconductor–liquid interface and  
5 its photoelectrochemistry under reverse bias. (B) Cyclic voltammograms of ferrocene (Fc,  $1.0$   
6  $\times 10^{-3}$  M) under illumination (solid line) and dark (dotted line) acquired on a functionalized LD  
7 n-type Si(111) electrode (0.1 V/s, 0.1 M Bu<sub>4</sub>NClO<sub>4</sub> in acetonitrile). (C) Simulated cyclic  
8 voltammograms (equations (24-25) of the supplementary information, 0.1 V/s) for a  
9 semiconductor photoanode under variable photocurrents ( $I_L = 25, 3, 2$  and  $1$   $\mu$ A, black, red,  
10 blue and green lines, respectively, see Section S2) and assuming completely reversible electron  
11 transfer kinetics. For the simulations, the diffusion coefficient was set to  $1.0 \times 10^{-5}$  cm<sup>2</sup>/s, the  
12 reverse saturation current (leakage) to  $1.0 \times 10^{-5}$   $\mu$ A, and the bulk concentration of Fc to  $1.0 \times$   
13  $10^{-3}$  M. (D) Experimental cyclic voltammograms of Fc solutions in acetonitrile ( $1.0 \times 10^{-3}$  M  
14 Fc, 0.1 M Bu<sub>4</sub>NClO<sub>4</sub>, and sweep rate of 0.1 V/s) recorded on a chemically modified LD n-type  
15 Si(111) electrode under variable illumination intensities (1030, 170, 100 and 80 klux). The  
16 current axis in the simulated voltammograms has been normalized to the dimensionless current  
17  $\psi$  (Section S2) and the abscissa represents the voltage,  $E$ , relative to the redox formal potential,  
18  $E^0$ , corrected for the open-circuit potential,  $E_{OC}$  (i.e.,  $E - E^0 - E_{OC}$ ).  
19

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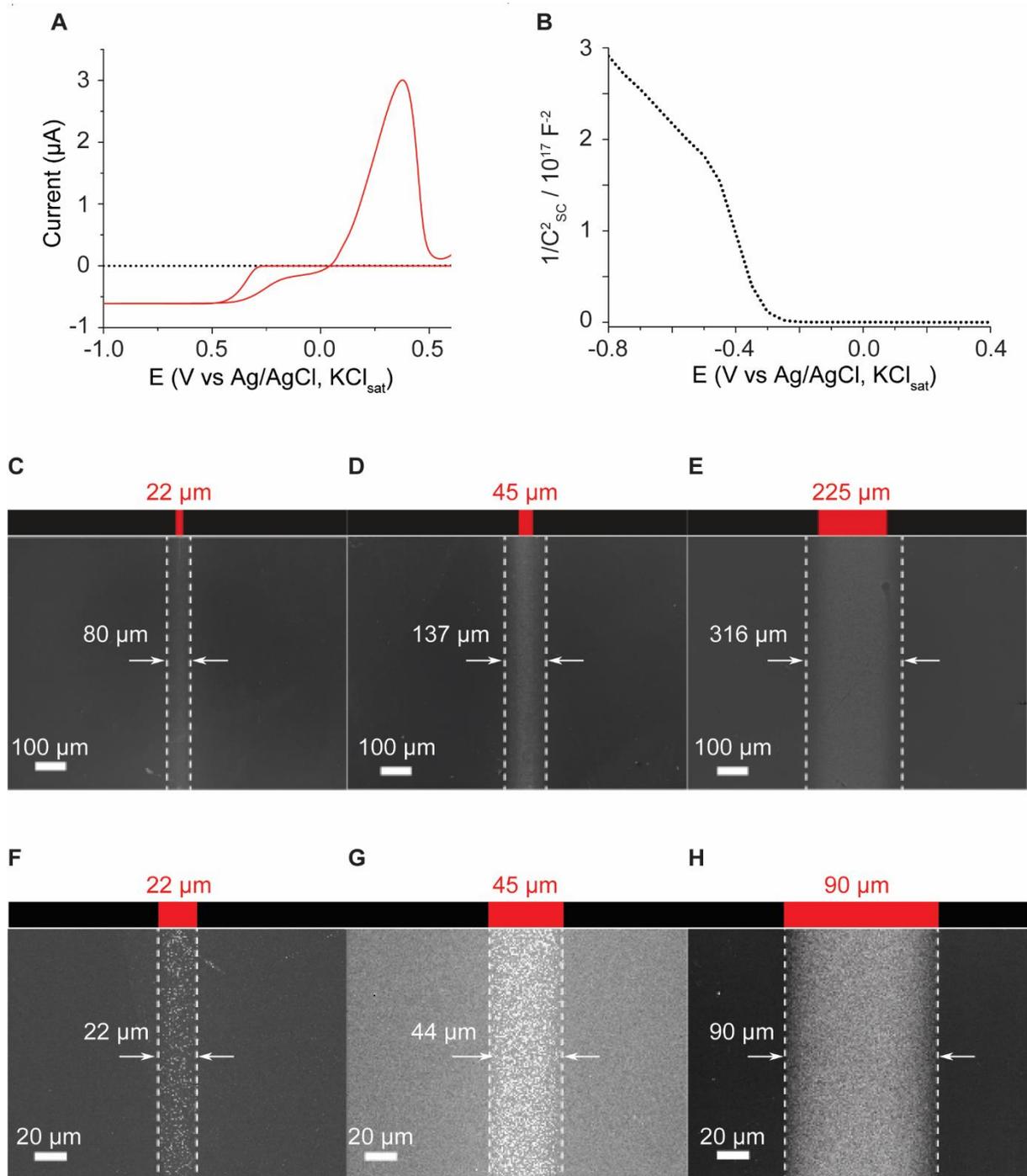
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2 **Fig. 3.** Voltammetric photocurrent–potential behaviour of freely diffusing  $\text{Fc}/\text{Fc}^+$  on Si(111)  
3 electrodes under local illumination. (A) Simulated cyclic voltammograms (equations (24–25)  
4 of the supplementary information, voltage sweep rate of 0.01 V/s) for a band semiconductor  
5 microelectrode as a function of its width ( $w$ , 22.5, 45, 90 and 225  $\mu\text{m}$ ). The photocurrent is 10  
6  $\mu\text{A}$ . (B) Simulated cyclic voltammograms (equations (24–25) of the supplementary information,  
7 0.01 V/s) for a band semiconductor microelectrode as a function of the photocurrent. The  
8 photocurrents are 10, 7, 5, 3 and 1  $\mu\text{A}$ .  $w$  is 22.5  $\mu\text{m}$ . For the simulations a diffusion coefficient  
9 of  $10^{-5} \text{ cm}^2/\text{s}$ , an initial concentration of  $10^{-3} \text{ M}$  of reduced species, and a reverse saturation  
10 current of  $10^{-5} \mu\text{A}$  were used. (C) Experimental cyclic voltammograms (0.01 V/s) of ferrocene  
11 solutions on a chemically modified LD n-type Si(111) electrode under local illumination. A  
12 line-shaped (band) of specific width (22.5, 45, 90, 225 and 450  $\mu\text{m}$ ) was projected on the  
13 interface. Cyclic voltammograms were recorded in acetonitrile with 0.1 M  $\text{Bu}_4\text{NClO}_4$  and  $1.0$   
14  $\times 10^{-3} \text{ M}$  of ferrocene. (D) Schematics of the silicon–electrolyte interface under localized  
15 illumination (red area). The ratio of the width of the illuminated area ( $w$ ) to the diffusion length  
16 ( $L$ ) is the key parameter to understand and predict the electrochemical behavior under local  
17 illumination of a semiconductor electrode. As  $w$  decreases, and becomes comparable to, or  
18 smaller than  $2L$ , carriers diffusing away from the generation site become the main contributors  
19 to the electrochemical current, and have therefore to be accounted for.  
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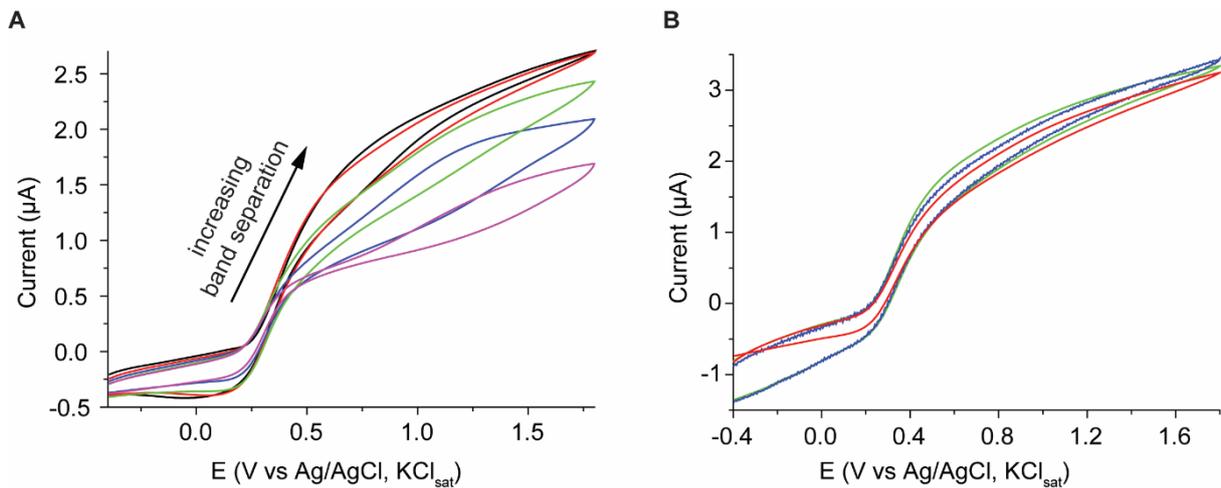
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2 **Fig. 4.** Cyclic voltammetry of light-addressed microelectrode band arrays on crystalline silicon.  
3 (A) Schematics representing the diffusion fronts of a microelectrode band array. (B) Simulated  
4 cyclic voltammograms (equations (24–25) of the supplementary information, voltage sweep  
5 rate is 0.1 V/s) for an array of semiconducting band microelectrodes as a function of the inter-  
6 band separation. The inter-band separations are 112.5 (black line), 11.25  $\mu\text{m}$  (red line) and  
7 0.225  $\mu\text{m}$  (green line). The photocurrent is set to 5  $\mu\text{A}$ . (C) Experimental voltammograms (0.1  
8 V/s) recorded when projecting arrays of 22.5  $\mu\text{m}$  wide line-shaped bands separated by gaps of  
9 225  $\mu\text{m}$  (red line) or 0  $\mu\text{m}$  (black line). Dotted data were obtained under dark conditions.  
10 Chemically modified LD n-type Si(111) surfaces were used as the working electrode. The  
11 electrolyte was acetonitrile with 0.1 M  $\text{Bu}_4\text{NClO}_4$  and  $1.0 \times 10^{-3}$  M ferrocene. (D) Simulated  
12 cyclic voltammograms ((equations (24–25) of the supplementary information, 0.1 V/s) for an  
13 array of semiconducting band microelectrodes as a function of the inter-band separation.  
14 Separations are 112.5 (black line), 11.25  $\mu\text{m}$  (red line) and 0.225  $\mu\text{m}$  (green line). The  
15 photocurrent is set to 1  $\mu\text{A}$ , the diffusion coefficient to  $1.0 \times 10^{-5}$   $\text{cm}^2/\text{s}$ , the concentration of  
16 reducible species to  $1 \times 10^{-3}$  M, and the reverse saturation current to  $1.0 \times 10^{-5}$   $\mu\text{A}$ .  
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1  
2 **Fig. 5.** Lateral diffusion as function of diffusion lengths. Redox mask-free lithography on  
3 silicon (Cu<sub>2</sub>O patterns). (A) Cyclic voltammetry (0.1 V/s) under dark (symbols) and light (solid  
4 line) on alkyne modified LD p-Si(100) electrodes. (B) Mott-Schottky plot used to estimate the  
5 flat-band potential for the alkyne modified LD p-Si(100) electrodes. (C)–(H) Scanning electron  
6 microscopy images of Cu<sub>2</sub>O patterns deposited on either crystalline silicon (C)–(E), or on  
7 amorphous silicon (F)–(H). Patterns were formed by projecting on the electrode a line-shaped  
8 light feature of width specified by the red ink labels placed above the micrographs. The  
9 experimental width of the Cu<sub>2</sub>O pattern is specified by the white ink labels. The electrolyte was  
10 an aqueous solution of  $5.0 \times 10^{-2}$  M of CuSO<sub>4</sub> and 0.5 M of K<sub>2</sub>SO<sub>4</sub>. The total charge transferred  
11 per unit of illuminated area was kept constant ( $1 \mu\text{C} \cdot \mu\text{m}^{-2}$ ). The electrodeposition potential was  
12 stepped from the open circuit to a potential of  $-0.5$  V.  
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1  
2 **Fig. 6.** Microelectrode arrays generated on amorphous silicon electrodes. Forming transient  
3 microelectrodes arrays with light in a material with short charge carriers' diffusion length. (A)  
4 Experimental cyclic voltammograms (0.1 V/s) of ferrocene solutions when projecting an array  
5 of 13.5 μm wide light bands with variable inter-band separation (270, 135, 68, 27 and 0 μm,  
6 shown as black, red, green, blue and magenta solid lines, respectively). (B) Experimental  
7 voltammetry of ferrocene at different voltage scan rates (0.3, 0.05 and 0.025 V/s, shown as  
8 green, blue and red solid lines, respectively) when projecting a fixed band array (13.5 μm band  
9 width and band separation of 270 μm). Chemically modified amorphous silicon was used as  
10 the working electrode and the electrolyte is 0.1 M Bu<sub>4</sub>NClO<sub>4</sub> in acetonitrile with 1.0 × 10<sup>-3</sup> M  
11 ferrocene.  
12

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## 1 **Authors' contributions**

2 S.C., J.G. and Y.B.V conceived and designed the research. Y.B.V. performed the experiments,  
3 analyzed and visualized the data, with assistance from J.G. and A.M. Codes for the digital  
4 simulations of the voltammetry were developed, written and executed by J.G., with assistance  
5 from A.M.. All authors contributed to the writing and editing of the manuscript.

## 7 **Declaration of competing interest**

8 The authors declare that they have no known competing financial interests or personal  
9 relationships that could have appeared to influence the work reported in this paper.

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## 16 **Supplementary materials**

17 Supplementary material associated with this article can be found in the online version.

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